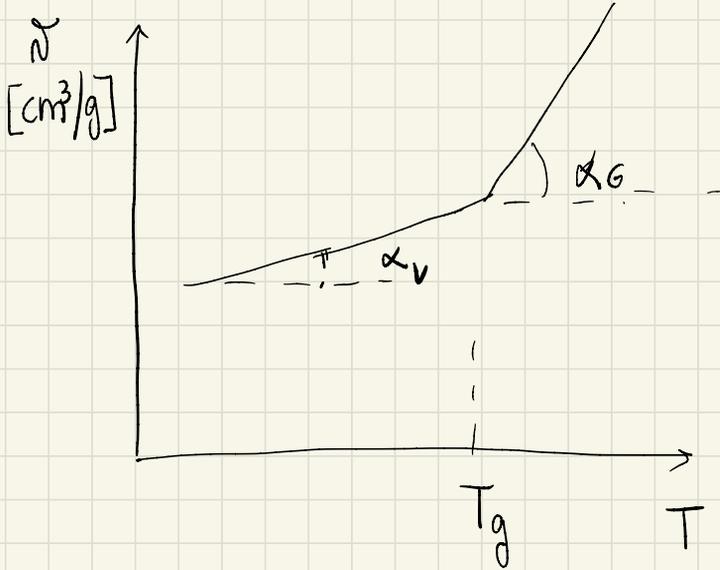


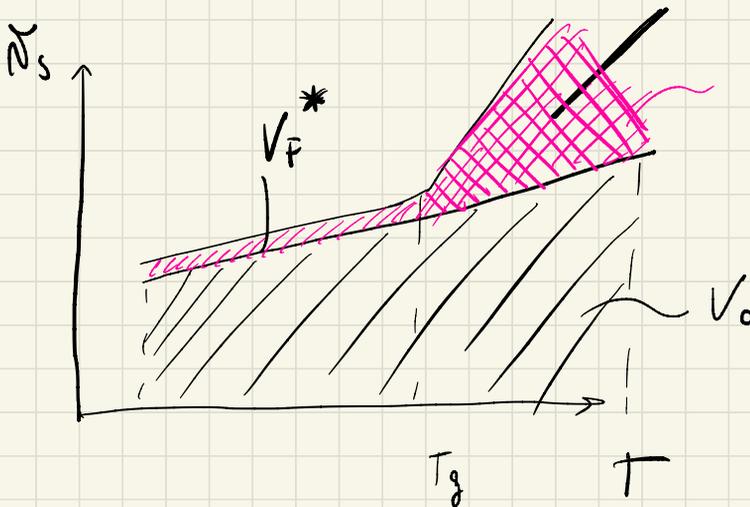
LEZIONE 9

18.10.2022

POLIMERO AMORFO



$$\alpha_v = \frac{1}{V} \frac{\partial V}{\partial T}$$



VOLUME
LIBERO
 V_F

IL VOLUME
REALMENTE
OCCUPATO
dagli atomi

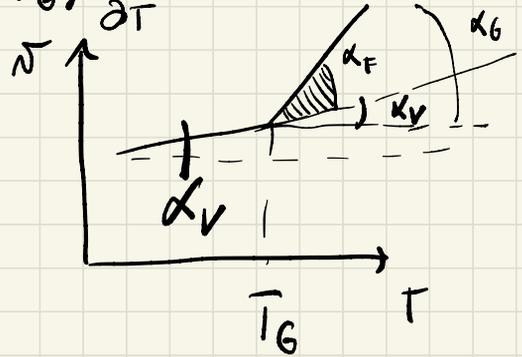
$$V_F = V_F^* + (T - T_G) \frac{\partial V}{\partial T}$$

$$\alpha_F = \alpha_G - \alpha_V$$

$$f_F = \frac{V_F}{V}$$

$$f_F^* = \frac{V_F^*}{V}$$

$$\alpha_F \approx \frac{1}{V} \frac{\partial V}{\partial T}$$



$$F_F = f_F^* + (T - T_G) \alpha_F$$

Per $T \leq T_G$ $F_F = f_F^*$

$T > T_G$
 $F_F = f_F$

$$f_F^* = 2.5\%$$

Eq. ne di William - Landell - Ferry WLF

$(AB)_m$

$$f_V^A = f_g^A + (T - T_g^A) a_f^A$$

(16.11)

COPOLIMERI

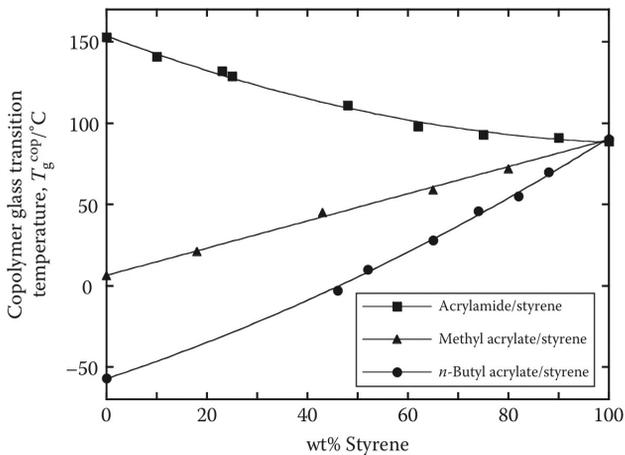


FIGURE 16.5 Variation of copolymer glass transition temperature with the weight fraction of styrene for various styrene-based statistical copolymers. (Data taken from Illers, K.H., *Kolloid-Zeitschrift.*, 190, 16, 1963.)

$$F_V^A = F_g^A + (T - T_g^A) \alpha_F^A$$

$$F_V^B = F_g^B + (T - T_g^B) \alpha_F^B$$

$$F_V^{cop} = F_g^{cop} + (T - T_g^{cop}) \alpha_F^{cop}$$

$$\alpha^{cop} = W_A \alpha_F^A + W_B \alpha_F^B$$

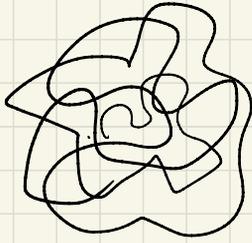
$(AB)_m$

$$W_A = \frac{M_A}{M_{TOT}}$$

$$W_B = \frac{M_B}{M_{TOT}}$$

$$\frac{1}{T_g^{cop}} \approx \frac{W_A}{T_g^A} + \frac{W_B}{T_g^B}$$

T_g^{cop} e α^{cop} dipendono dalle frazioni ponderali

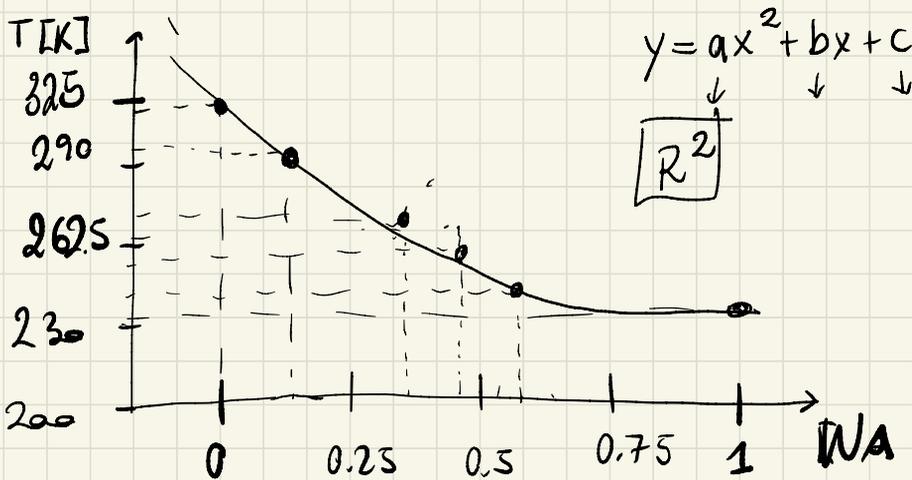


RANDOM COIL

16.4 The following data were obtained for the values of glass transition temperature T_g of poly[(vinylidene fluoride)-*co*-chlorotrifluoroethylene] as a function of the weight fraction w_A of vinylidene fluoride. By fitting these data to a suitable curve estimate, the value of T_g for a copolymer with $w_A = 0.75$.

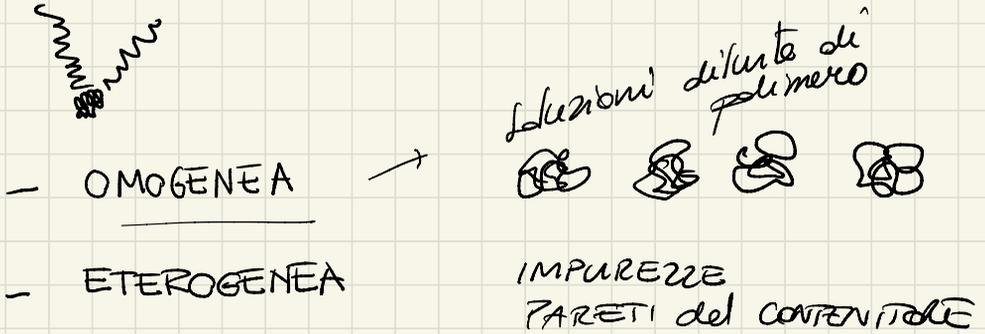
| w_A | 0 | 0.14 | 0.35 | 0.4 | 0.54 | 1.0 |
|---------|-----|------|------|-----|------|-----|
| T_g/K | 319 | 292 | 270 | 265 | 258 | 235 |

$$0^\circ\text{C} = -273.15\text{K}$$



Cristallizzazione e fusione

↳ NUCLEAZIONE e ACCRESCIMENTO



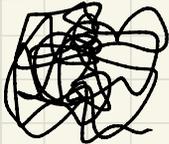
→ PROCESSO TERMODINAMICAMENTE FAVORITO

$$G = H - TS \rightarrow \begin{matrix} \nearrow \text{temperatura} \\ \text{Entropia} \\ \searrow \text{Entalpia (calore latente)} \end{matrix}$$

Energia libera di Gibbs

$\Delta G \rightarrow 0$ più il processo è favorito

FUSO POLIMERICO



Alta S



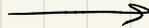
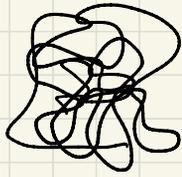
SOLIDO

AMORFO

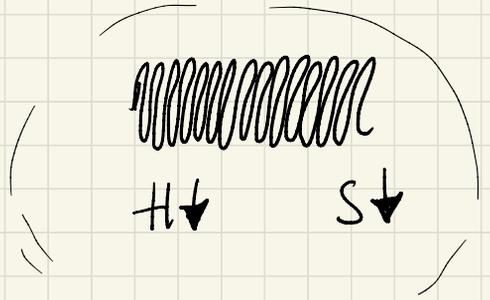


H = cost

S



SOLIDO CRISTALLINO



$$\Delta G = \Delta H - T\Delta S$$



CRISTALLINITA'
del MATERIALE



FASE AMORFA

Viene favorita la cristallizzazione

In condizioni quasi-stazionarie

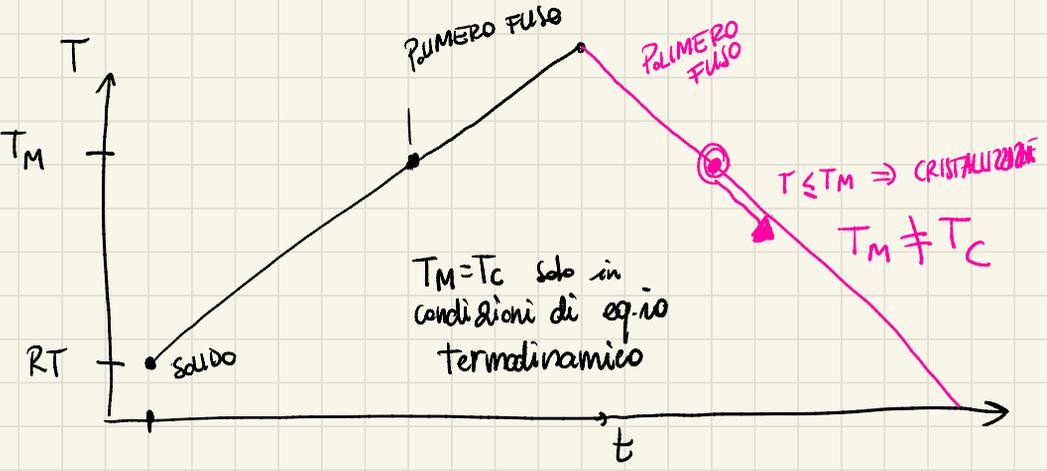
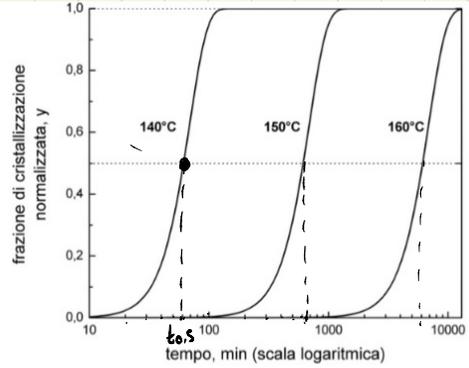
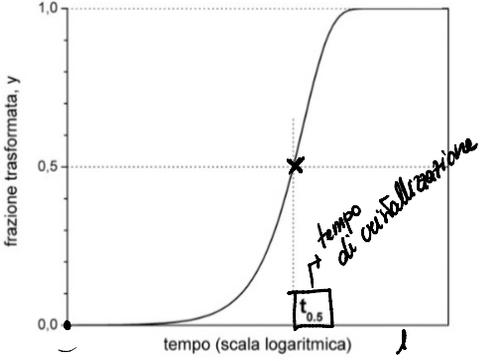
Equazione di Avrami

FRIZIONE
di MATERIALE
CRISTALLIZZATO

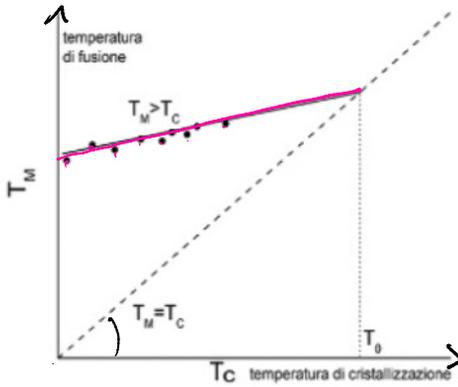
$$y = 1 - e^{-kt^m}$$

K, m
costanti che dipendono
dal materiale

Se $T_M = 200^\circ\text{C}$



2. Temperatura di fusione e temperatura di cristallizzazione



Se lavoro in condizioni di
Eq. 10 TD $\Rightarrow T_C = T_M$
($y = x$)

Se non lavoro in condizioni
di Eq. 10 TD $\Rightarrow T_M > T_C$

Fig. 6.4 - Determinazione della temperatura T_0 di transizione solido-liquido per un cristallo polimerico perfetto e di dimensioni molto grandi (teoricamente infinite).

DSC: Differential Scanning Calorimetry

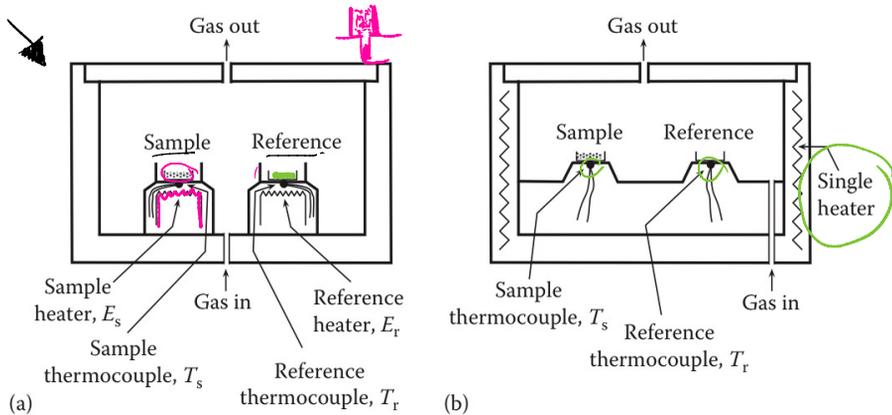


FIGURE 17.33 Schematic diagrams showing the key features of cell design for (a) power-compensation DSC and (b) heat-flux DSC. The sample is shown contained in an aluminium pan with an empty aluminium pan as the reference. T_s and T_r are, respectively, the temperatures of the sample and reference. In power-compensation DSC, the separate energy inputs to the sample and reference, E_s and E_r respectively, are adjusted such that $T_s = T_r$ as both cells are heated or cooled. Heat-flux DSC employs a single heater and $\Delta T (= T_s - T_r)$ is monitored as the cell is heated or cooled. The atmosphere can be controlled by flowing a gas through the cell enclosure.

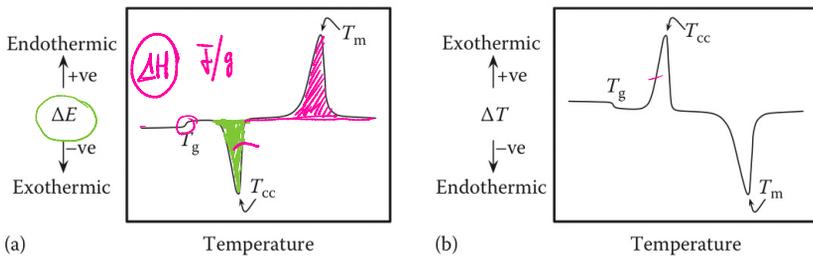


FIGURE 17.34 Schematic diagrams showing (a) power-compensation and (b) heat-flux DSC traces expected for an amorphous sample of a crystallizable polymer, where T_g is the glass transition temperature, T_{cc} is the cold crystallization temperature (i.e. for crystallization in the solid state) and T_m is the melting temperature.

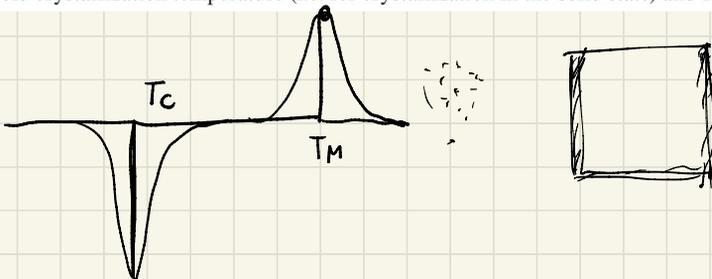


TABLE 17.4

Approximate Values of Melting Temperature, T_m , for Various Polymers

| Repeat Unit | $T_m / ^\circ\text{C}$ |
|---|------------------------|
| $-\text{CH}_2-\text{CH}_2-$ | 137-146 |
| $-\text{CH}_2-\text{CH}_2-\text{O}-$ | 67 |
| $-\text{CH}_2-\text{CH}_2-\text{CO}-\text{O}-$ | 122 |
| $-\text{H}_2\text{C}-\text{C}_6\text{H}_4-\text{CH}_2-$ | 397 |
| $-\text{CH}_2-\text{CH}_2-\text{CO}-\text{NH}-$ | 330 |
| $-\text{CH}_2-\text{CH}_2-\text{CH}_2-\text{CO}-\text{NH}-$ | 260 |
| $-\text{CH}_2-\text{CH}_2-\text{CH}_2-\text{CH}_2-\text{CO}-\text{NH}-$ | 258 |
| Side group (X) | |
| $-\text{CH}_2-\text{CHX}-$ | |
| $-\text{CH}_3$ | 187 |
| $-\text{CH}_2-\text{CH}_3$ | 125 |
| $-\text{CH}_2-\text{CH}_2-\text{CH}_3$ | 78 |
| $-\text{CH}_2-\text{CH}(\text{CH}_3)_2$ | 235 |
| $-\text{C}_6\text{H}_5$ | 240 |

Handwritten notes:
 - "grasso AMMIDICO" written next to the amide repeat units.
 - A skeletal structure of a branched alkane (isobutyl) is drawn below the side group table.
 - A benzene ring with a dashed circle around it is labeled "PS" (polystyrene).

TABLE 16.1

Approximate Values of Glass Transition Temperature, T_g , for Various Polymers

| Repeat Unit | Polymer Name | $T_g / ^\circ\text{C}$ |
|---|---|------------------------|
| $-\text{CH}_2-\text{CH}_2-$ | Polyethylene | -130 to -10 |
| $-\text{CH}_2-\text{CH}_2-\text{O}-$ | poly(ethylene oxide) | -67 |
| $-\text{H}_2\text{C}-\text{C}_6\text{H}_4-\text{CH}_2-$ | poly(<i>p</i> -xylylene) | 80 |
| Vinyl polymers | | |
| $-\text{CH}_2-\text{CHX}-$ | Side group (X) | |
| | $-\text{CH}_3$ | -23 |
| | $-\text{CH}_2-\text{CH}_3$ | -24 |
| | $-\text{CH}_2-\text{CH}_2-\text{CH}_3$ | -40 |
| | $-\text{CH}_2-\text{CH}(\text{CH}_3)_2$ | 30 |
| | $-\text{C}_6\text{H}_5$ | 100 |
| | $-\text{Cl}$ | 81 |
| | $-\text{OH}$ | 85 |
| | $-\text{CN}$ | 97 |

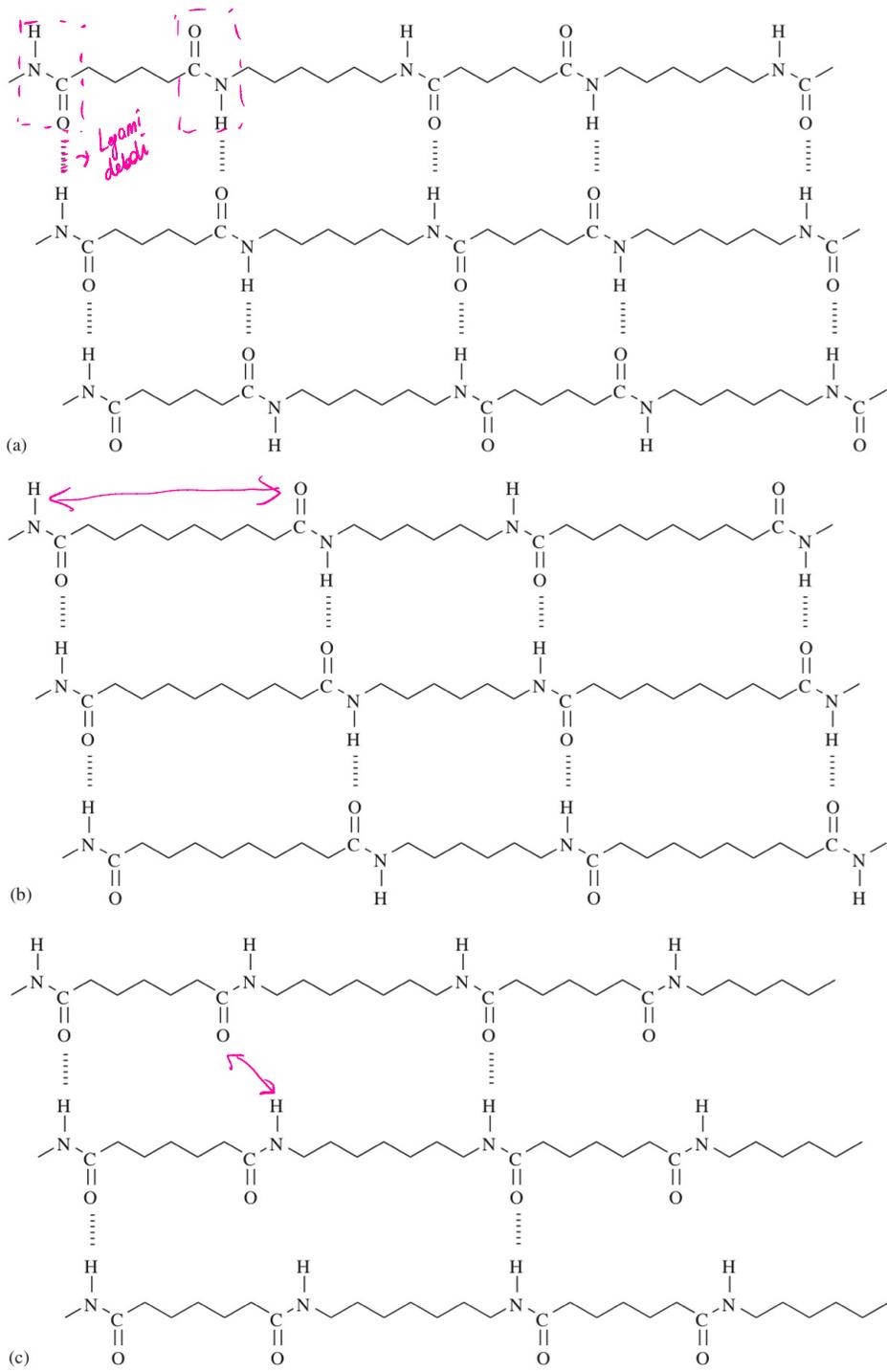
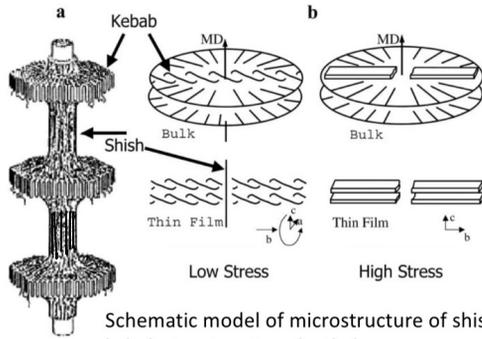
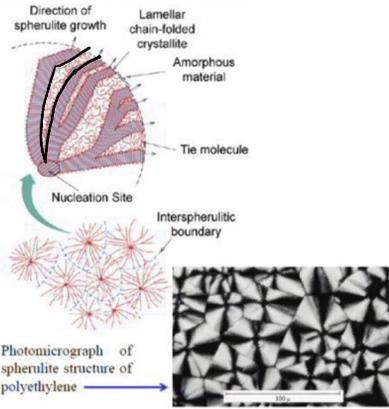


FIGURE 17.37 Differences in patterns of hydrogen bonding in crystals of various nylon polymers: (a) nylon 6.6, $T_m = 265^\circ\text{C}$; (b) nylon 6.10, $T_m = 222^\circ\text{C}$; (c) nylon 7.7, $T_m = 210^\circ\text{C}$.

Struttura cristalline

Spherulites: Aggregates of lamellar crystallites ~ 10 nm thick, separated by amorphous material. Aggregates are formed upon solidification from a melted state and are approximately spherical in shape.

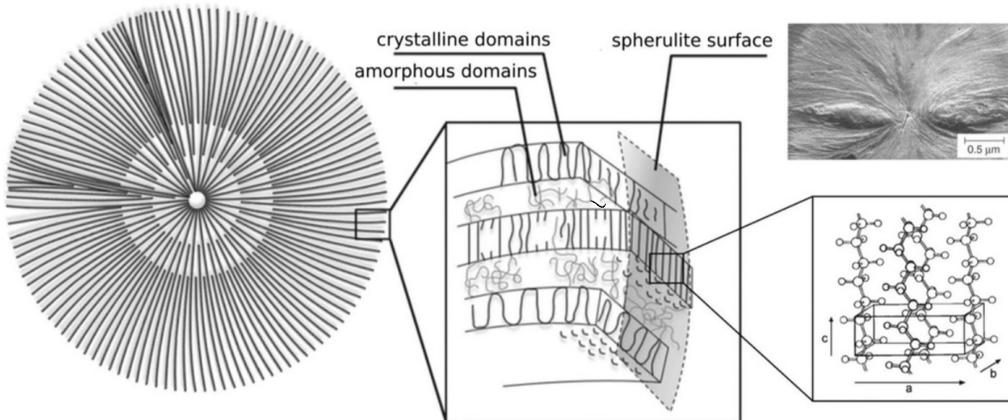


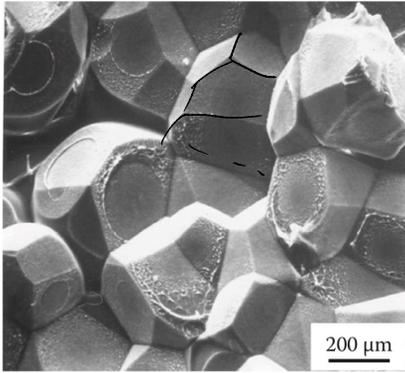
Schematic model of microstructure of shish-kebab structure in polyethylene. Note the example of lamellar or kebab twisting in low stress conditions.

[From Processing of Polymers, edited by H.E.H. Meijer, 1997;18:189.

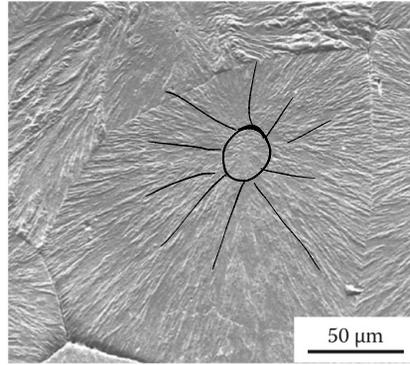
Copyright (1997) VCH, New York].

Struttura cristalline





(a)



(b)

FIGURE 17.14 Scanning electron micrographs of polypropylene spherulites. (a) Fracture surfaces of a brittle sample of low molar mass polymer showing interspherulitic fracture. (Courtesy of K. Friedrich, University of Kaiserslautern, Kaiserslautern, Germany.) (b) An etched surface of polypropylene crystallized at 128°C showing the structure of the individual spherulites and the spherulite boundaries. (Courtesy of M. Burke, University of Manchester, Manchester, U.K.)