

Superconductors in magnetic fields: vortex pinning and relevance to rf applications

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Microwave group and collaborations



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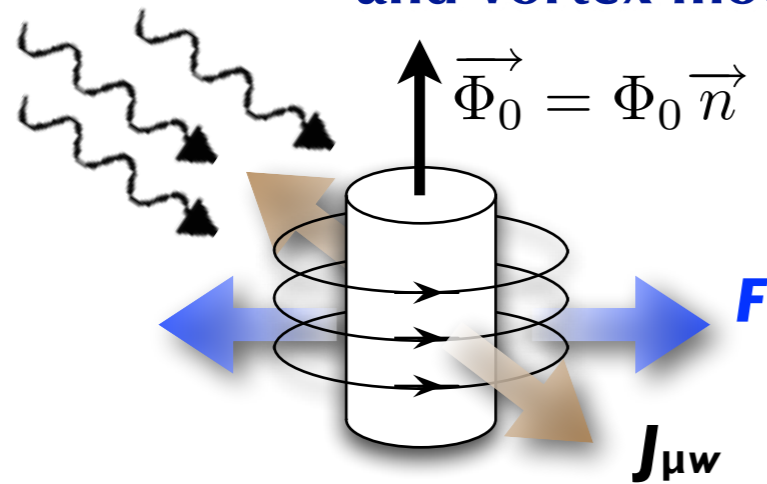


Collaborations



1

Superconductors, microwaves and vortex motion



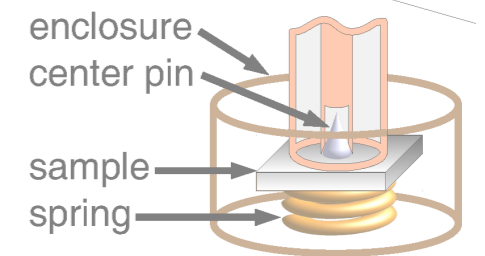
2



What can we learn

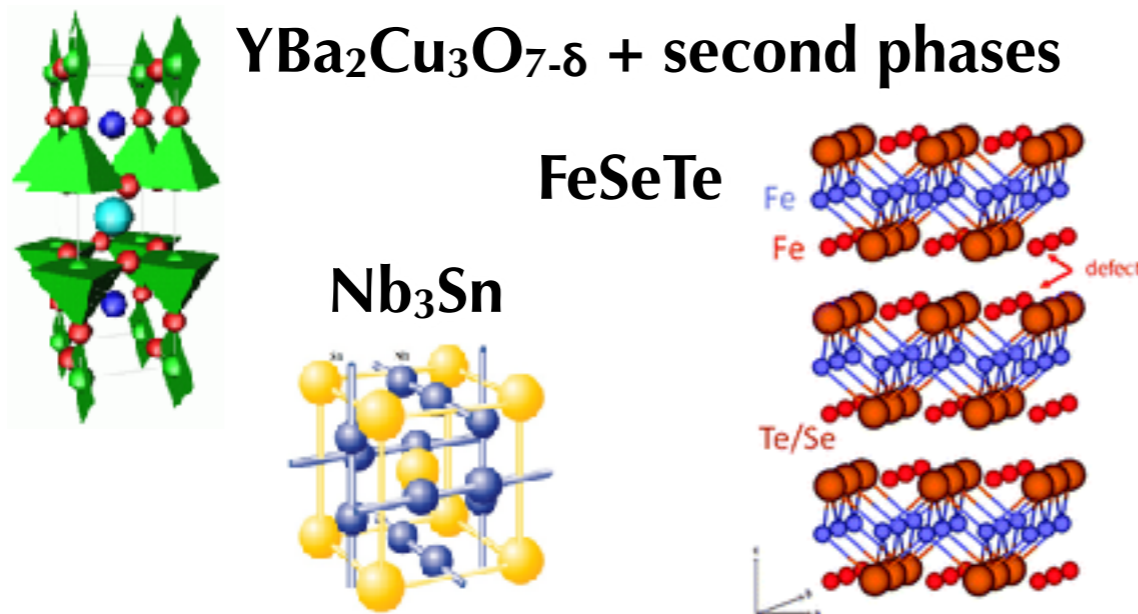
3

Experimental techniques



4

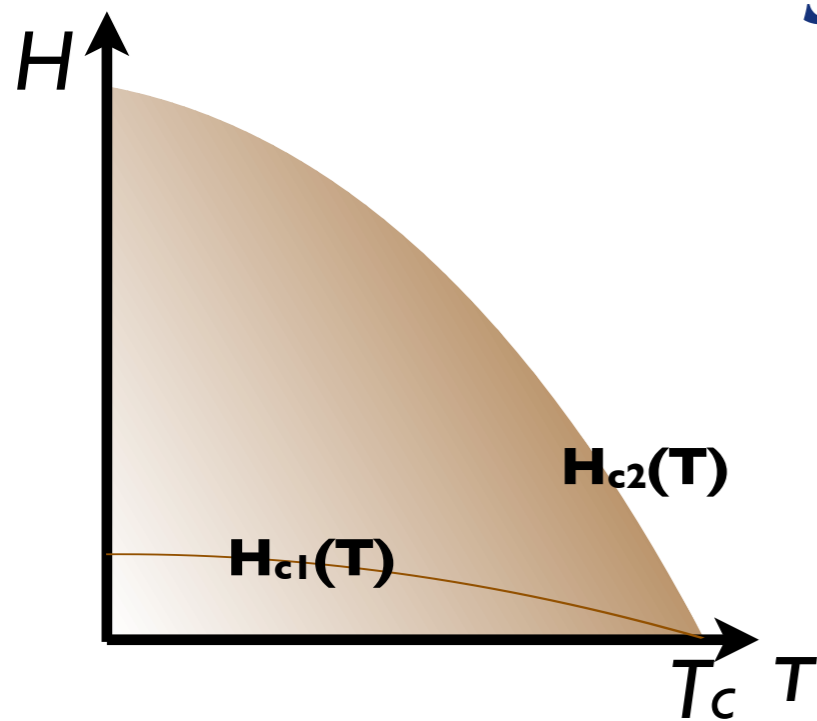
Results (a partial overview)

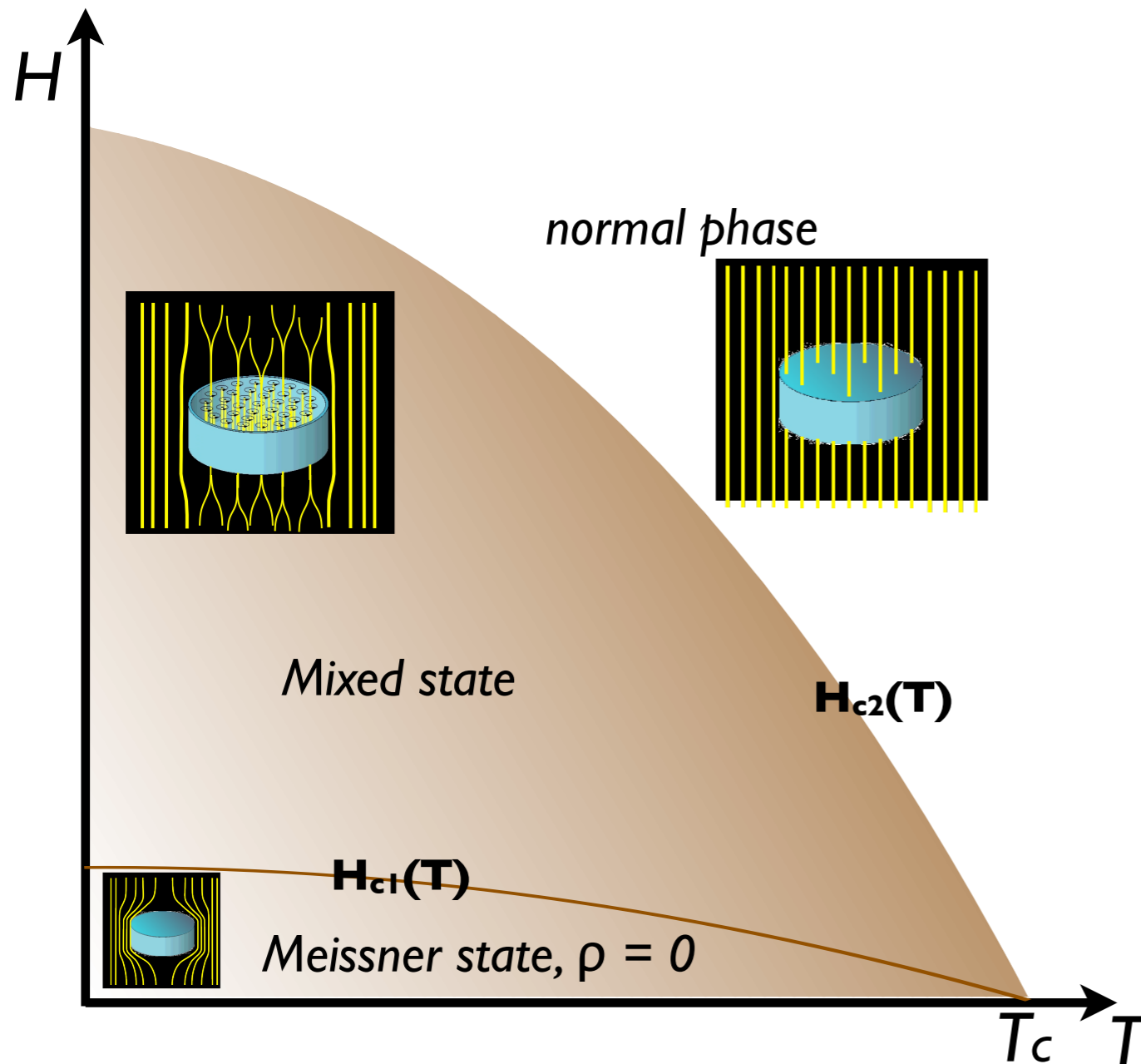


5

Perspectives for rf in high magnetic fields

Superconductors



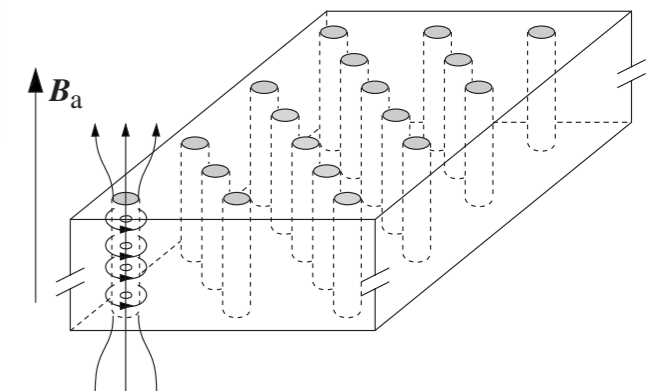
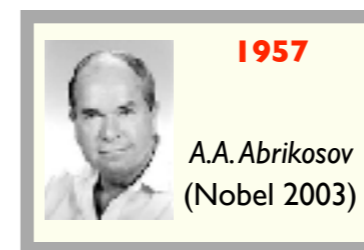


Three regions:

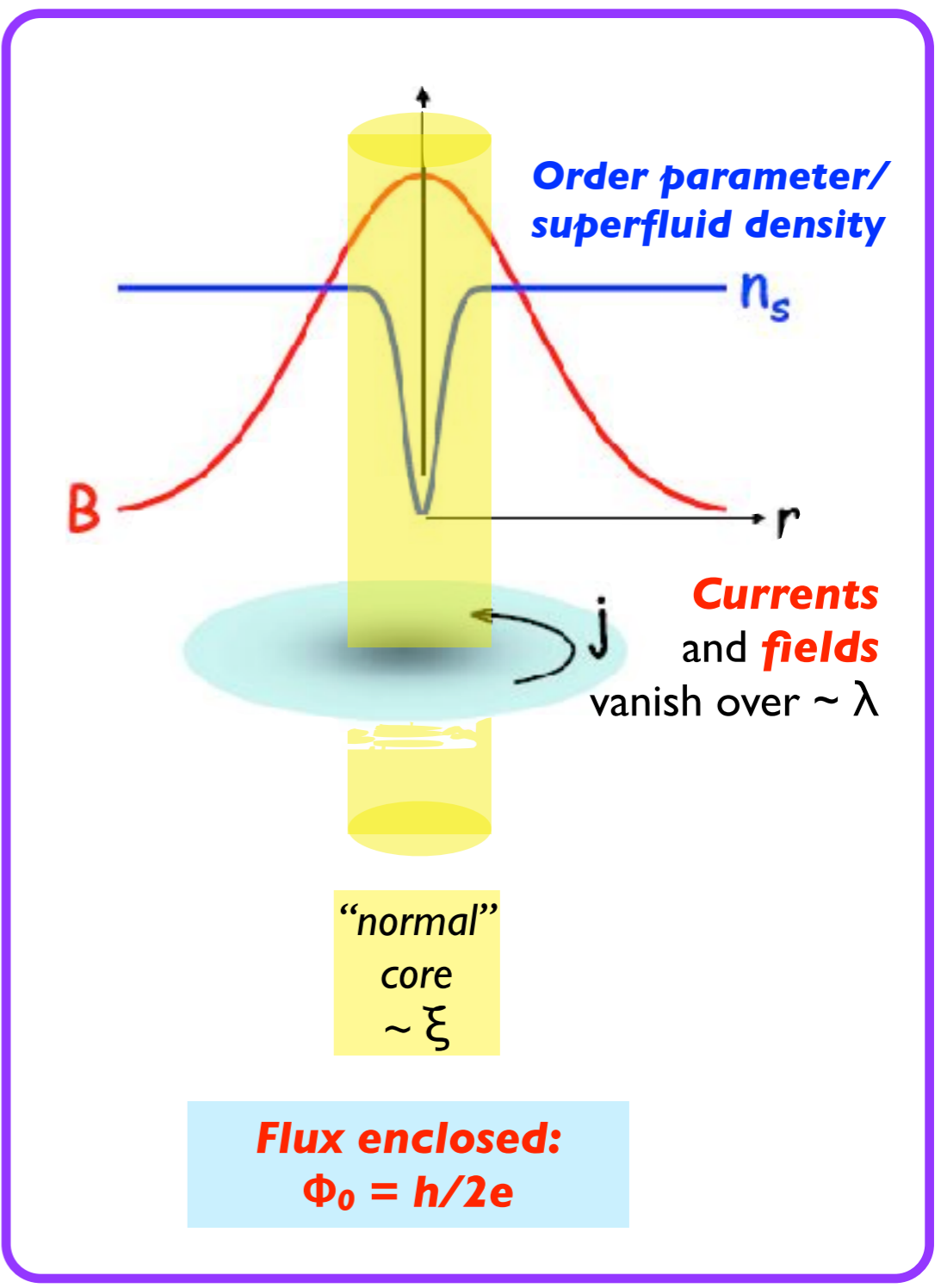
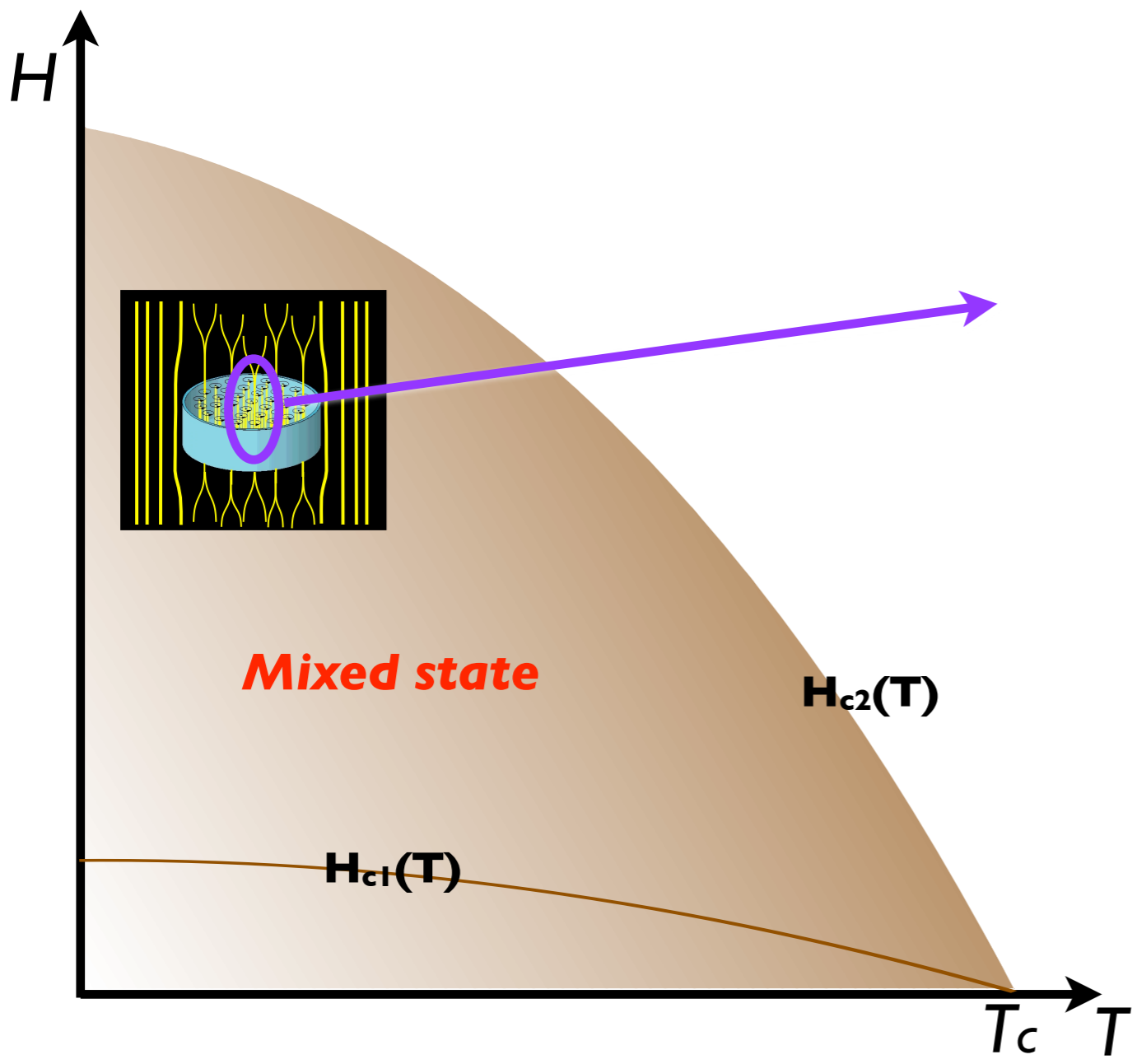
- Meissner state, $H < H_{c1}(T)$
- Mixed state, $H_{c1}(T) < H < H_{c2}(T)$
- Normal state, $H_{c2}(T) < H$

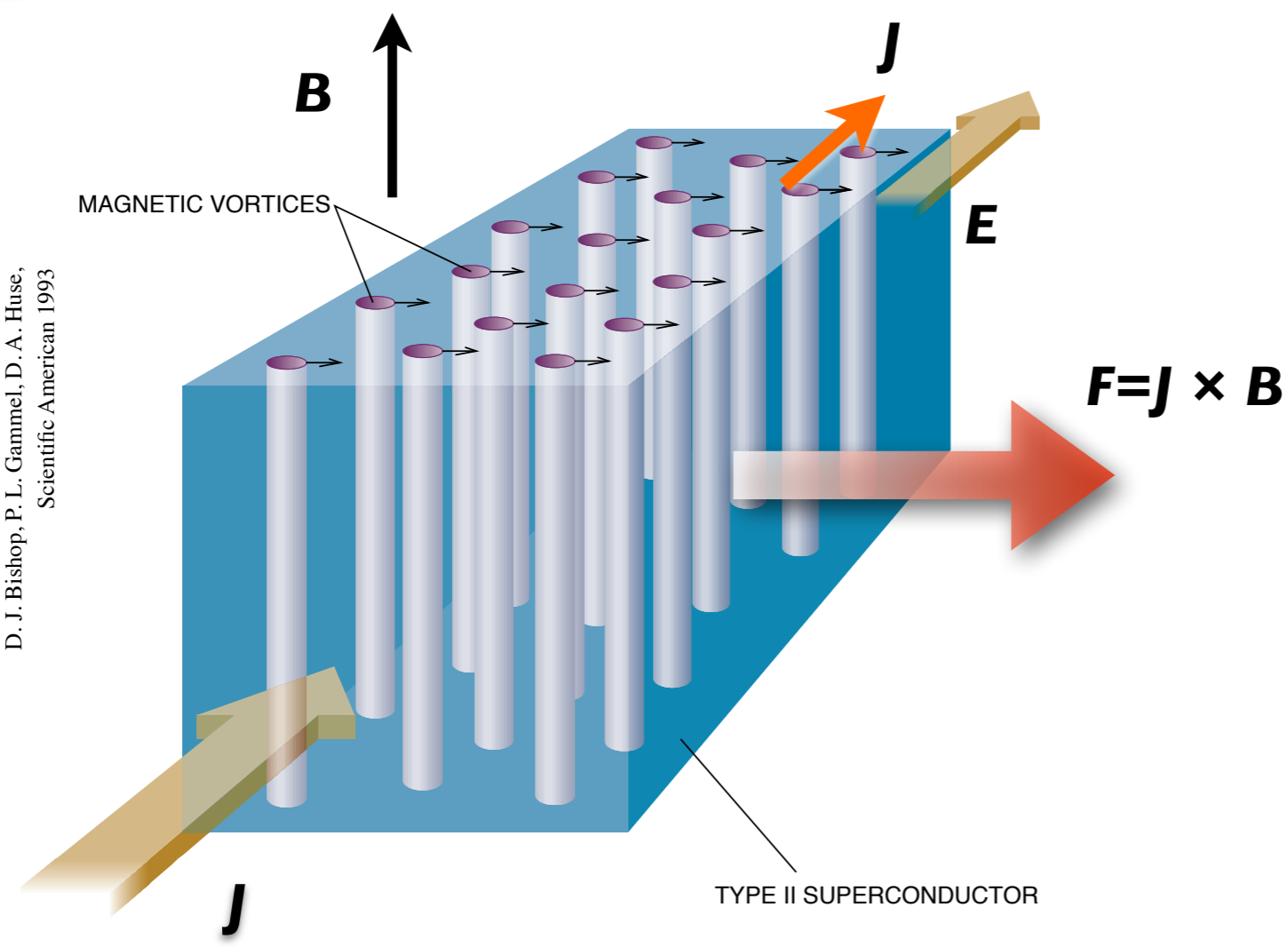
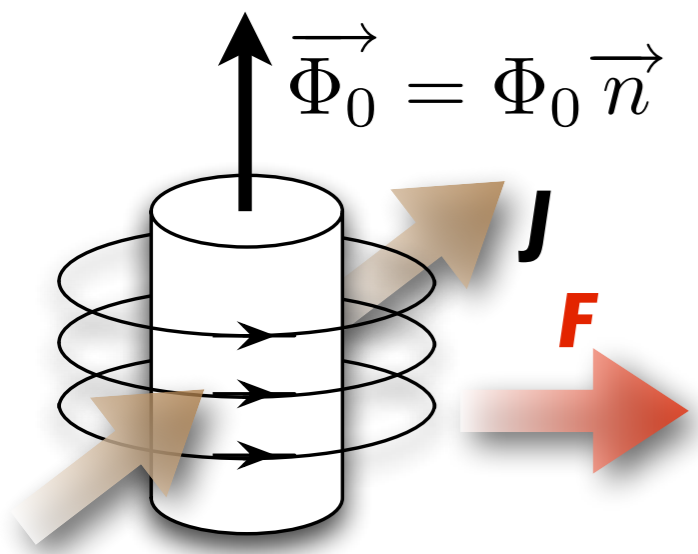
Mixed state, $H_{c1}(T) < H < H_{c2}(T)$:

- Partial penetration of the flux
- Flux quanta ("fluxons", "vortices")
- "Vortex lattice"

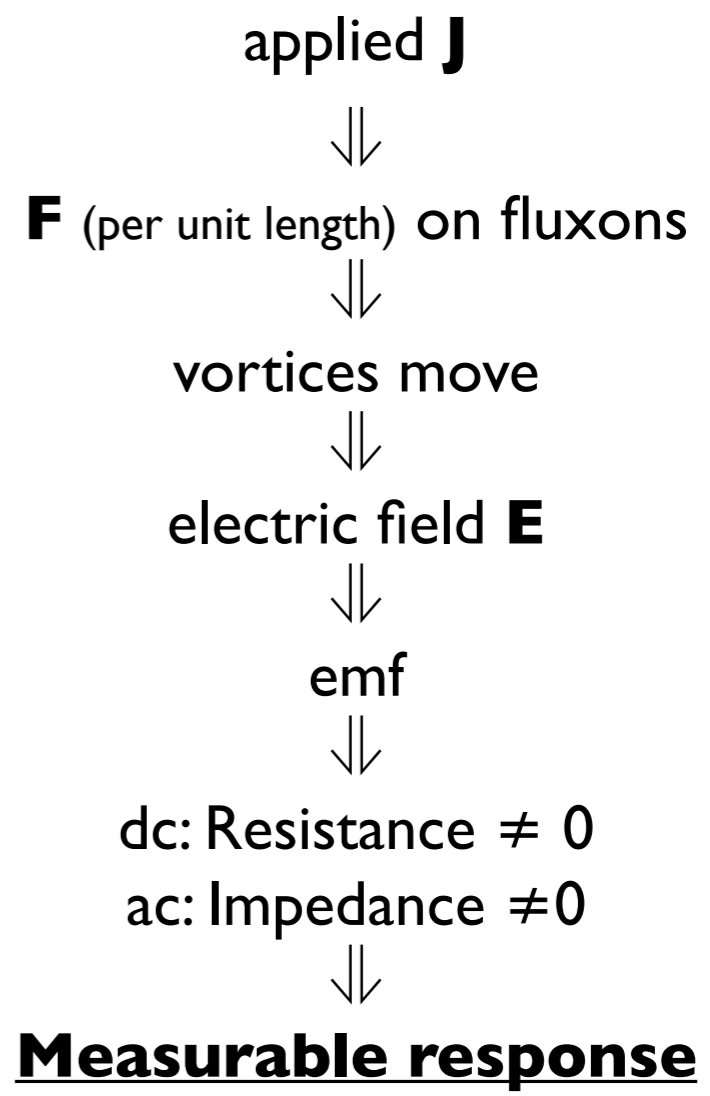


Fluxons / Vortices



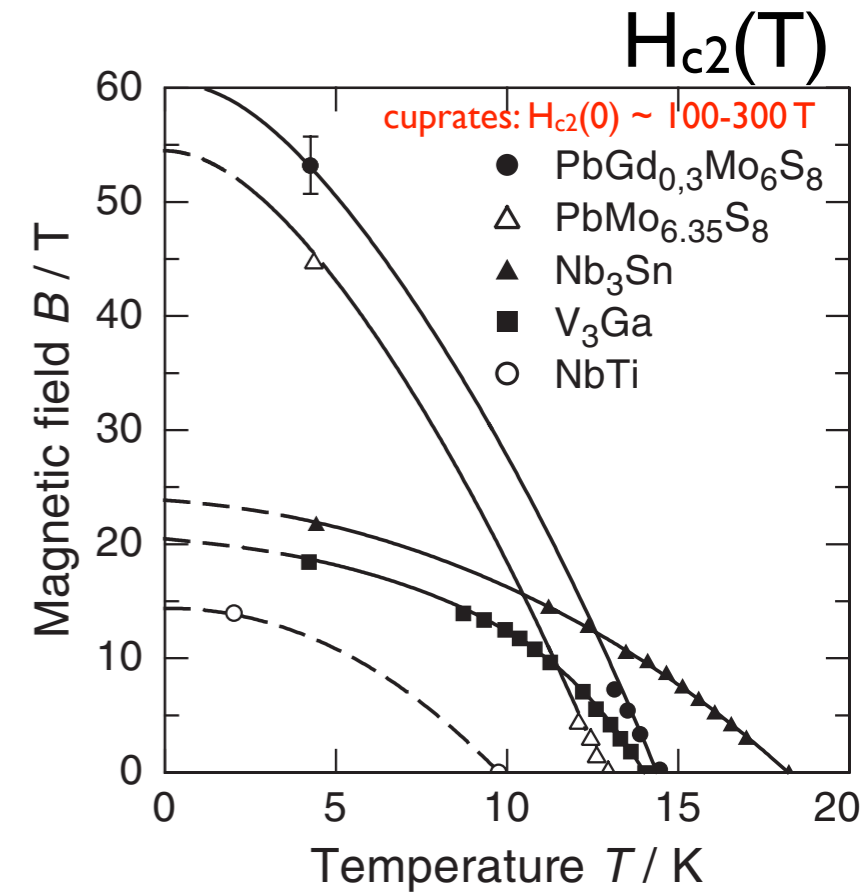
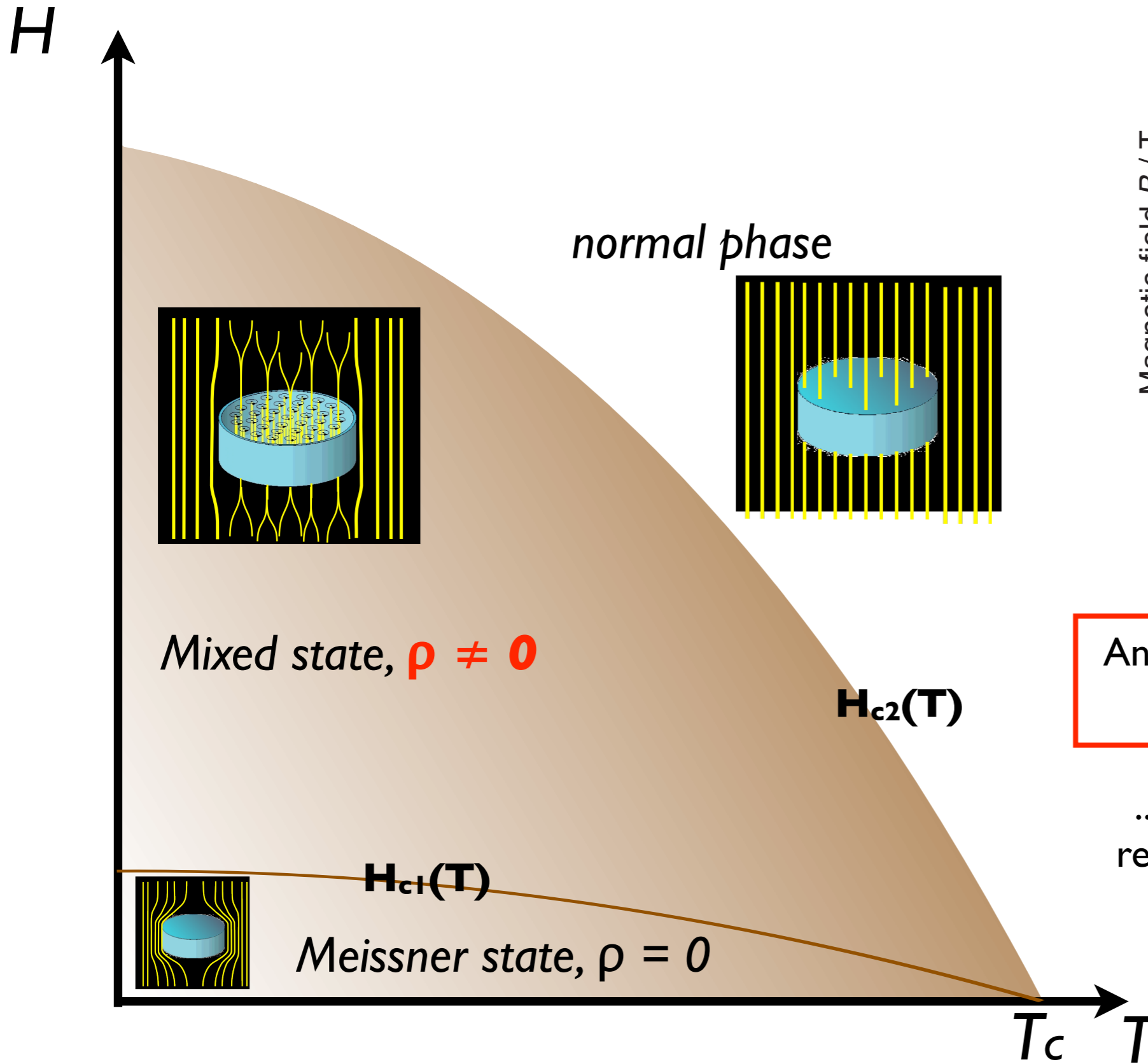


D. J. Bishop, P. L. Gammel, D. A. Huse,
Scientific American 1993



free motion: "flux-flow"

Ideal type-II superconductors



An ideal type-II SC is *resistive* in the mixed state!

... but the normal phase is reached at large/huge fields.

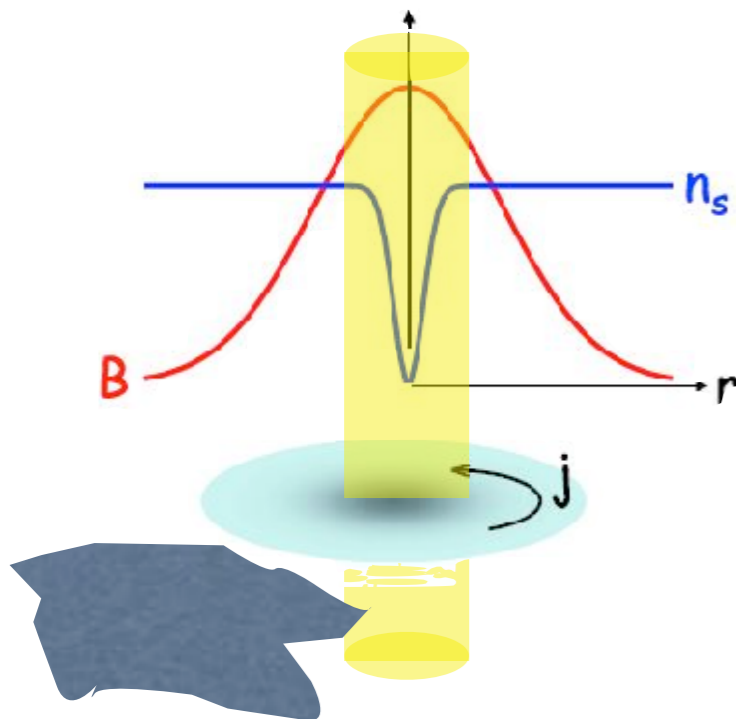
⇒ worth some work!

Stop the fluxons!

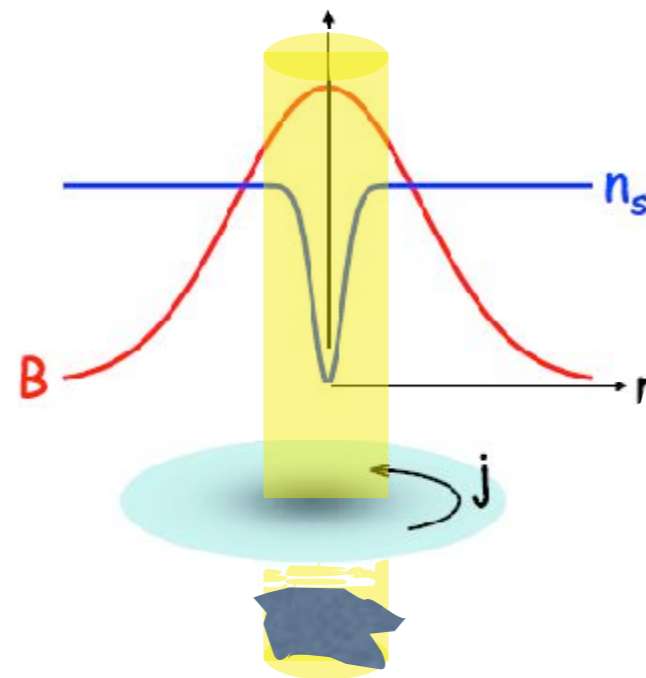
Moving fluxons: dissipation

hinder fluxon motion:
“pinning”

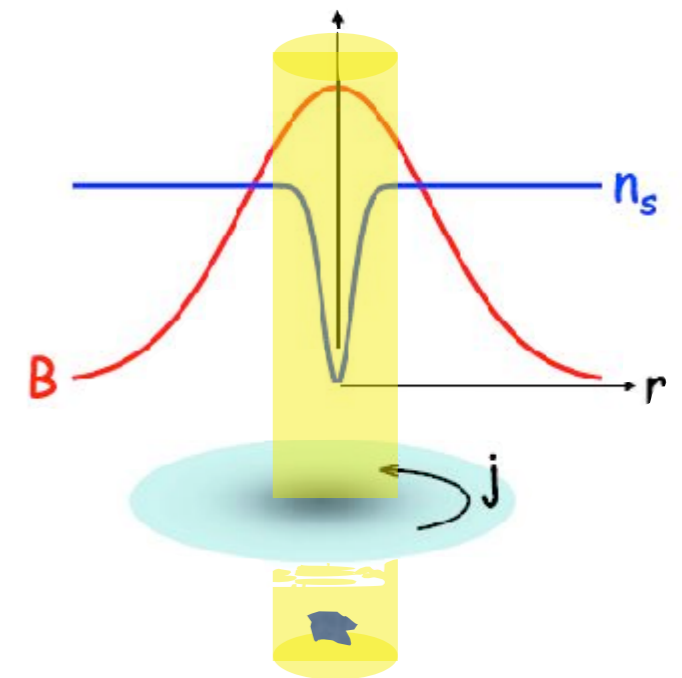
“suitable” defects: preferential sites for the normal cores



too large: size $\gg \xi$



efficient: size $\sim \xi$



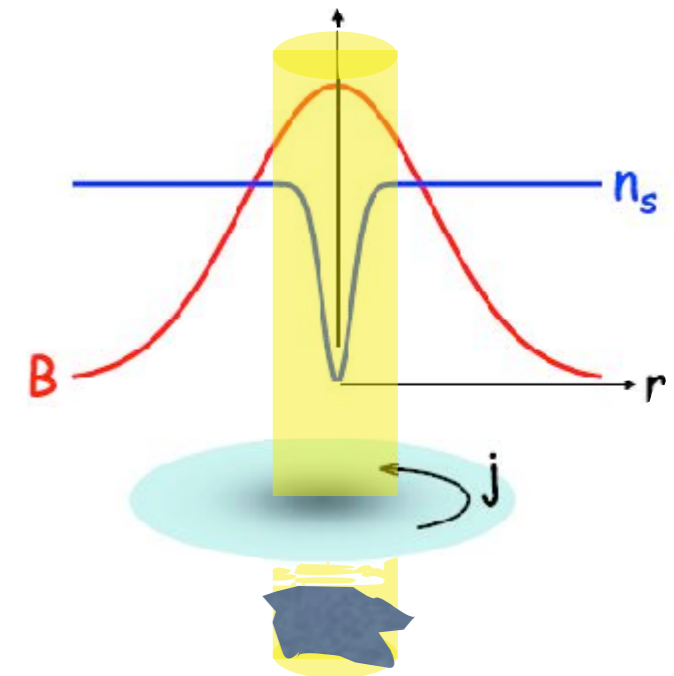
too small: size $\ll \xi$

ξ : “magic size” for defects

hinder fluxon motion:
“pinning”

ξ : “magic size” for defects

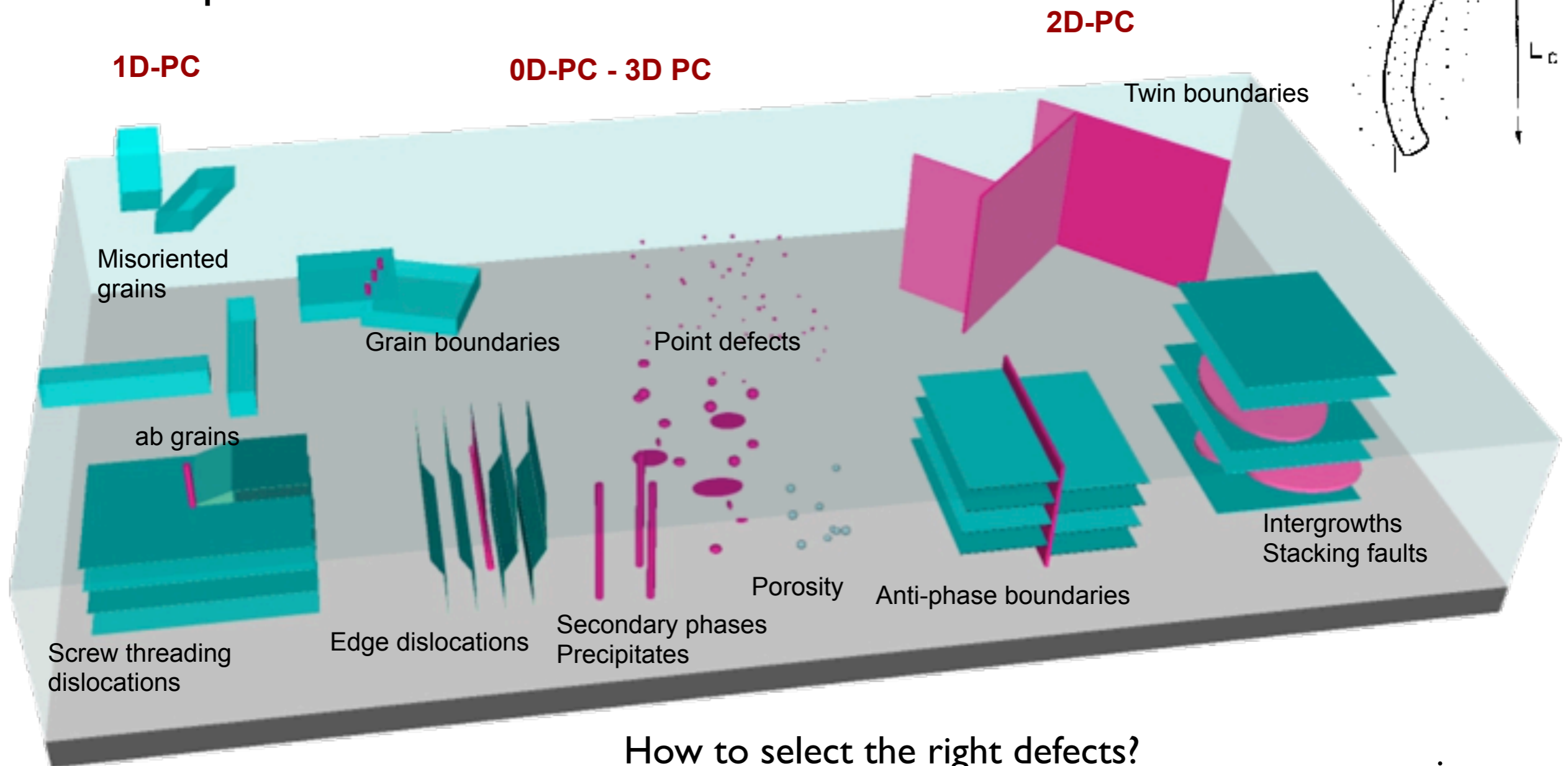
superconductor	ξ (nm)
Nb	39
Nb ₃ Sn	3
MgB ₂	35-40
YBa ₂ Cu ₃ O _{7-x}	1.5 (ab plane) 0.3 (c axis)



Here comes nanoscience!

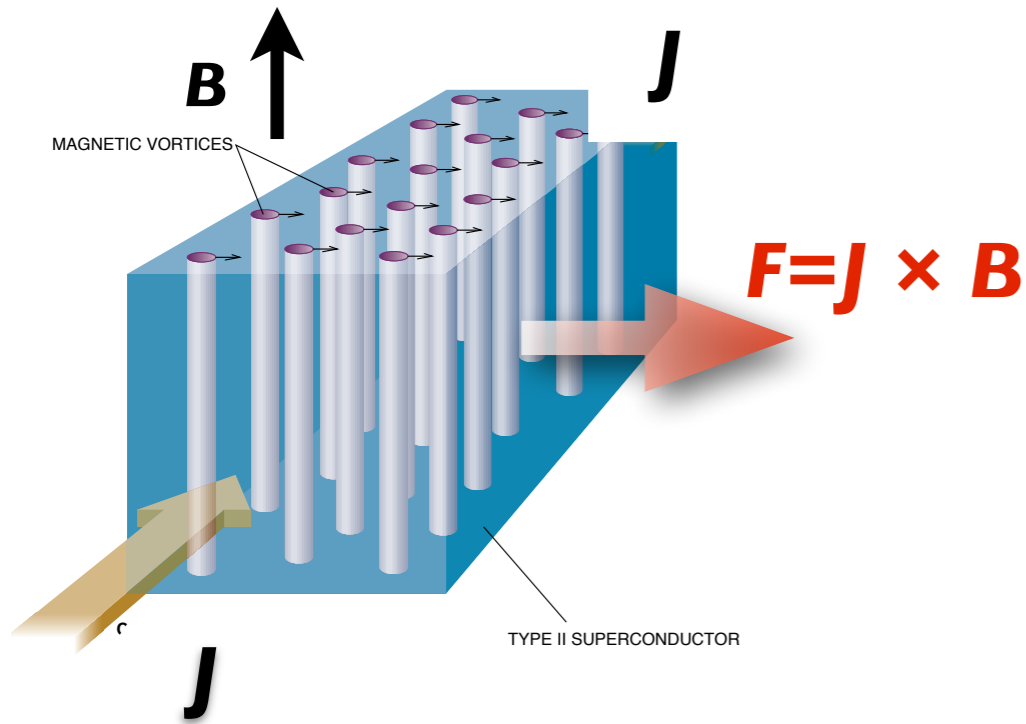
Note: add defects, reduce (maybe to 0) the resistance!
(metals: quite the opposite)

Fluxons can be flexible (energy balance between pinning and vortex elasticity)
 \Rightarrow shape of the defects is essential



How to select the right defects?
 How to insert into the material?

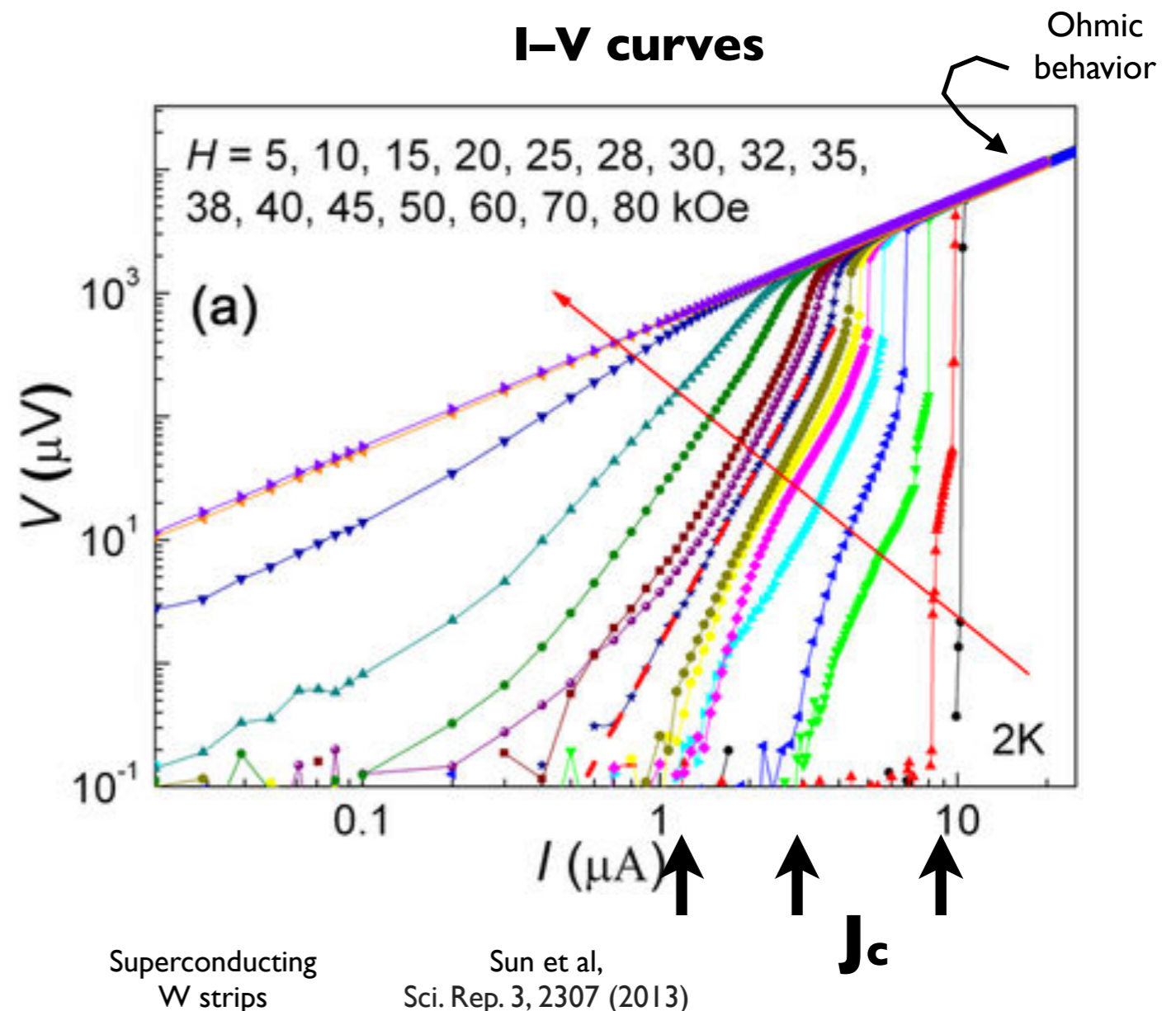
\Rightarrow nanoscience

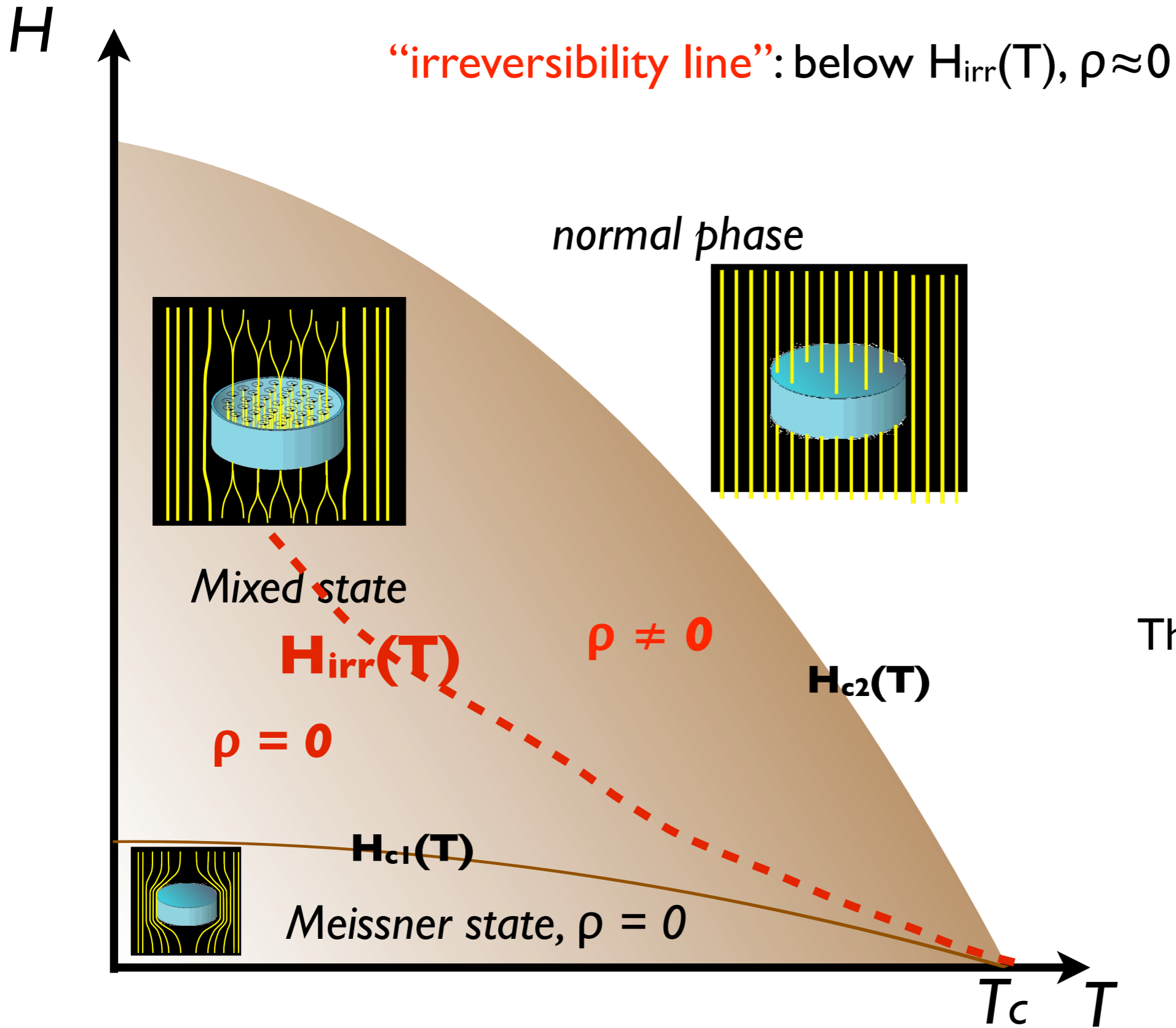


Fluxons stay pinned until the pinning force F_p exceeds $J \times B$

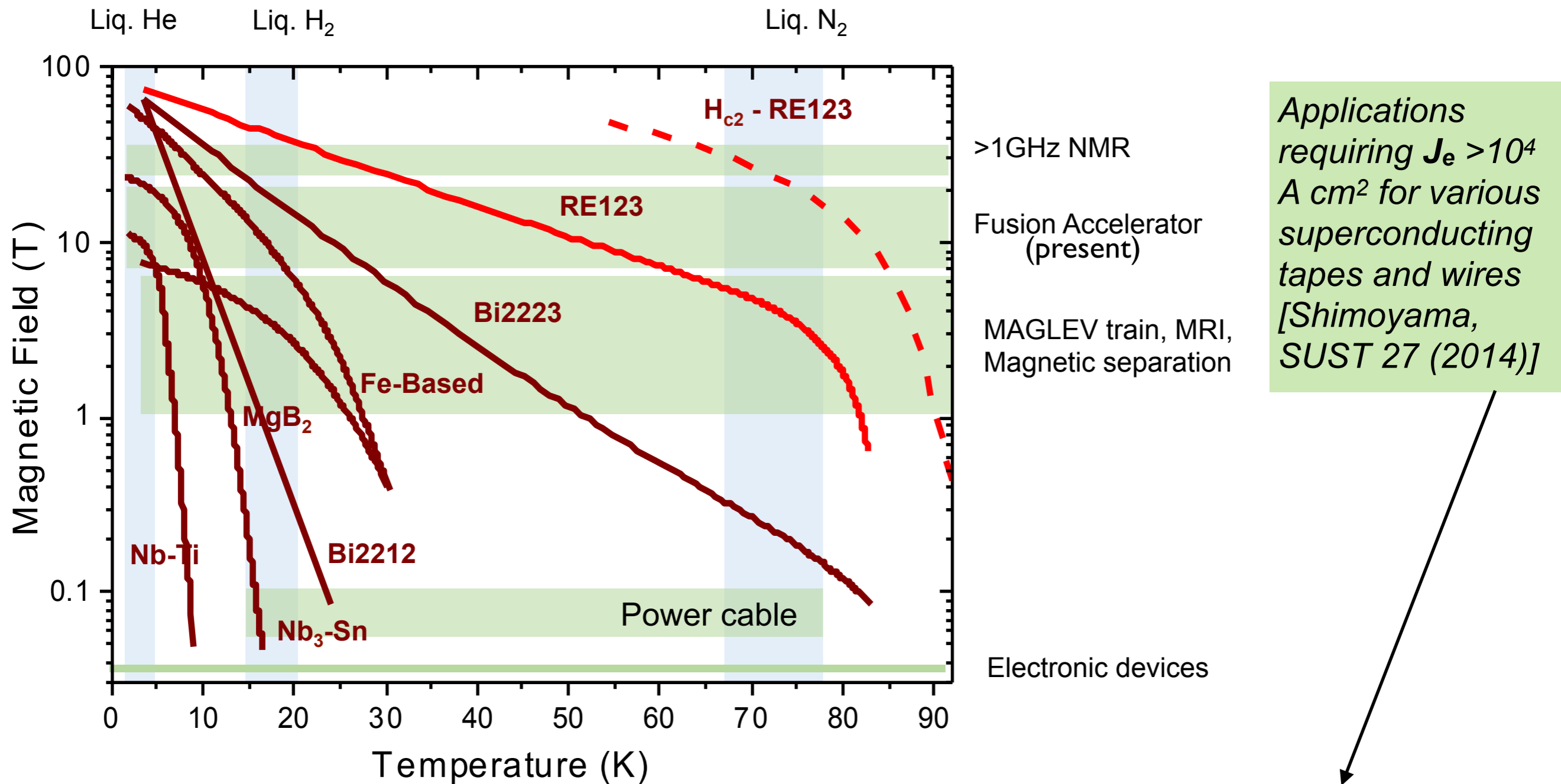
Critical current density J_c :
defined as the onset of dissipation
(ill-defined)

J_c : The quantity to maximize




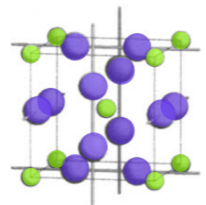
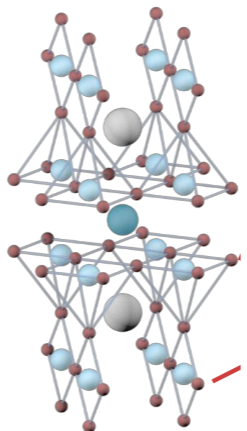


This is a “dc” picture !



1. stay below the irreversibility line
2. obtain a LARGE J_e

J_e : “engineering current density”, critical current I_c divided the section of the wire (which includes materials other than the SC alone)

superconductor	ξ (nm)	structure
Nb	39	
Nb ₃ Sn	3	
YBa ₂ Cu ₃ O _{7-x}	1.5 (ab plane) 0.3 (c axis)	

Isotropic materials
 Low T (little thermal energy)
 ⇒ no extreme defect engineering
 ⇒ Metallurgy
 (somewhat sophisticated for Nb₃Sn)

Strongly anisotropic material
 High T (large thermal energy)

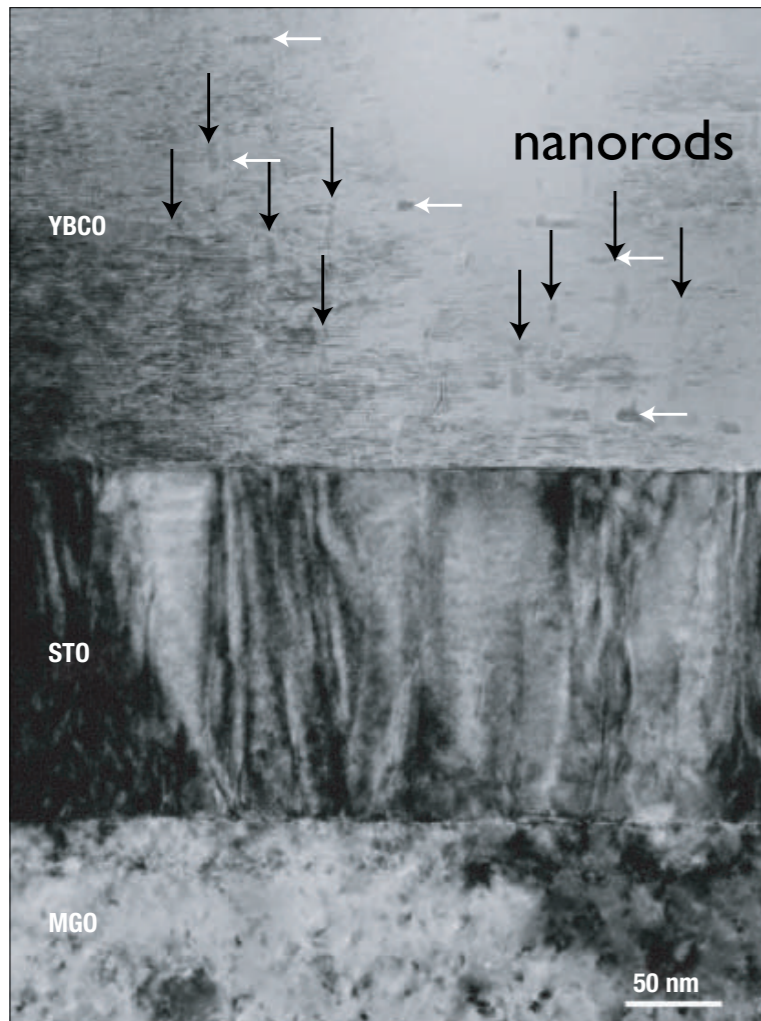
?

2004: the breakthrough

Strongly enhanced current densities in superconducting coated conductors of $\text{YBa}_2\text{Cu}_3\text{O}_{7-x} + \text{BaZrO}_3$

J. L. MACMANUS-DRISCOLL^{1*}, S. R. FOLTYN, Q. X. JIA, H. WANG, A. SERQUIS, L. CIVALE, B. MAIOROV, M. E. HAWLEY, M. P. MALEY AND D. E. PETERSON

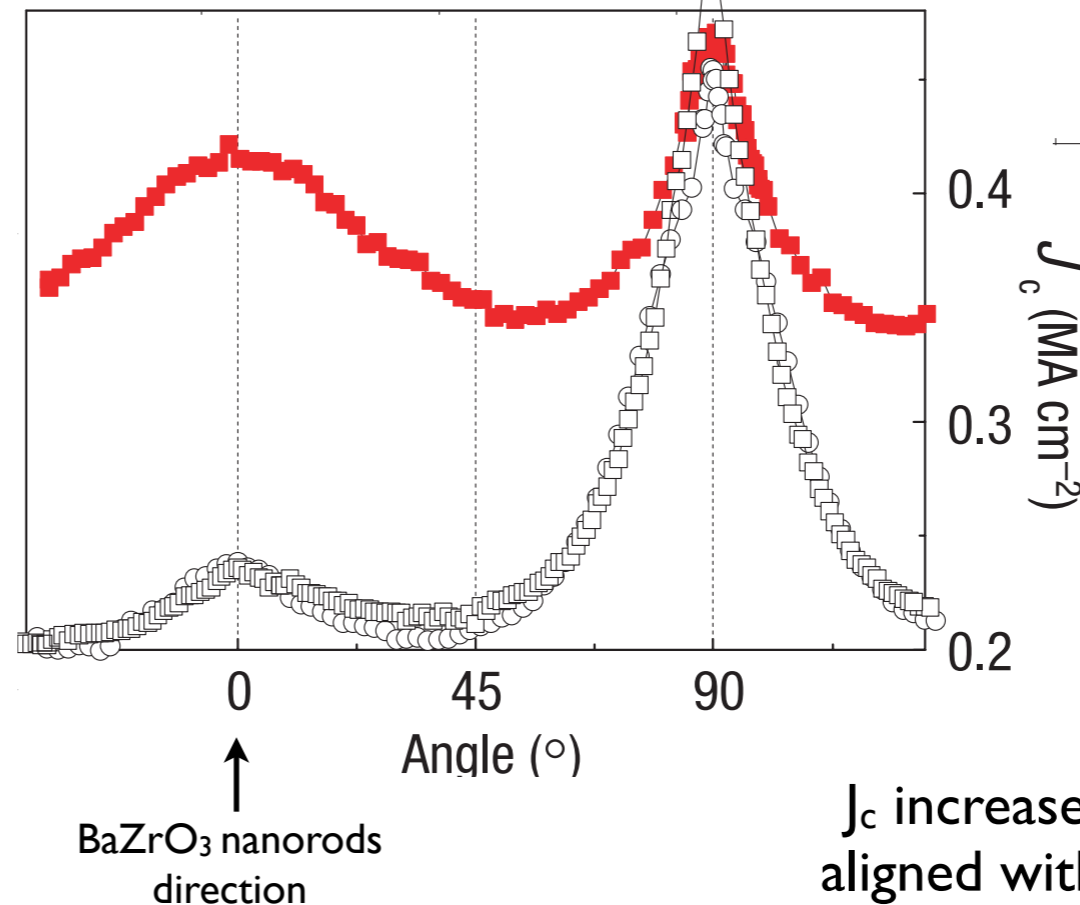
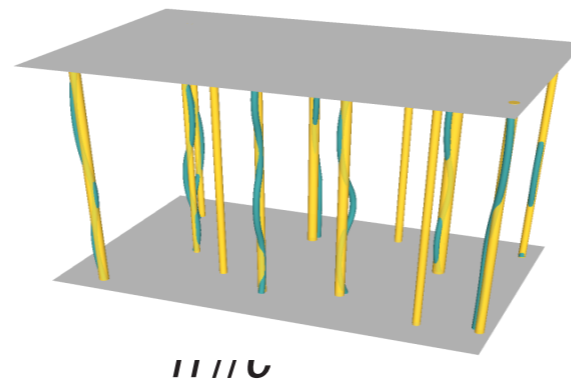
nature materials | VOL 3 | JULY 2004 |



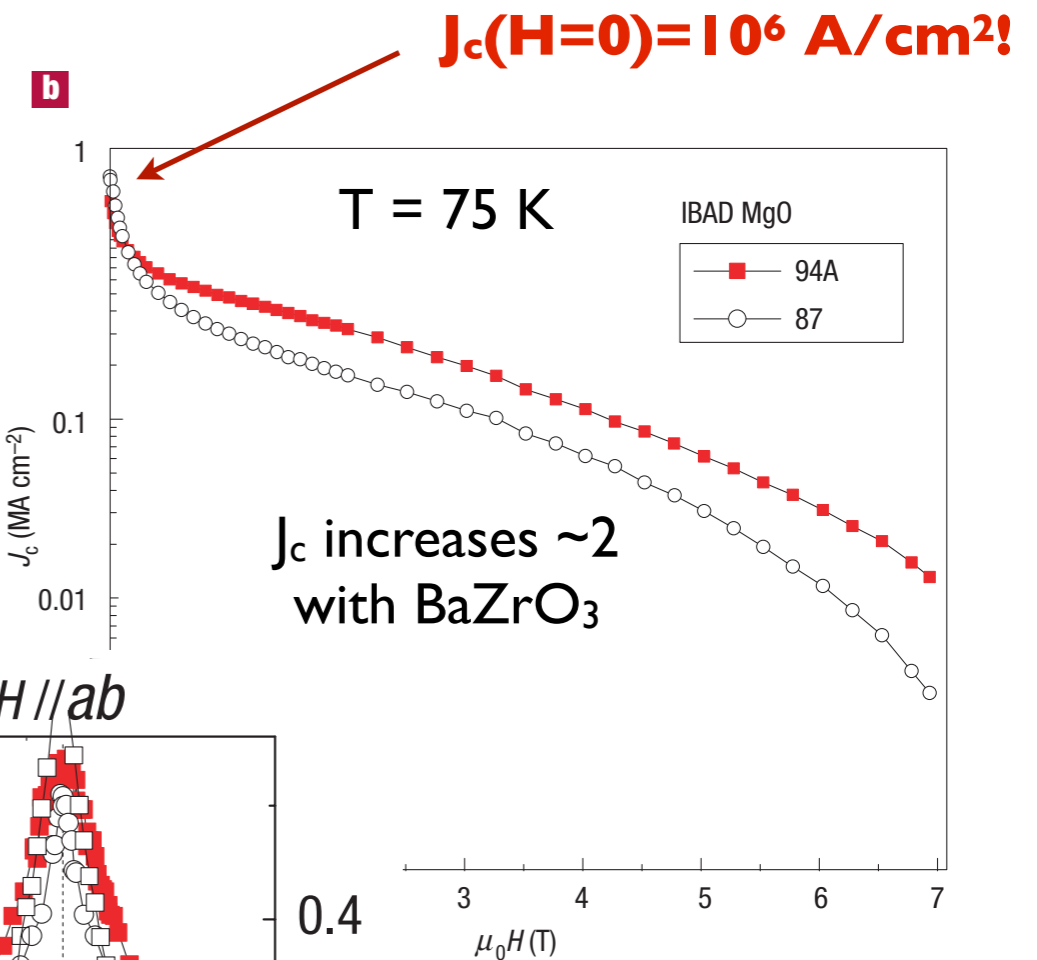
BaZrO_3 cubic second phase inserted in the YBCO matrix as *nanorods*, dia. $\sim 5\text{nm}$

(Cambridge)

Successful insertion of nanorods in YBCO (pulsed laser technique)



J_c increases when the field is aligned with BaZrO_3 nanorods



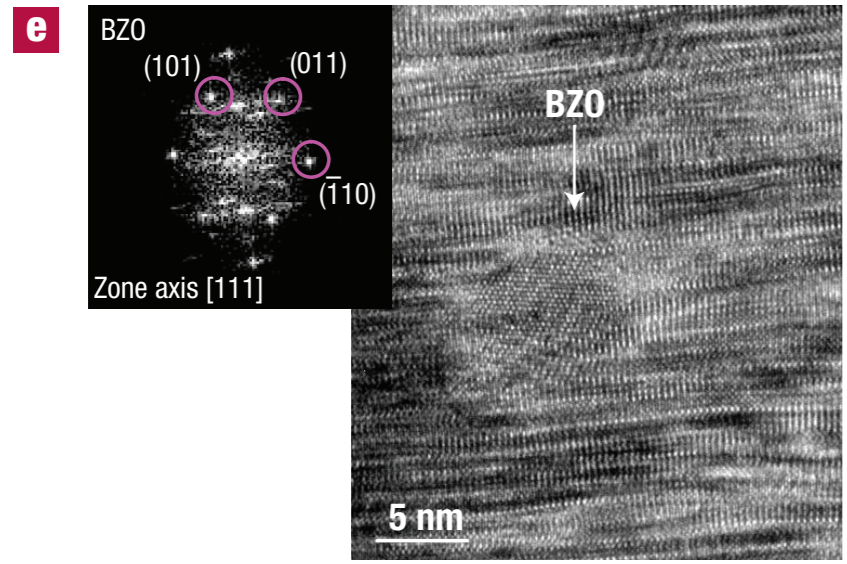
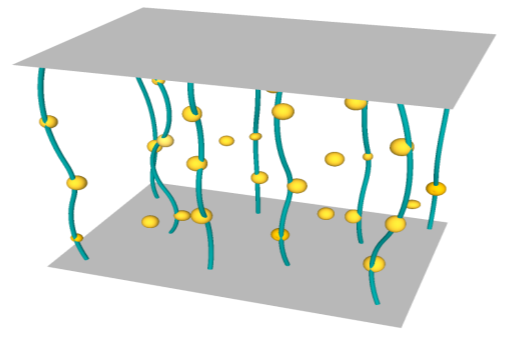
J_c increases ~ 2 with BaZrO_3

Strong isotropic flux pinning in solution-derived $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ nanocomposite superconductor films

J. GUTIÉRREZ*, A. LLODÉS*, J. GÁZQUEZ, M. GIBERT, N. ROMÀ, S. RICART, A. POMAR, F. SANDIUMENGE, N. MESTRES, T. PUIG AND X. OBRADORS*

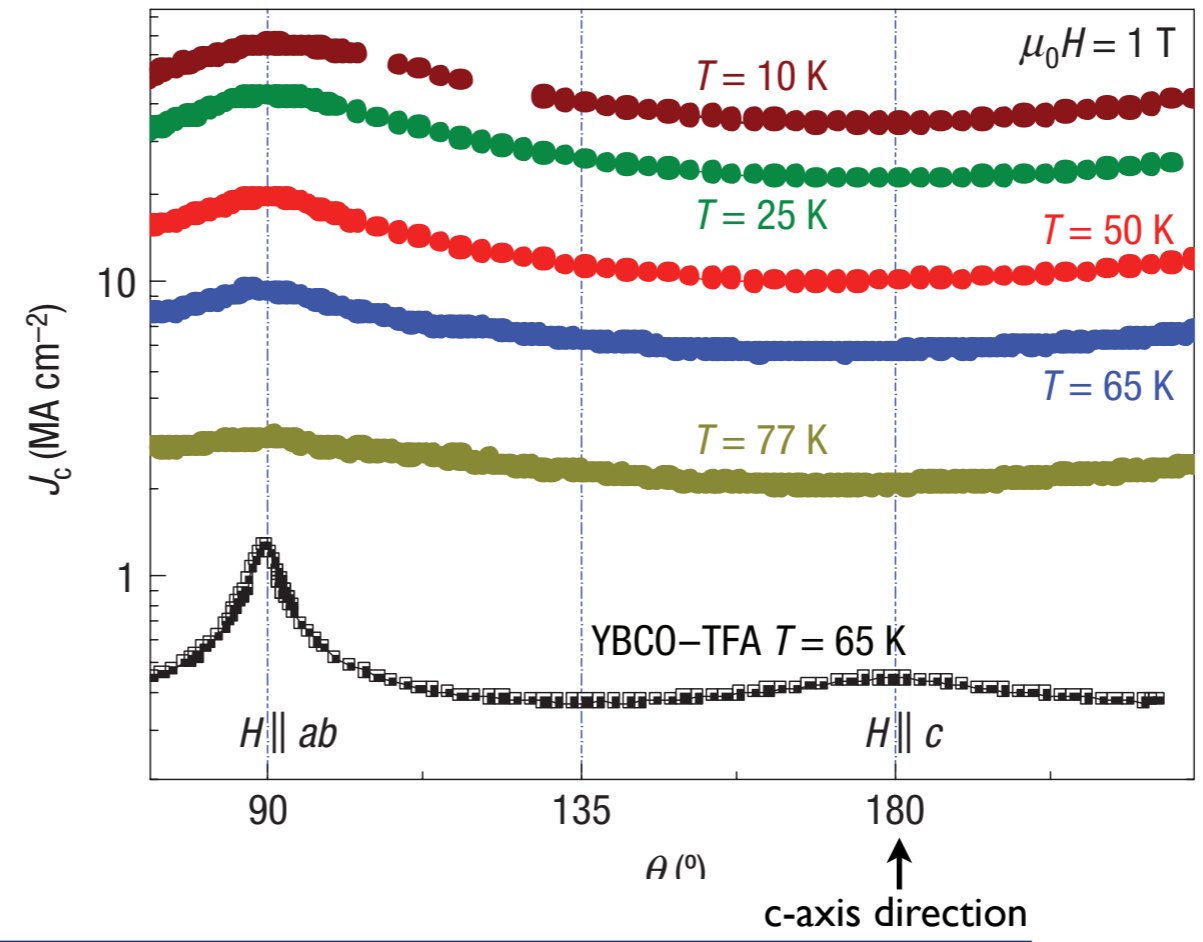
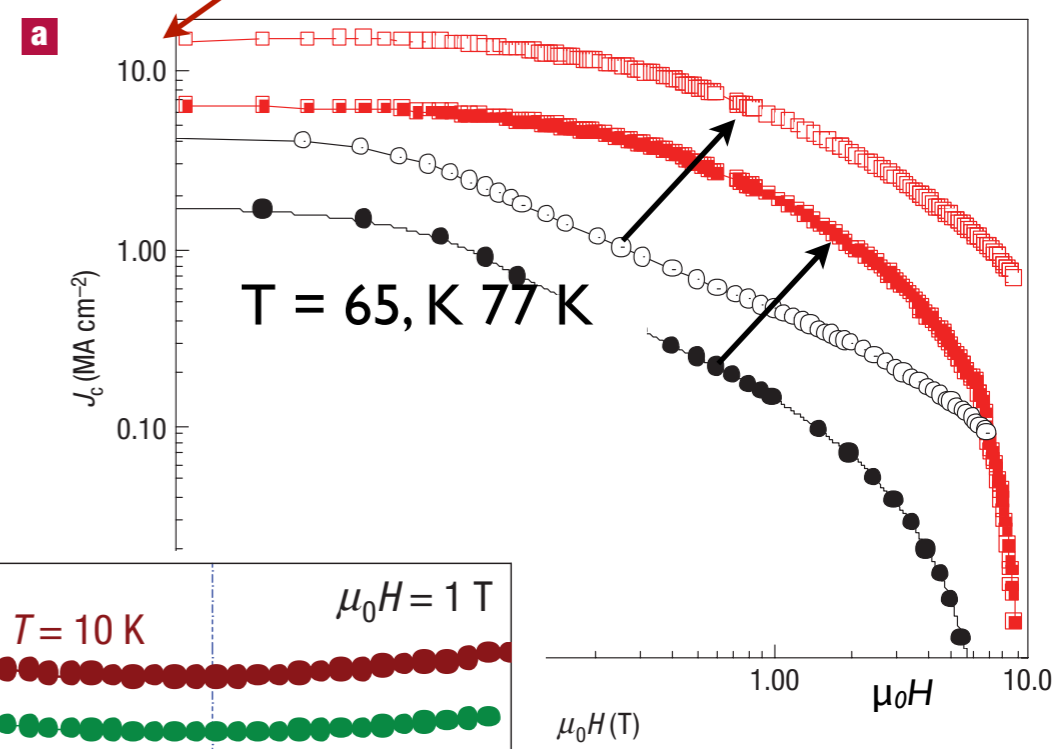
nature materials | VOL 6 | MAY 2007

(Barcelona)
Successful insertion of nanoparticles in YBCO (chemical deposition technique)



HRTEM:
 BaZrO_3 cubic second phase inserted in the YBCO matrix as nanoparticle, dia. $\sim 7\text{nm}$

$J_c(H=0) > 10^7 \text{ A/cm}^2!$



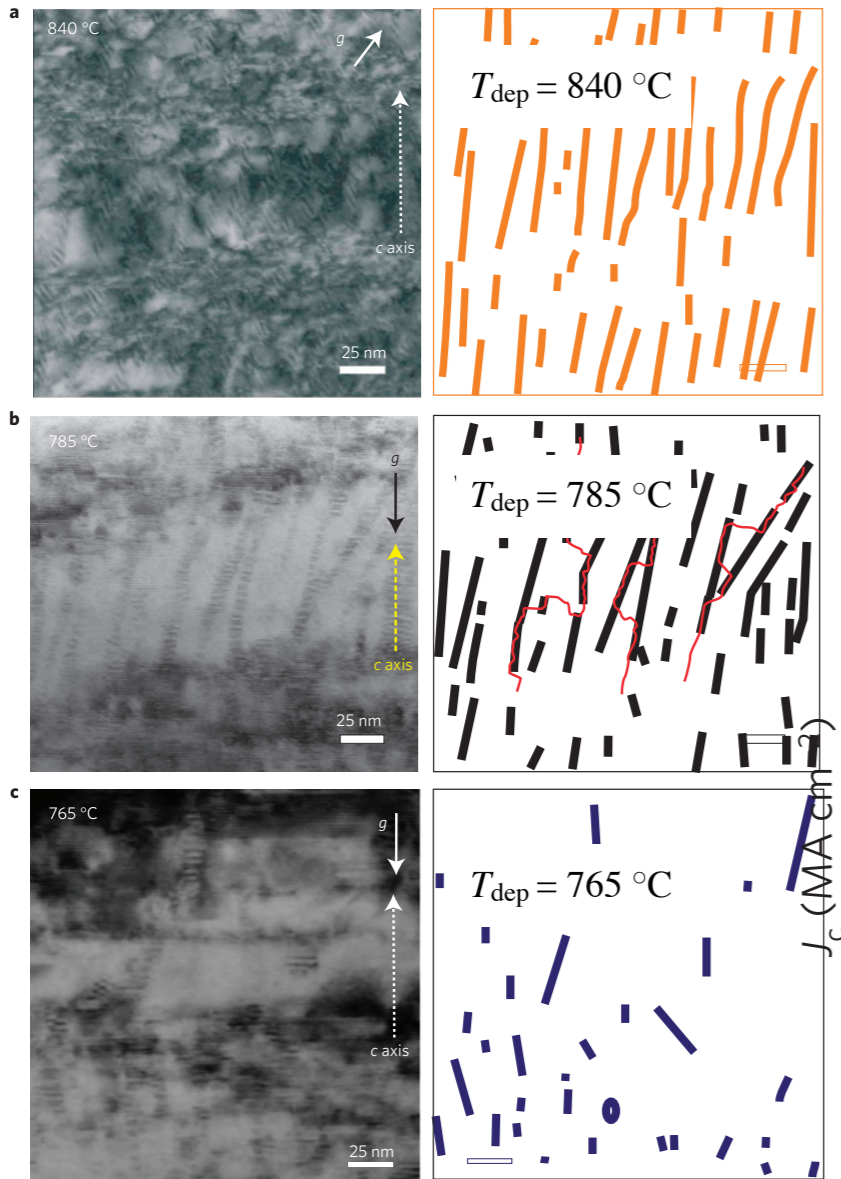
J_c increases ~ 4 with BaZrO_3

J_c with BaZrO_3 nanoparticles *flatter* than in pristine: good for applications

Synergetic combination of different types of defect to optimize pinning landscape using BaZrO₃-doped YBa₂Cu₃O₇

B. Maiorov*, S. A. Baily†, H. Zhou‡, O. Ugurlu§, J. A. Kennison, P. C. Dowden, T. G. Holesinger, S. R. Foltyn and L. Civale

NATURE MATERIALS | VOL 8 | MAY 2009



TEM:

BaZrO₃ nanorods + nanoparticle
Splay depending on deposition T

Enrico Silva

(Los Alamos)

Successful insertion of splayed nanorods in YBCO + nanoparticles (pulsed laser technique). Splay of nanorods + nanoparticles favors efficient pinning (fluxons are flexible!)

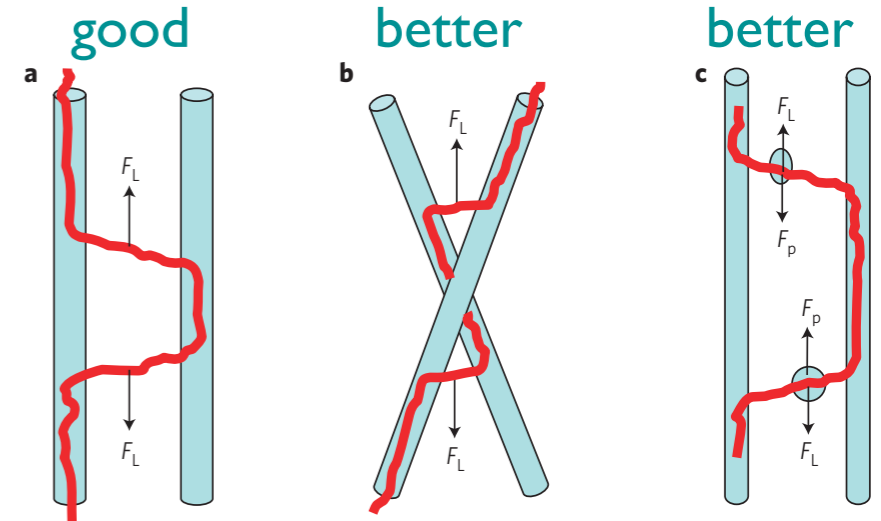
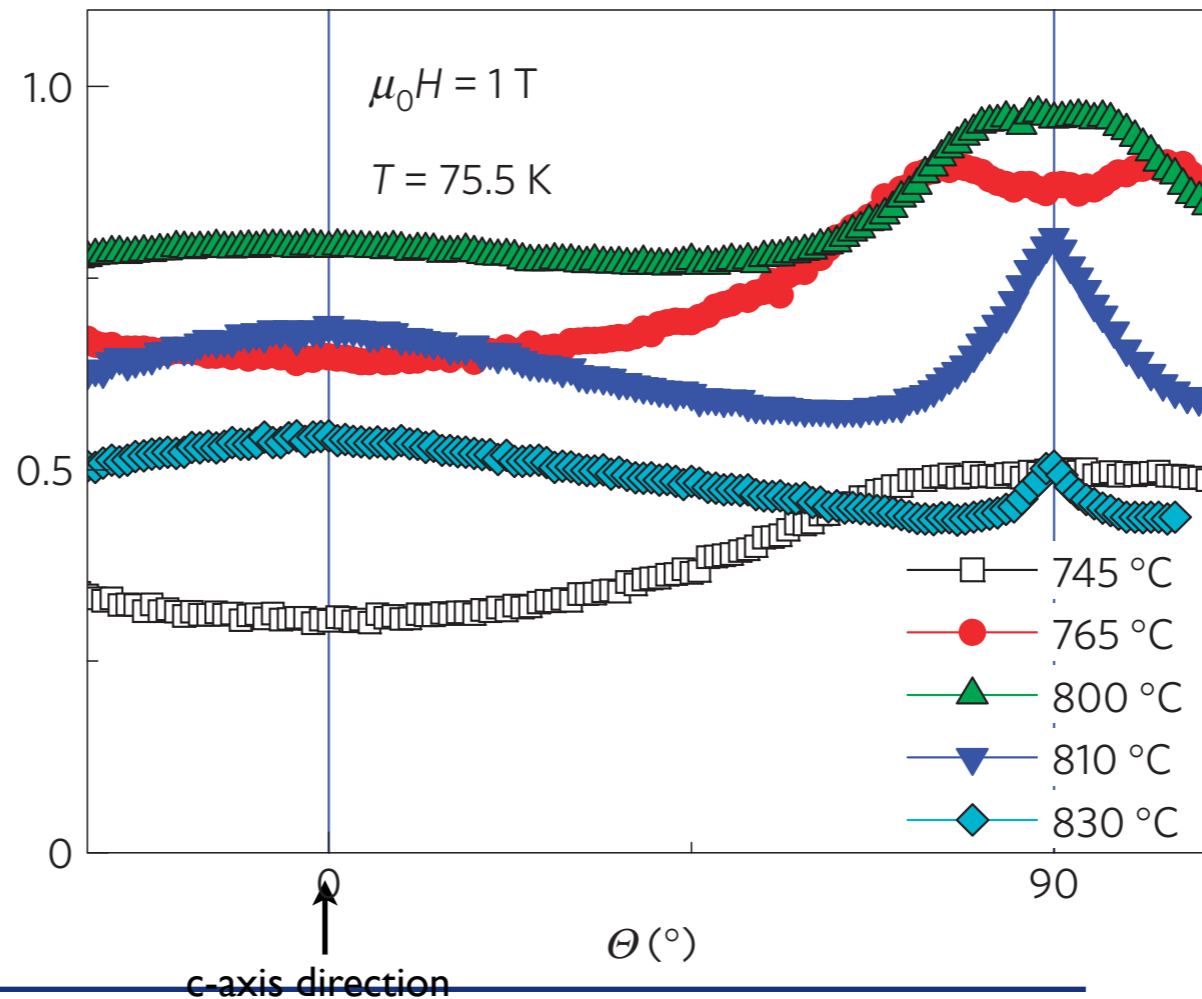


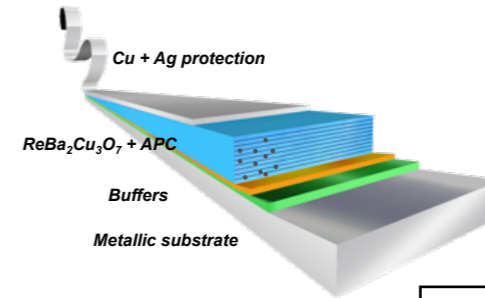
Figure 5 | Diagrams of expansion of vortex double kinks for configurations involving different defects. **a**, Diagram of two parallel



Engineering nanodefects shape and landscape

(Houston)

APPLIED PHYSICS LETTERS 106, 032601 (2015)



High critical currents in heavily doped (Gd,Y)Ba₂Cu₃O_x superconductor tapes

V. Selvamanickam,^{1,a)} M. Heydari Gharahcheshmeh,¹ A. Xu,¹ E. Galstyan,¹ L. Delgado,¹ and C. Cantoni²

¹Department of Mechanical Engineering and Texas Center for Superconductivity, University of Houston, 4800 Calhoun Rd., Houston, Texas 77204-4006, USA

²Oak Ridge National Laboratory, Oak Ridge, Tennessee 37381, USA

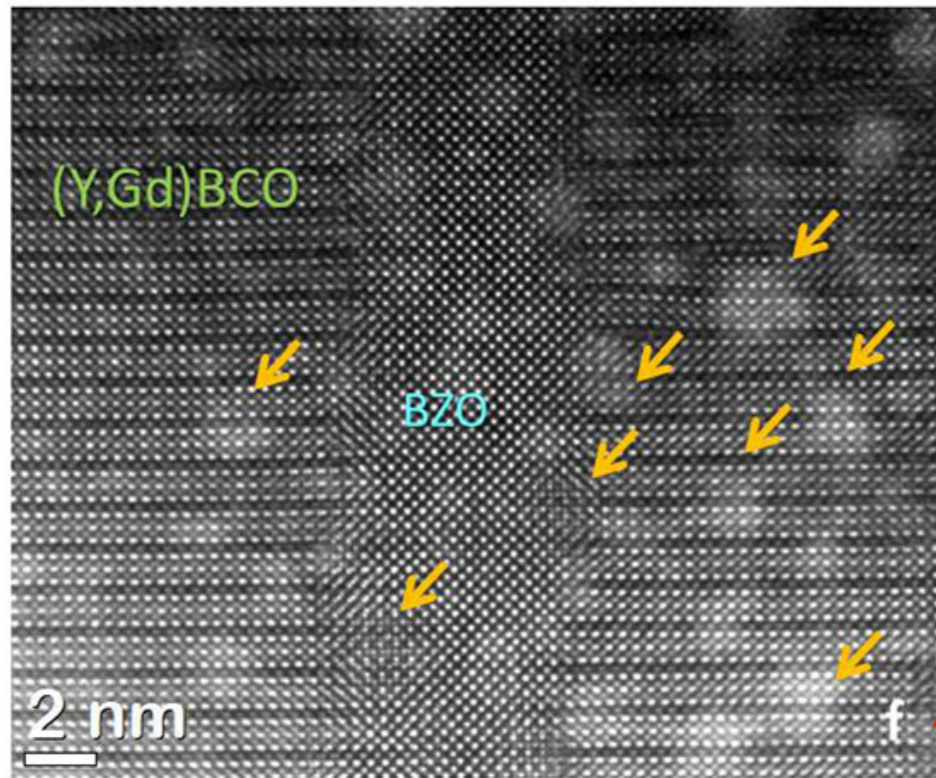
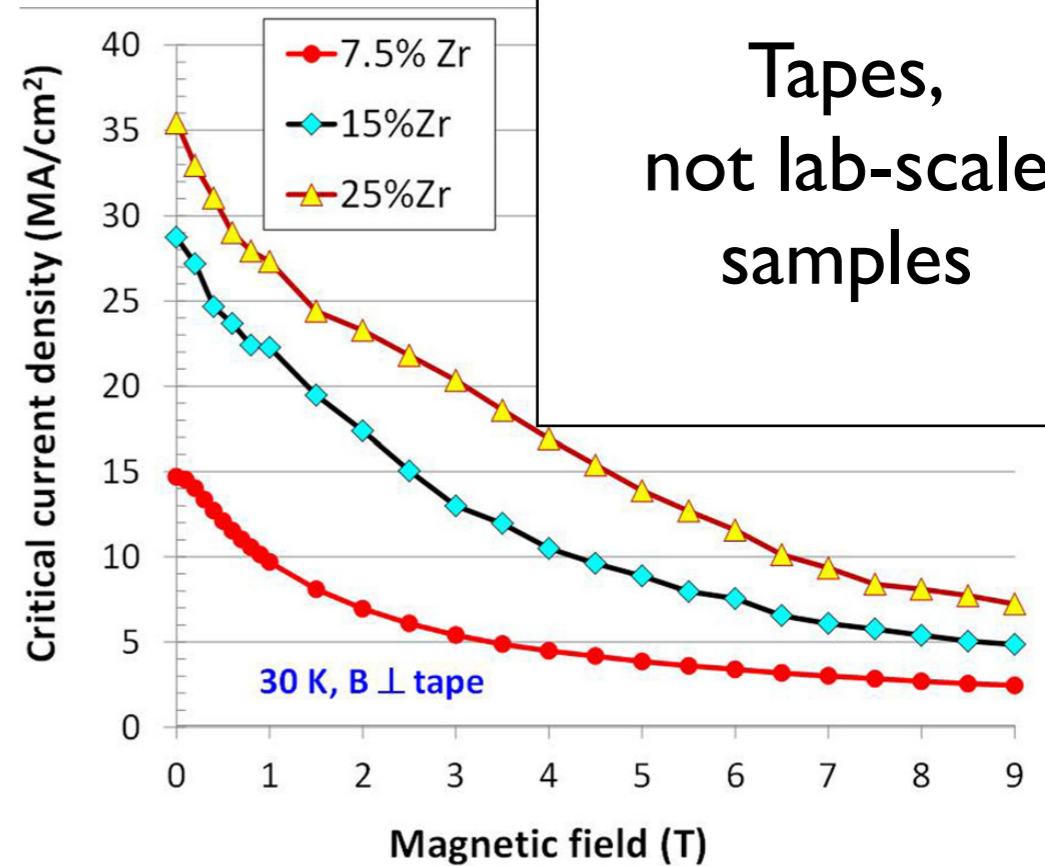


FIG. 4. Microstructures of a **25 mol. %** Zr-added (Gd,Y)BCO superconductor tape made by MOCVD analyzed by cross sectional TEM (a) and (b), plan



Tapes,
not lab-scale
samples

exceptionally robust
macroscopic quantum
state!

FIG. 1. Magnetic field dependence of critical current density of 7.5, 15, and 25 mol. % Zr-added (Gd,Y)BCO superconductor tapes at 30 K in the orientation of field perpendicular to tape. Inset—Pinning force characteristics of 25 mol. % Zr-added (Gd,Y)BCO superconductor tape at 30 K and 20 K in the orientation of field perpendicular to tape.

Microwave field

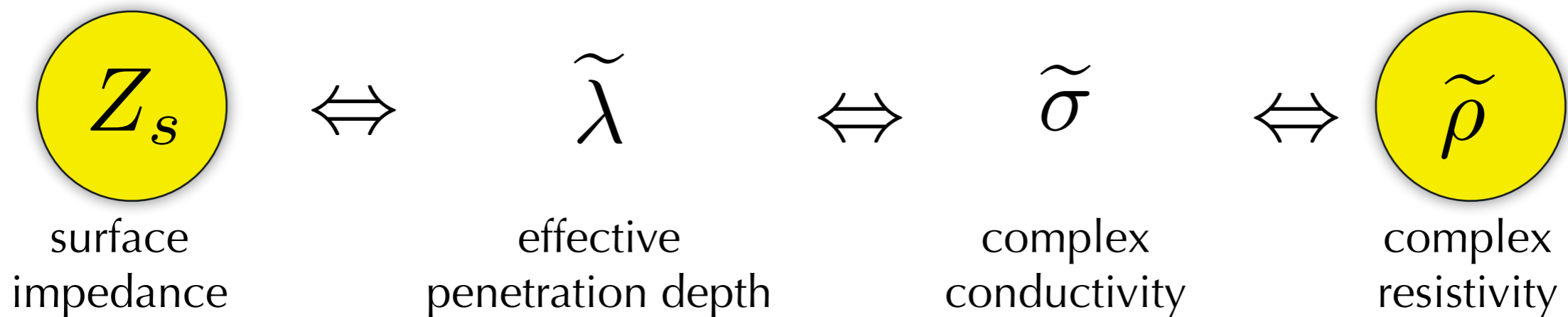


Microwave response and the mixed state



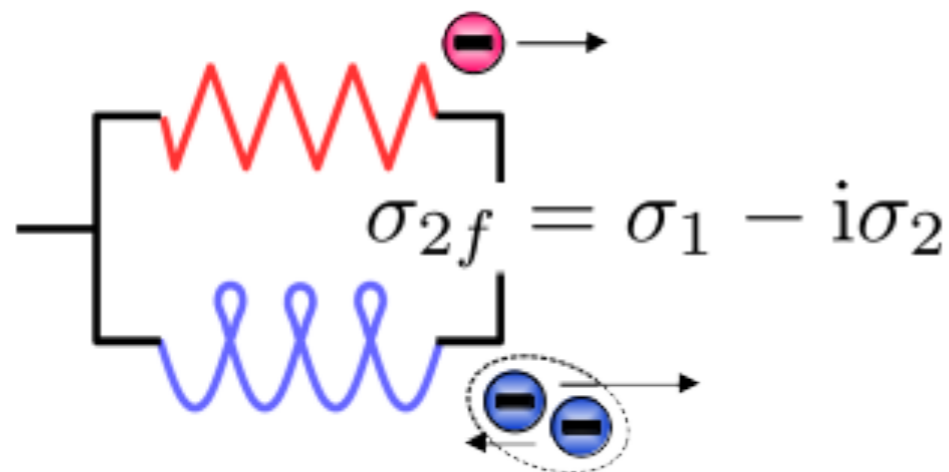
$$Z_s = i\omega\mu_0\tilde{\lambda}(H, T, \omega) = \sqrt{i\omega\mu_0\tilde{\rho}(H, T, \omega)} = \sqrt{i\frac{\omega\mu_0}{\tilde{\sigma}(H, T, \omega)}}$$

“screening + dissipation”
 “reactance + resistance”
 “out-of-phase + in-phase”

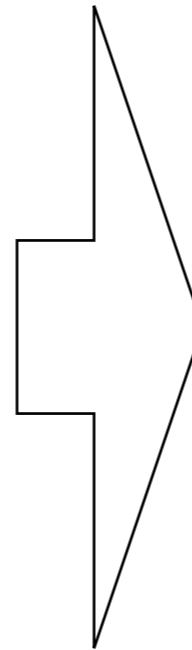


- Two fluid model (or BCS)

$$\sigma_1 = \frac{n_n e^2 \tau}{m}$$



$$\sigma_2 = \frac{n_s e^2}{m\omega} = \frac{\omega\mu_0}{\lambda^2}$$



$$\tilde{\rho} = \frac{1}{\sigma}$$

$$Z_S = \sqrt{\frac{i\omega\mu_0}{\sigma}} + R_{res}$$

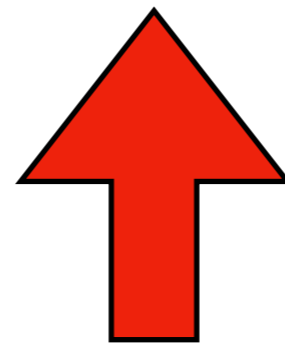
“The cleaner the material,
the better”

(low R_{res})

but...



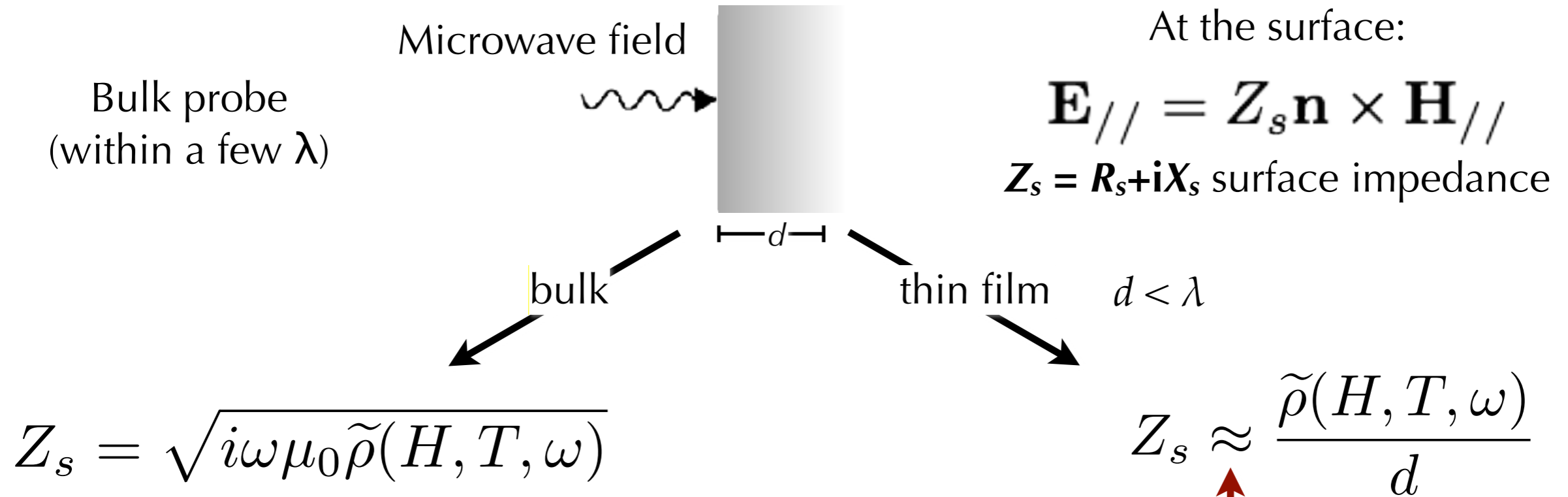
$$Z_s = i\omega\mu_0\tilde{\lambda}(H, T, \omega) = \sqrt{i\omega\mu_0\tilde{\rho}(H, T, \omega)}$$



vortex motion in the complex resistivity !

Superfluid/quasiparticle contribution to R_s is irrelevant with respect to vortex motion!

$$\tilde{\rho} = \frac{\rho_{vm} + i\frac{1}{\sigma_2}}{1 + i\frac{\sigma_1}{\sigma_2}}$$



Thin film:

complexities due to different substrates.

Care in the extraction of superconducting parameters:

Metallic (coated conductors)

N. Pompeo et al., IEEE TAS 28, 9000505 (2018)

J. Phys.: Conf. Ser. 1065 052018 (2018)

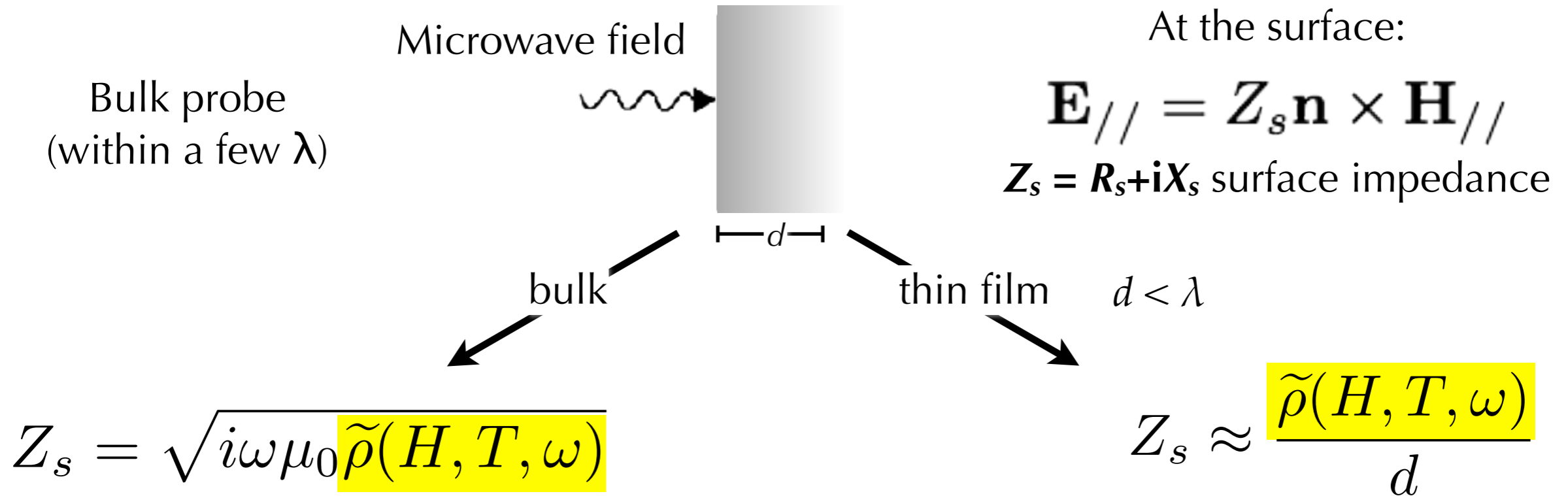
Para/ferroelectric (SrTiO₃)

N. Pompeo et al., SuST 20, 1002 (2007)

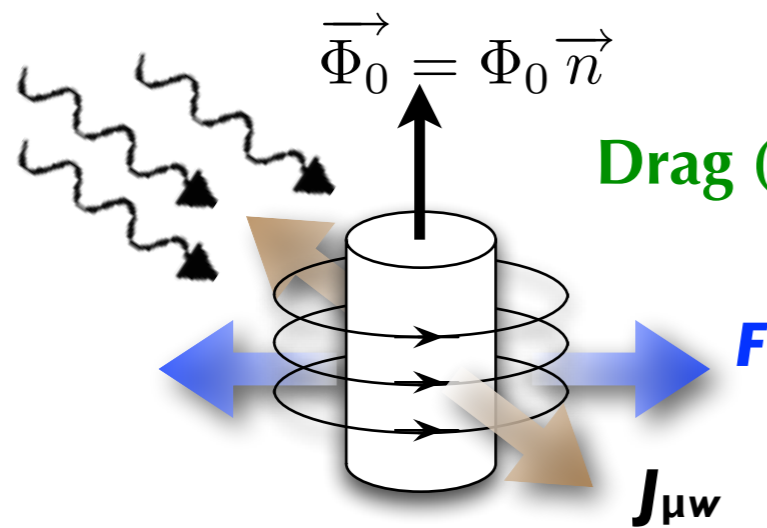
N. Pompeo et al. Proc. IEEE I2MTC 2017 doi: 10.1109/I2MTC.2017.7969902

Semiconductors

N. Pompeo, R. Marcon, L. Méchin, E. Silva, SuST 18, 531 (2005)



Driven vortices \rightarrow complex resistivity



+
elastic recall (pinning)

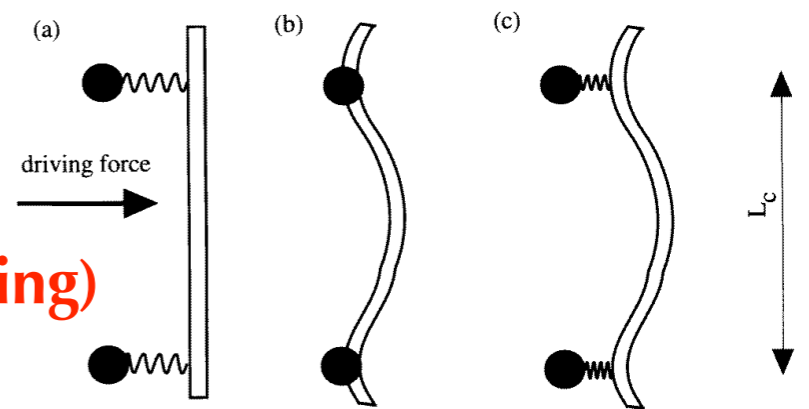
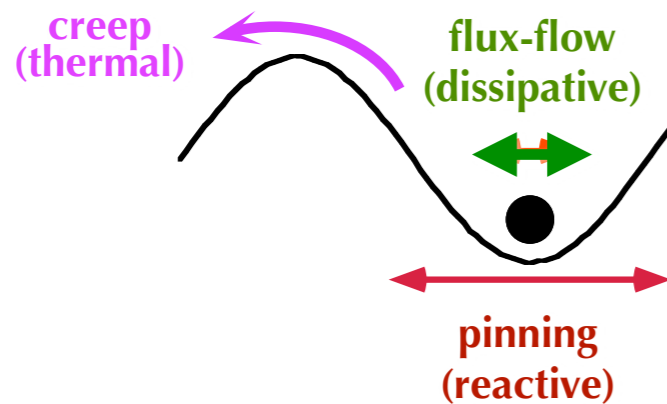


Fig. by M. Golosovsky et al, SuST. 9 (1996) 1–15.

Microwaves:

vortex shaking

- **subcritical current: linear regime**
 - negligible perturbation on pinning potential
 - very small rms vortex displacements ($\ll 1$ nm)
- \Rightarrow vortex-vortex interaction relevant for the **static** arrangement only

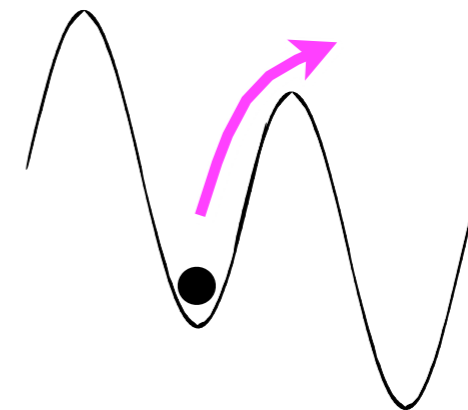


- dissipation+reactance (two observables)

DC (J_c, ρ_{dc}):

vortex drag

- large current: **strongly nonlinear regime**
 - strongly tilted washboard potential
 - large vortex displacement
- \Rightarrow vortex-vortex interaction relevant in the **dynamics** (vortex fluids, molasses, etc etc)

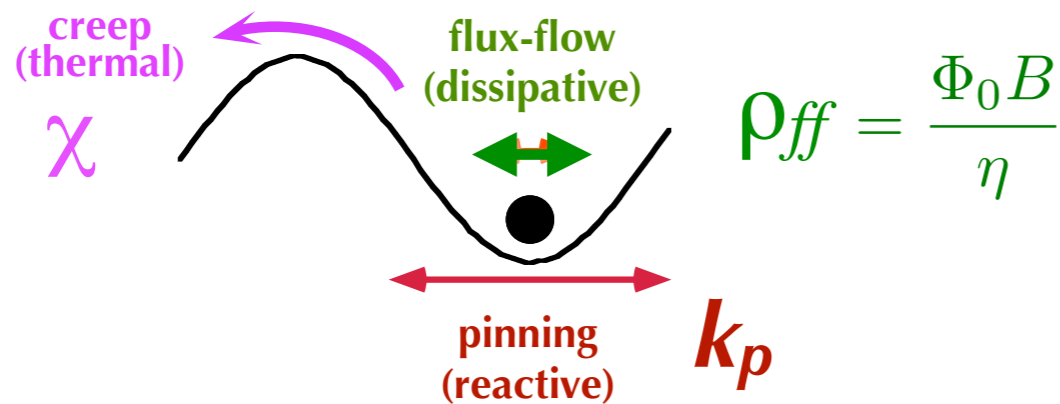


- technical issues (heating, large currents, ...)

Microwaves:

vortex shaking

- **subcritical current: linear regime**
 - negligible perturbation on pinning potential
 - very small rms vortex displacements ($\ll 1$ nm)
- \Rightarrow vortex-vortex interaction relevant for the **static** arrangement only



$$\eta \mathbf{v} + k_p \mathbf{x} = \mathbf{J} \times \hat{\mathbf{n}} \Phi_0 + \mathbf{F}_{\text{thermal}}$$

Gittleman, Rosenblum PRL 16 734 (1966)

No creep

* Coffey, Clem PRL 67 386 (1991)

Sinusoidal potential

Brandt PRL 67 2219 (1991)

Thermally relaxing k_p

Placais, Mathieu, Simon, Sonin, Traito, PRB 54 13083 (1996)

Two-mode

** N. Pompeo, E.Silva PRB 78, 094503 (2008)

vortex motion complex resistivity

Many models*, one equation**:

$$\rho_{v1}(H) + i\rho_{v2}(H) = \rho_{ff} \frac{\chi + i\frac{\nu}{\nu_0}}{1 + i\frac{\nu}{\nu_0}}$$

1. **depinning frequency**

$$\nu_0 \rightarrow \nu_p = k_p / 2\pi\eta$$

($\nu_p \neq \nu_0$: "depinning frequency")

pinning constant

(elastic recall)

$$k_p = \nu_p \eta$$

2. **flux-flow resistivity**

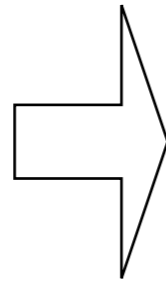
$$\rho_{ff} = \Phi_0 \mathbf{B} / \eta$$

3. **creep factor** χ

vs. T, H, θ

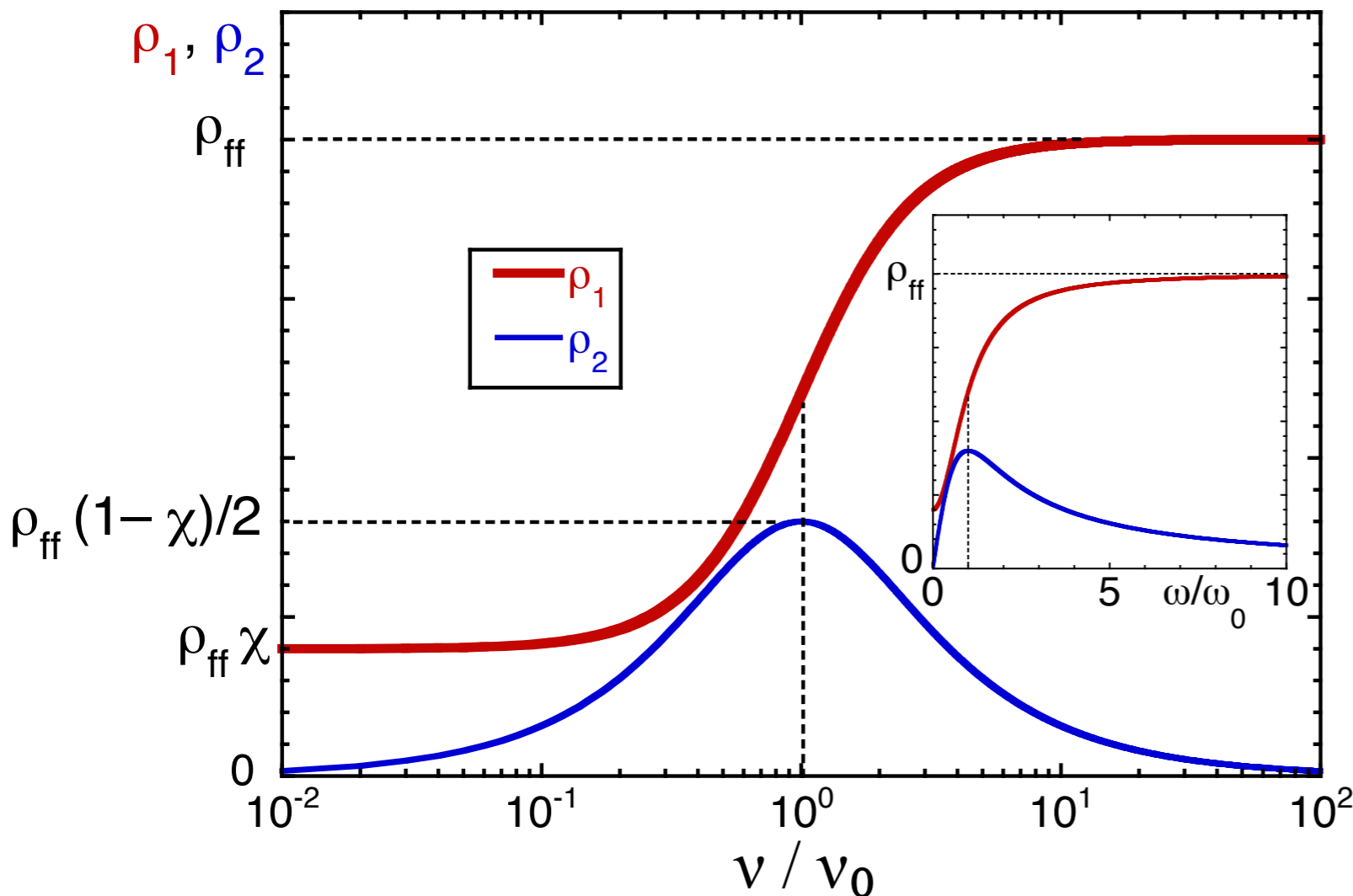
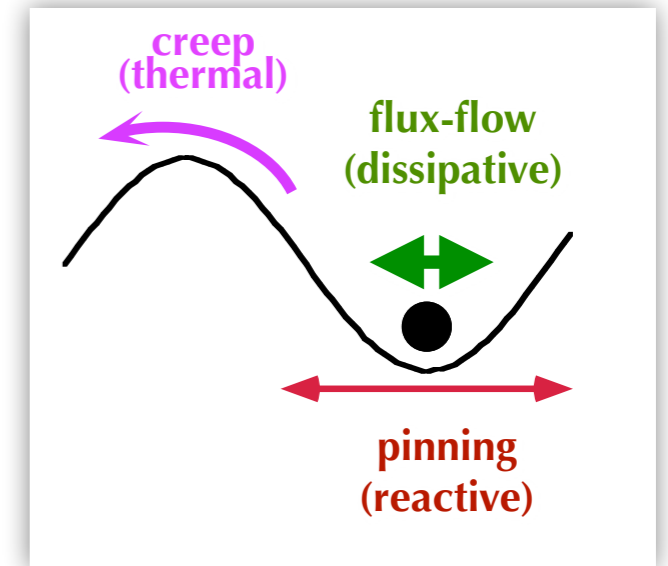
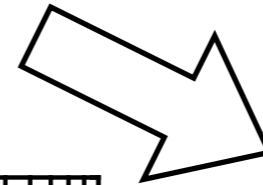
High frequency vortex motion

Even strongly pinned vortices oscillates



- dissipation arises
- elastic recall from pinning centers
- flux creep

$$\Delta\rho_1(H) + i\Delta\rho_2(H) = \rho_{ff} \frac{\chi + i\frac{\nu}{\nu_0}}{1 + i\frac{\nu}{\nu_0}}$$



1. **depinning frequency**

$$\nu_0 \rightarrow \nu_p = k_p / 2\pi\eta$$

($\nu_p \neq \nu_0$: "depinning frequency")

pinning constant

(elastic recall)

$$k_p = \nu_p \eta$$

2. **flux-flow resistivity**

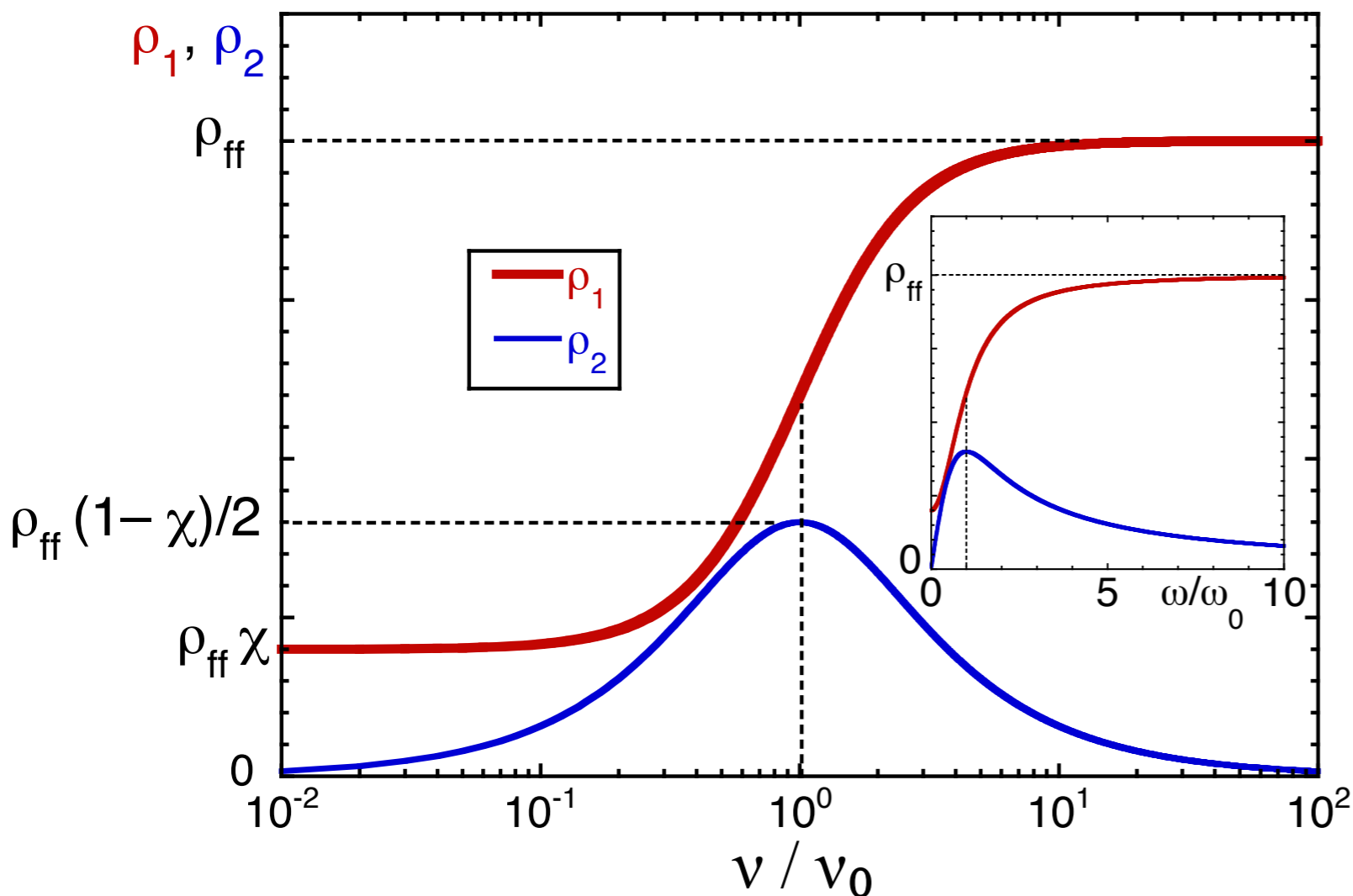
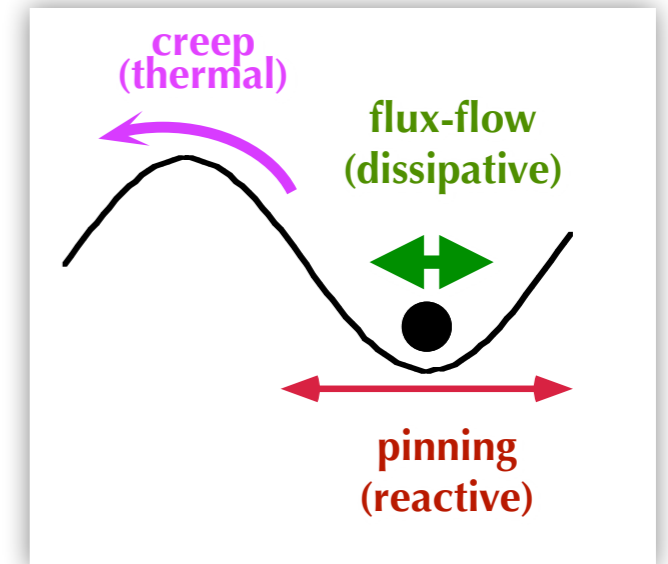
$$\rho_{ff} = \Phi_0 \mathbf{B} / \eta$$

3. **creep factor** χ

vs. T, H, θ

Decrease losses at fixed v : high v_0 , small losses:

- high k_p : strong pinning. Can be engineered (to a point)
- small ρ_{ff} . Cannot be engineered, microscopic state



1. **depinning frequency**

$$v_0 \rightarrow v_p = k_p / 2\pi\eta$$

($v_p \neq v_0$: "depinning frequency")

pinning constant

(elastic recall)

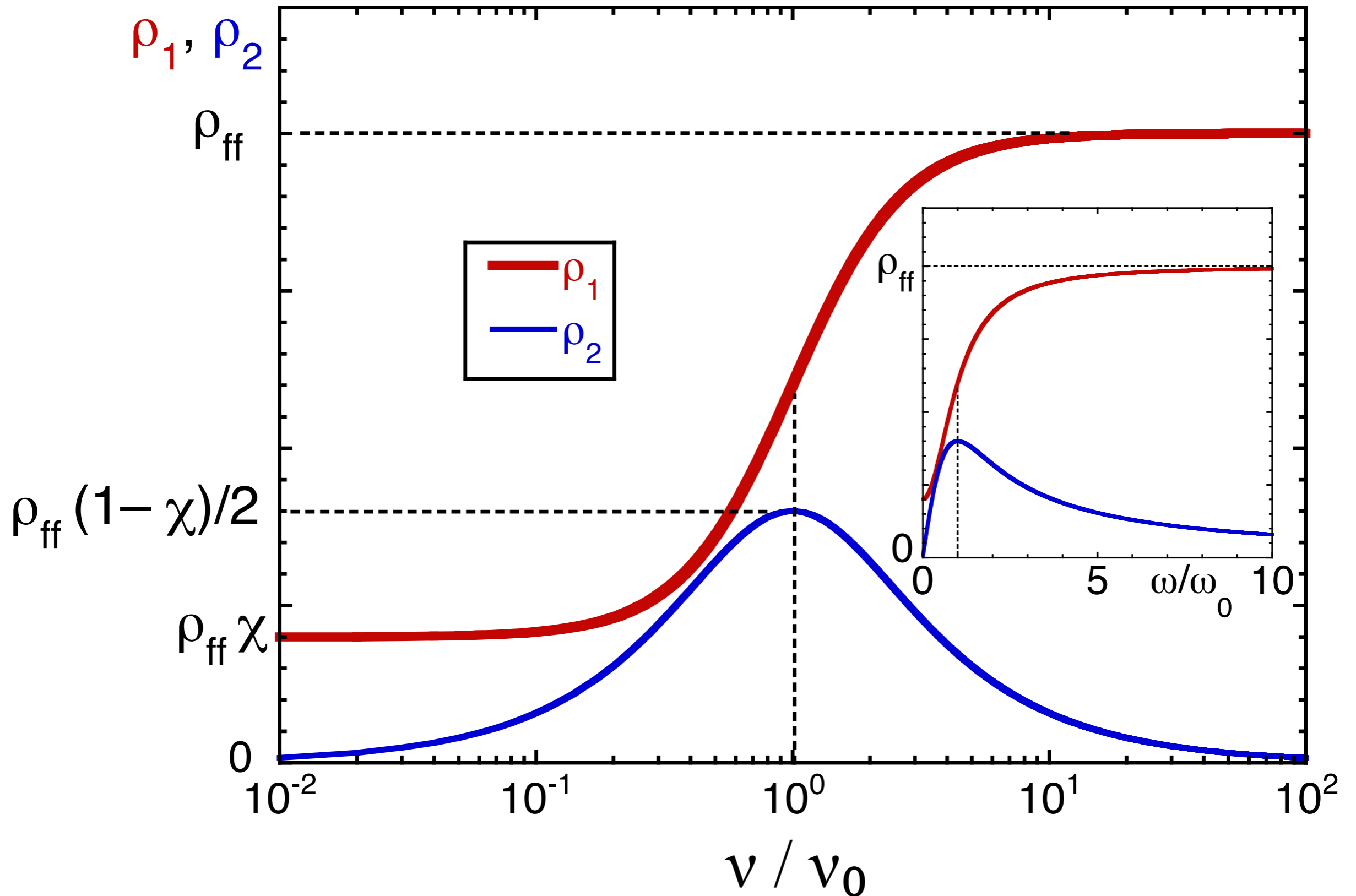
$$k_p = v_p \eta$$

2. **flux-flow resistivity**

$$\rho_{ff} = \Phi_0 \mathbf{B} / \eta$$

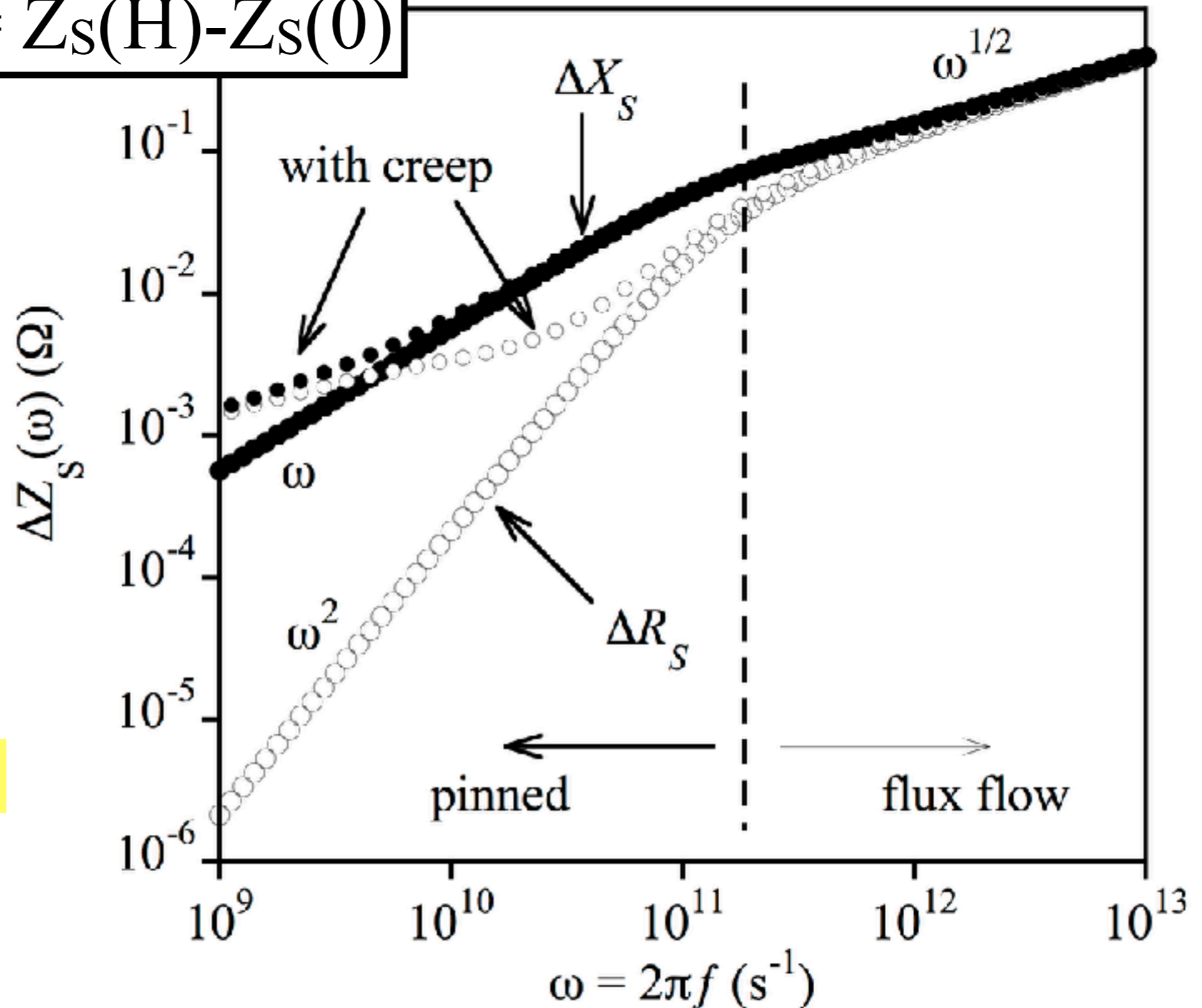
3. **creep factor** χ

vs. T, H, θ



$$\Delta Z_S = Z_S(H) - Z_S(0)$$

$k_p = 4 \times 10^4 \text{ Nm}^{-2}$
 $\eta = 3 \times 10^{-7} \text{ Nsm}^{-2}$
 (appropriate for YBa₂Cu₃O₇
 thin films around 60 K)
 $\rightarrow \omega_p = 2\pi \cdot 21 \text{ GHz}$.
 Small symbols:
 same parameters, with $\chi = 0.1$



BCS R_s irrelevant!

Why?

Flux-flow: **free** motion of vortices (pinning irrelevant)

ρ_{ff} \rightarrow Physics of the vortex core: fundamental processes

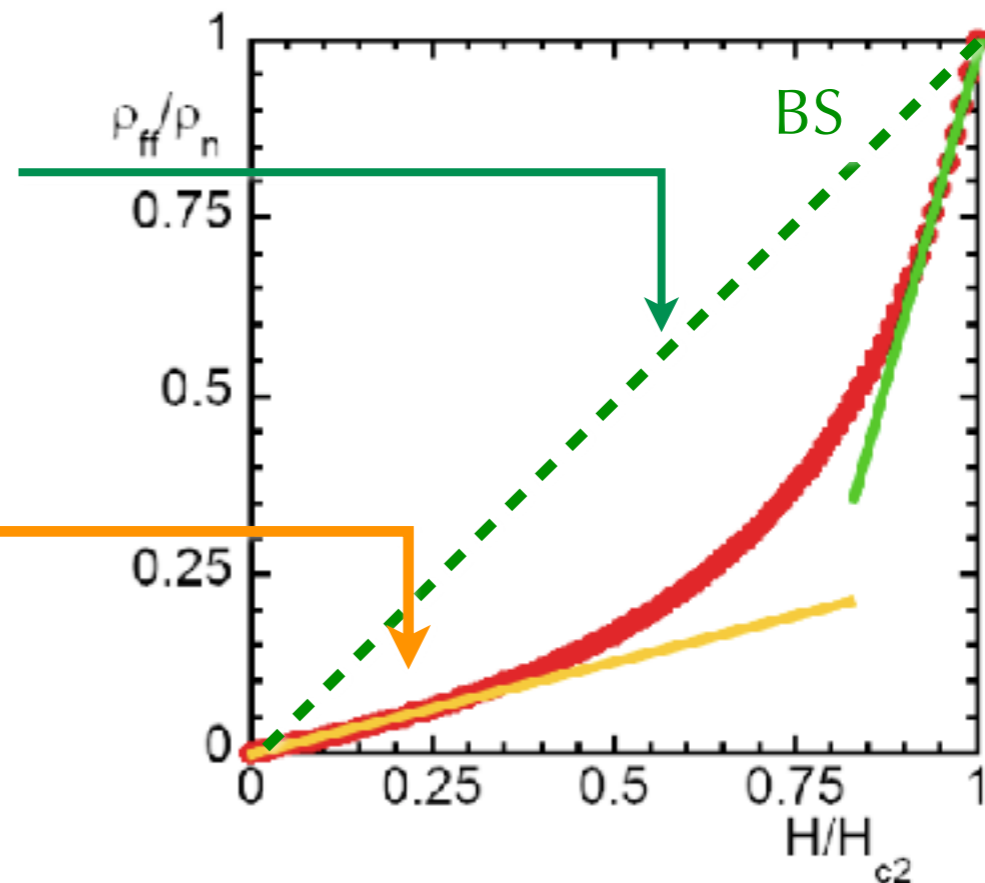
A window on quasiparticles in the mixed state

“Bardeen-Stephen

(+ maybe corrections)”

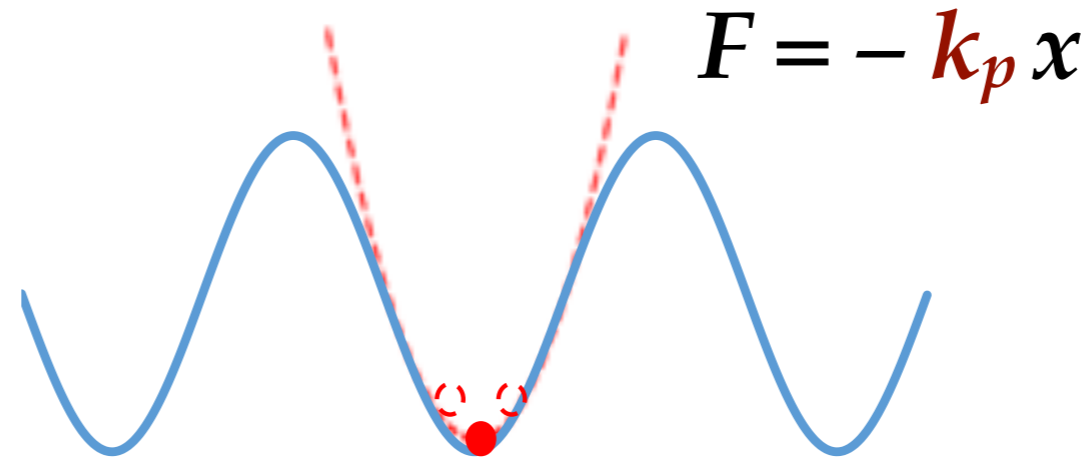
$$\rho_{ff} = \rho_n \frac{B}{B_{c2}(T)}$$

$$\rho_{ff} = \rho_n \frac{1}{\beta(T)} \frac{B}{B_{c2}(T)}$$



κ_p

Material science: pinning constant measures the *steepness* of the pinning well

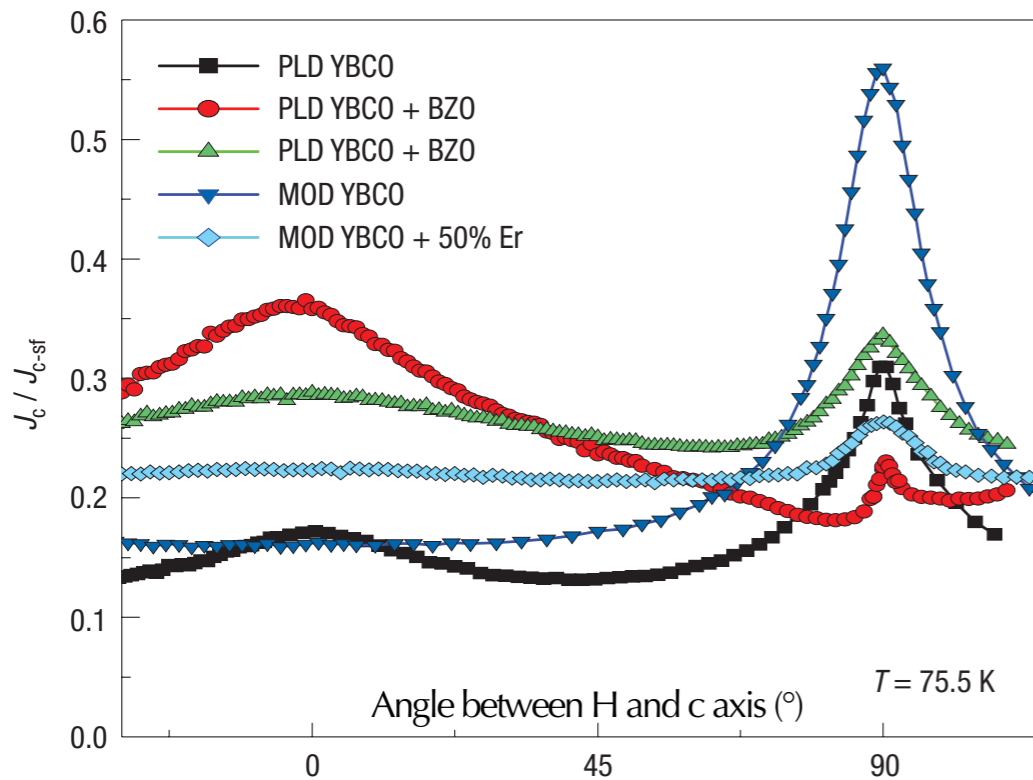


⇒ optimization of the pinning landscape

(complementary information to $J_c \rightarrow$ height of the pinning well)

ν_p

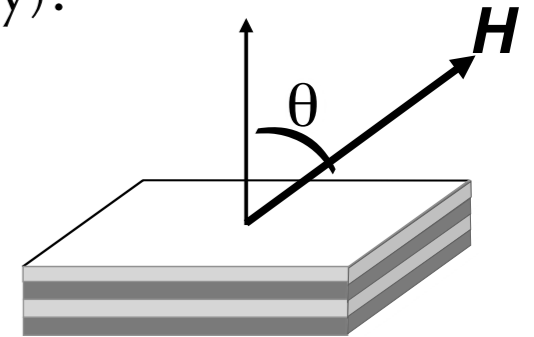
Depinning frequency: crossover between low and high losses regimes (→ applications in $H \neq 0$)



Directional pinning:
dramatic effect on J_c anisotropy.

Do defects change the microscopic state
(i.e. the mass anisotropy)?

S. R. Foltyn, I. Civale,
J. I. MacManus-Driscoll, Q. X. Jia,
B. Maiorov, H. Wang and M. Maley
Nature Materials vol.6, 631 (2007)



Measured anisotropy:

mass anisotropy + extended defects

"intrinsic"

$$\gamma = \frac{H_{c2||}}{H_{c2\perp}}$$

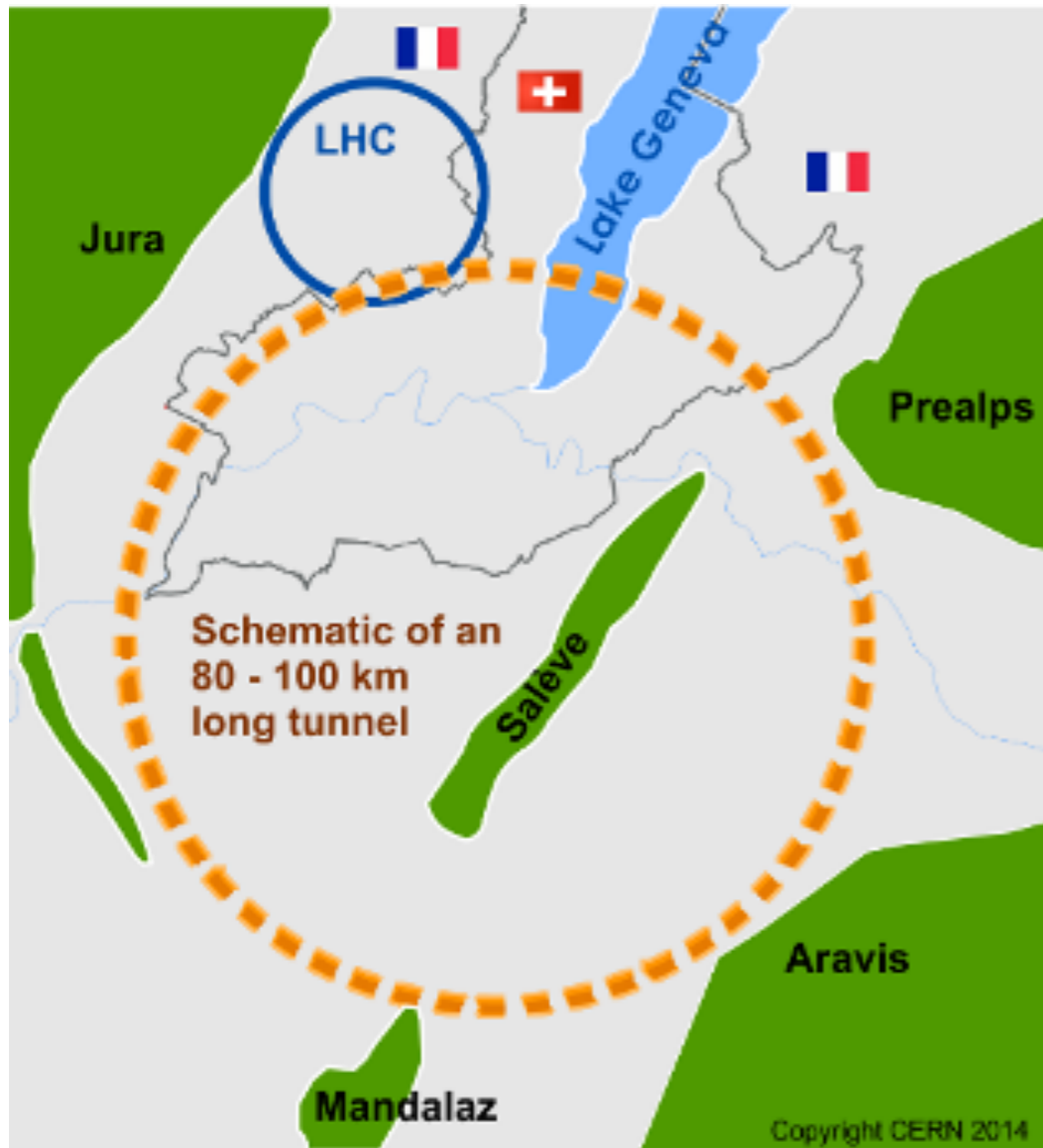
ρ_{ff} does **not** depend on pinning

"effective"

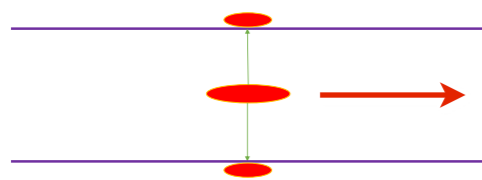
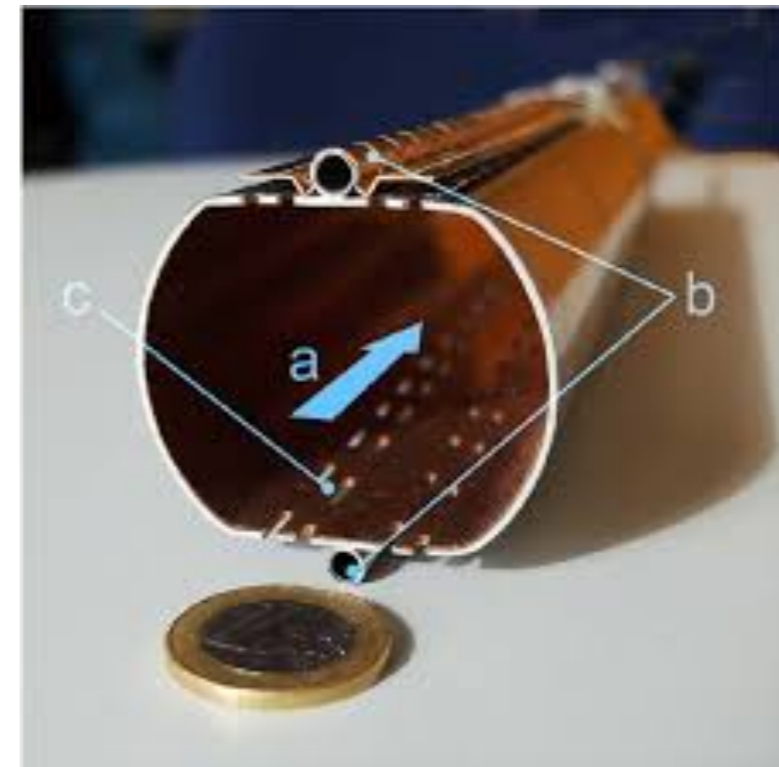
γ_{eff}

k_p strongly depends on pinning

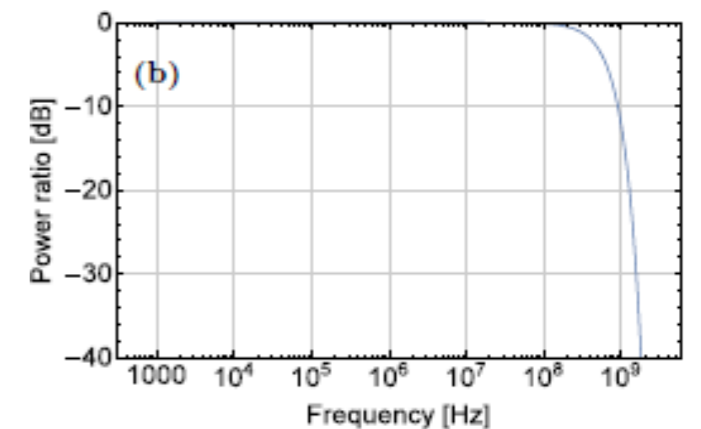
Microwave
measurements
disentangle



Low R_s for wake fields from travelling charged particles. "Beam-screen"



Operates in high-field: 16 T!
(care to energy content at low frequency, below 2 GHz)



→ high depinning frequency v_p

- Cavity experiments for axion detections

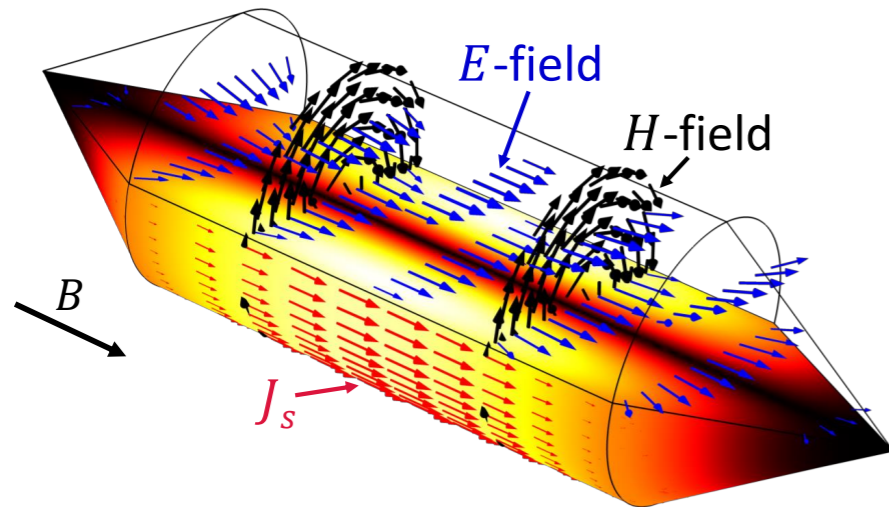
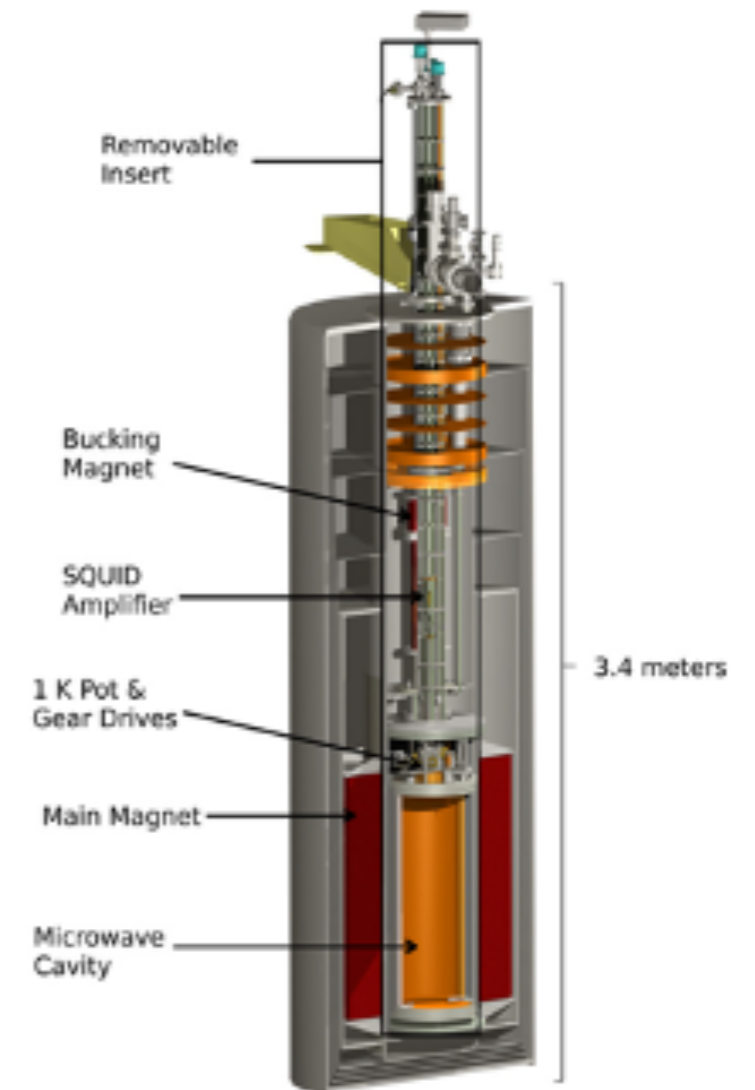


Figure 1. Electromagnetic simulation of the haloscope excited with the TM_{010} mode. The distributions of the radio frequency electric E -field (in blue), the magnetic H -field (in black), and the surface current density J_s (in red) are shown. The direction of the externally applied static B field is shown to be parallel to the longitudinal axis of the cavity.

Operates in “moderate” fields: few T
 Operating frequency 5-25 GHz
 High Q required (above Cu)
 \Rightarrow low R_s

J.S. Sloan et al,
 Physics of the dark universe
 14, 95-102 (2016)



\rightarrow high depinning frequency ν_p

Experimental method: the dielectric resonator

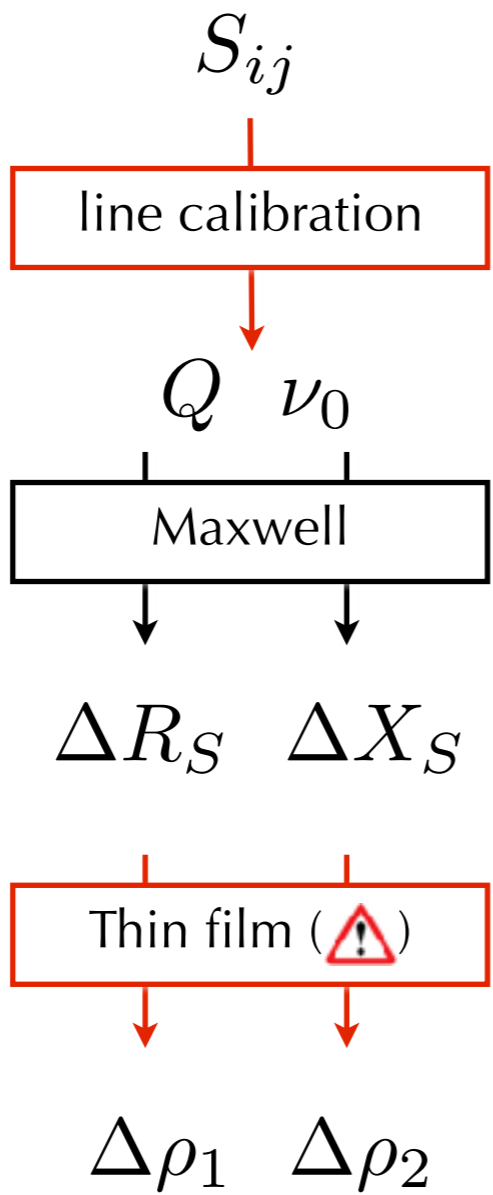
Here of interest:
surface perturbation

Measurand vs. ν
at various T e H :
scattering coefficients

Quality factor,
resonant frequency

variation of the surface
resistance and reactance

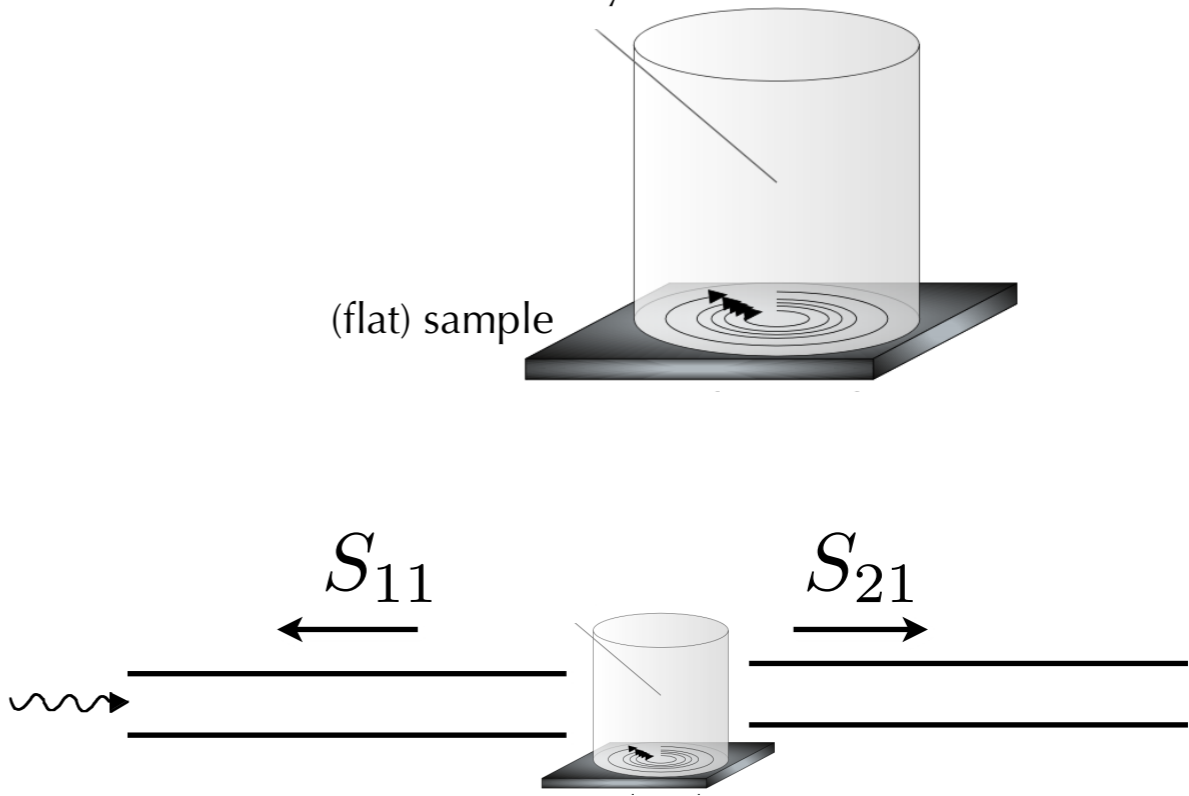
variation of the complex
resistivity



- Difficulties:
- calibrations
 - cryogenics (down to 3 K)
 - high magnetic fields (up to 12 T)

resonant cavity

(flat) sample



Calibration of the resonator

R_S ΔX_S

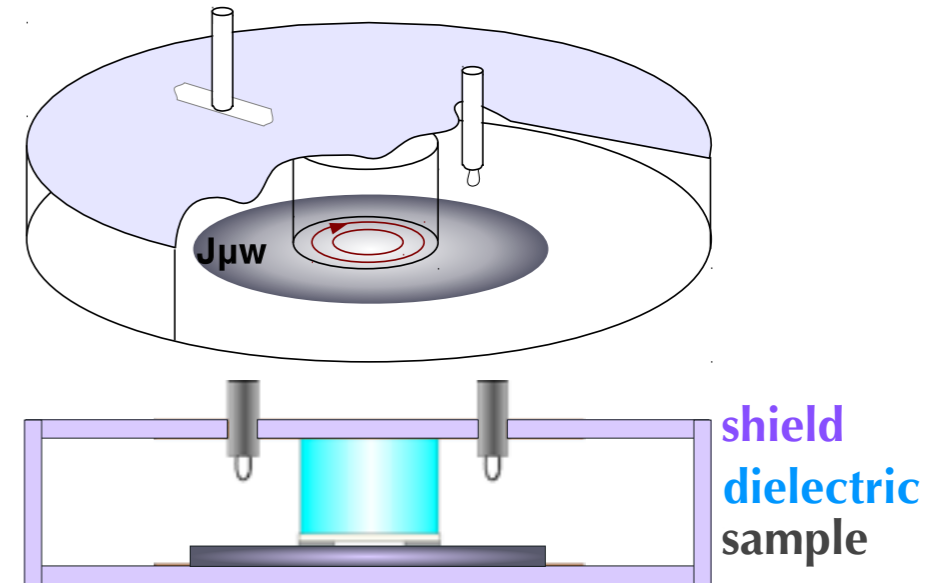
Additional assumptions,
reference measurements

R_S X_S

$$\frac{1}{Q} = \frac{R_s}{G_s} + \frac{R_m}{G_m} + \eta \tan \delta_\epsilon$$

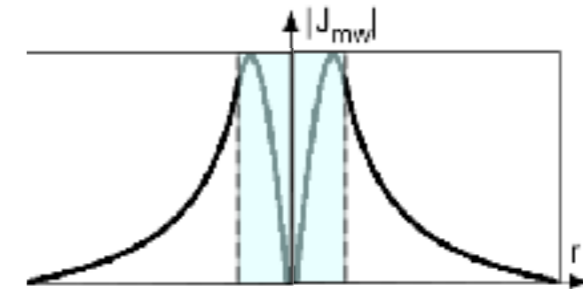
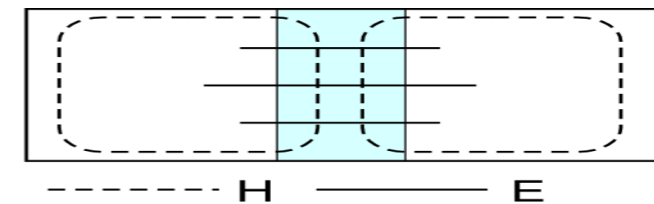
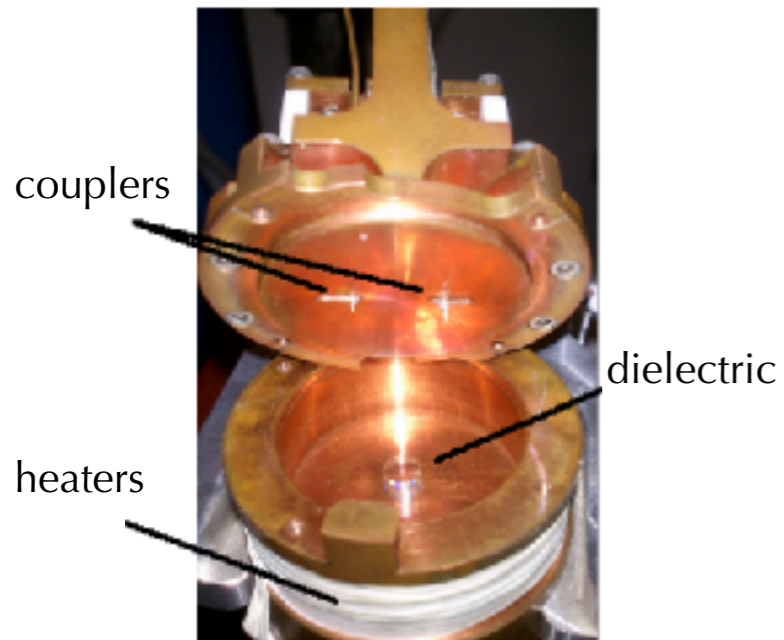
$$-2 \frac{\Delta \nu_0}{\nu_0} = \frac{\Delta X_s}{G_s} + \frac{\Delta X_m}{G_m} + \eta \frac{\Delta \epsilon'_r}{\epsilon'_r}$$

variation of X_s (e.g., with T or H)



16, 27 GHz dual mode sapphire resonator.

8 GHz rutile resonator.



"Hakki-Coleman".

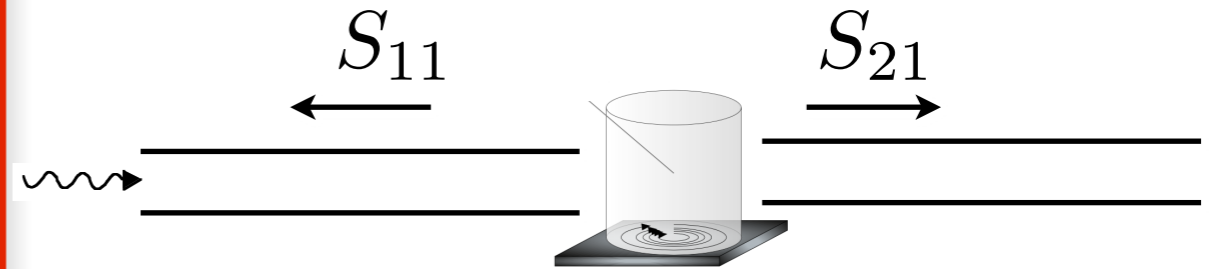
TE₀₁₁: high Q, planar and circular currents

N. Pompeo et al., Meas. Sci. Rev. 14, 164 (2016)
 K. Torokhtii et al J. Phys.: Conf. Ser. 1065 052027 (2018)
 A. Alimenti et al., Meas. Sci. Technol., 30, 065601 (2019)

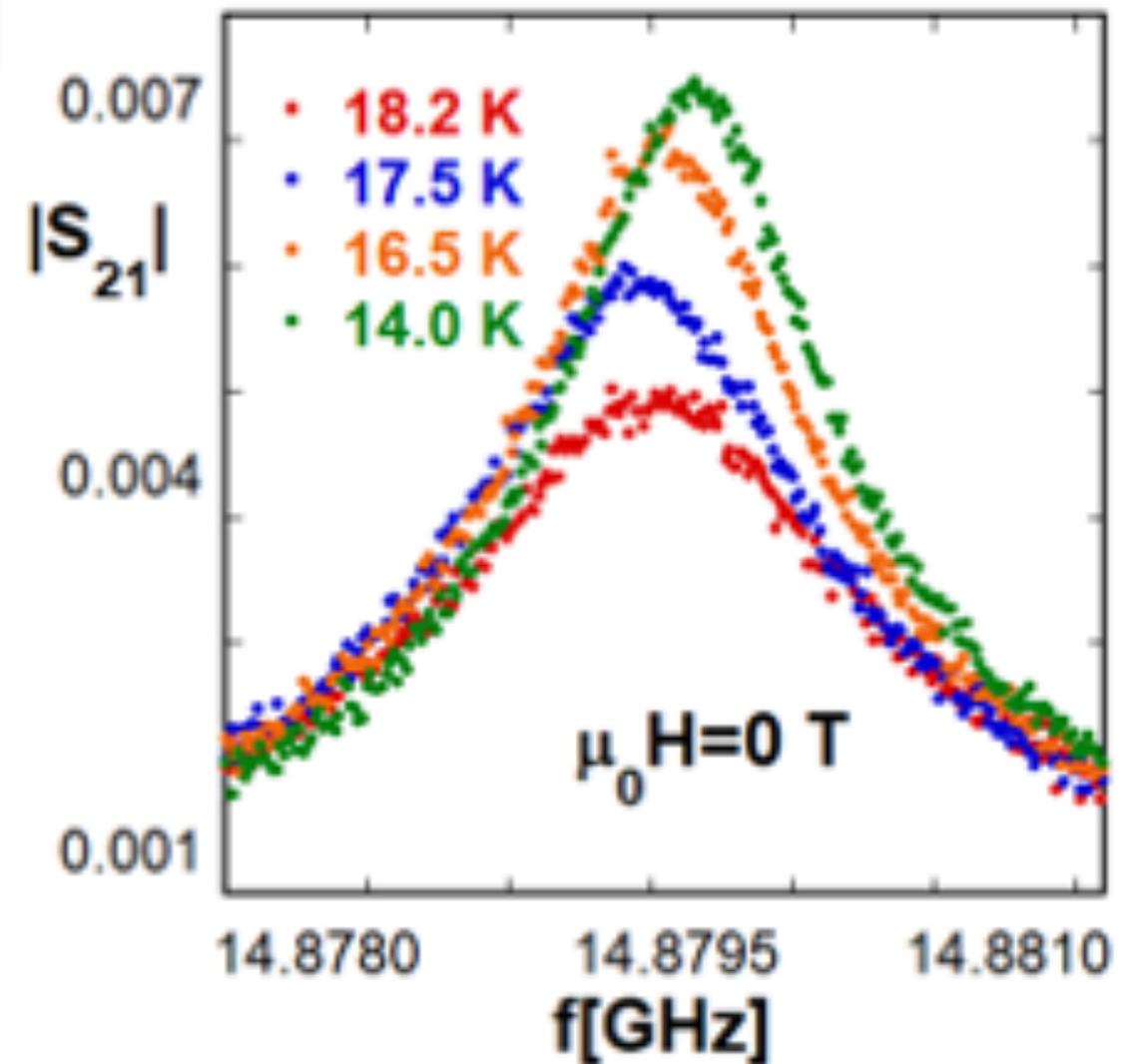
$$\frac{1}{Q} = \frac{R_s}{G_s} + \frac{R_m}{G_m} + \eta \tan \delta_\epsilon$$

$$-2 \frac{\Delta \nu_0}{\nu_0} = \frac{\Delta X_s}{G_s} + \frac{\Delta X_m}{G_m} + \eta \frac{\Delta \epsilon'_r}{\epsilon'_r}$$

variation of X_s (e.g., with T or H)

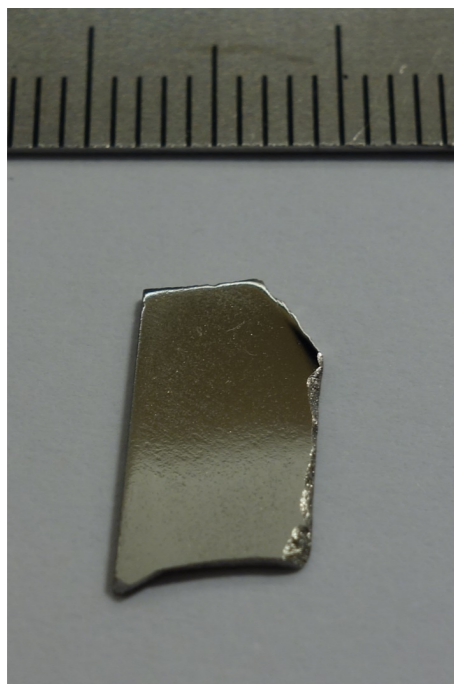
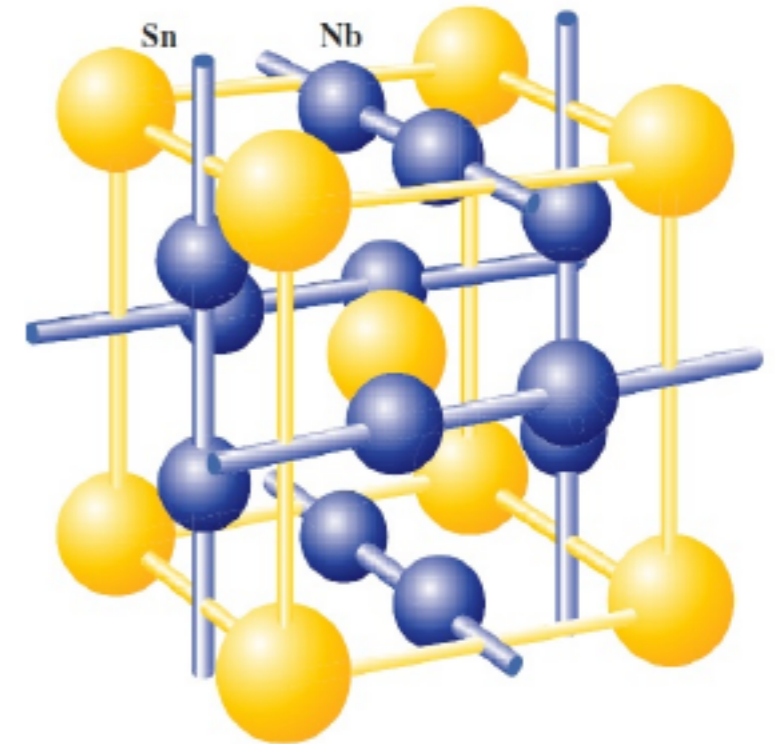


16, 27 GHz dual mode sapphire resonator.



Flux-flow: a Bardeen-Stephen case

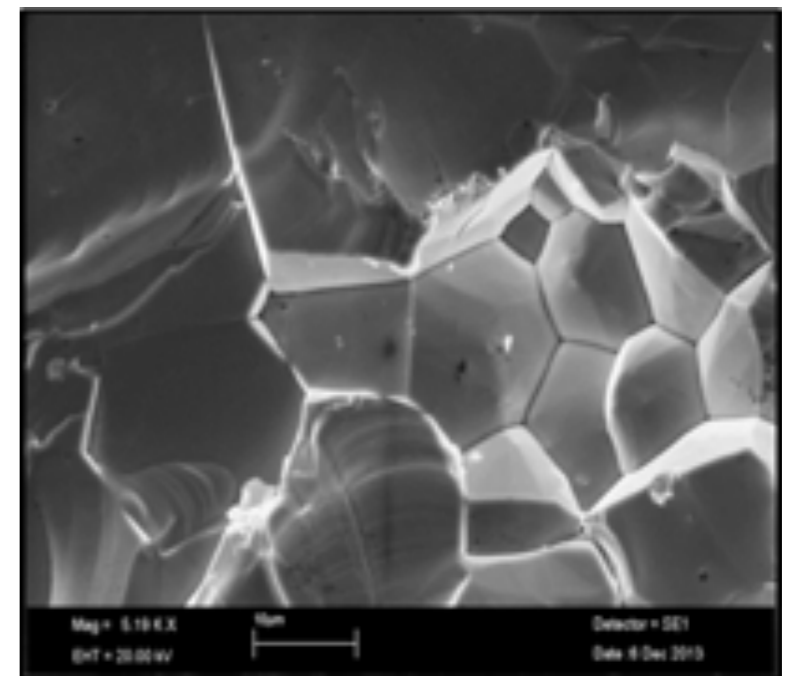
Depinning frequency: promising for specific applications

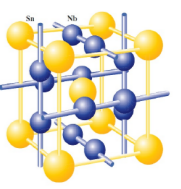


Samples:
R. Flükiger, T. Spina

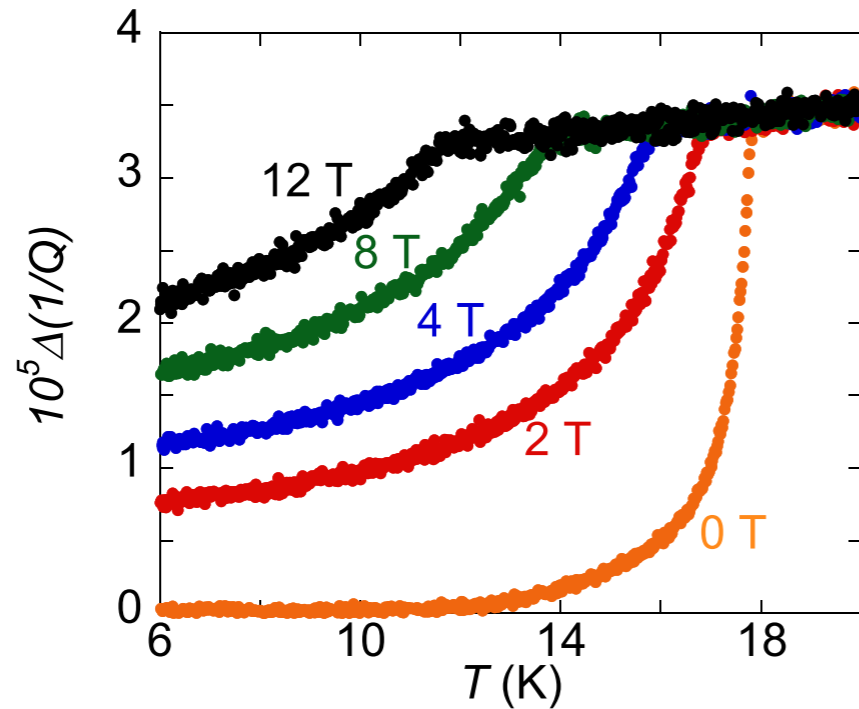
bulk polycrystalline Nb₃Sn platelets
sintering Nb & Sn powder mixture in Ar under Hot Isostatic Pressure

Grain size 20 μm

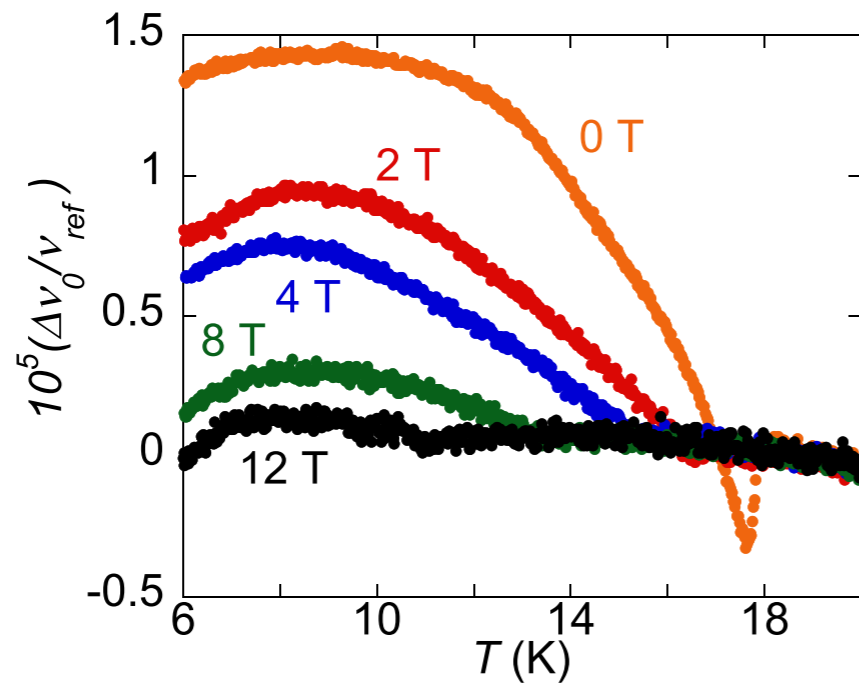




Field-cooled transitions

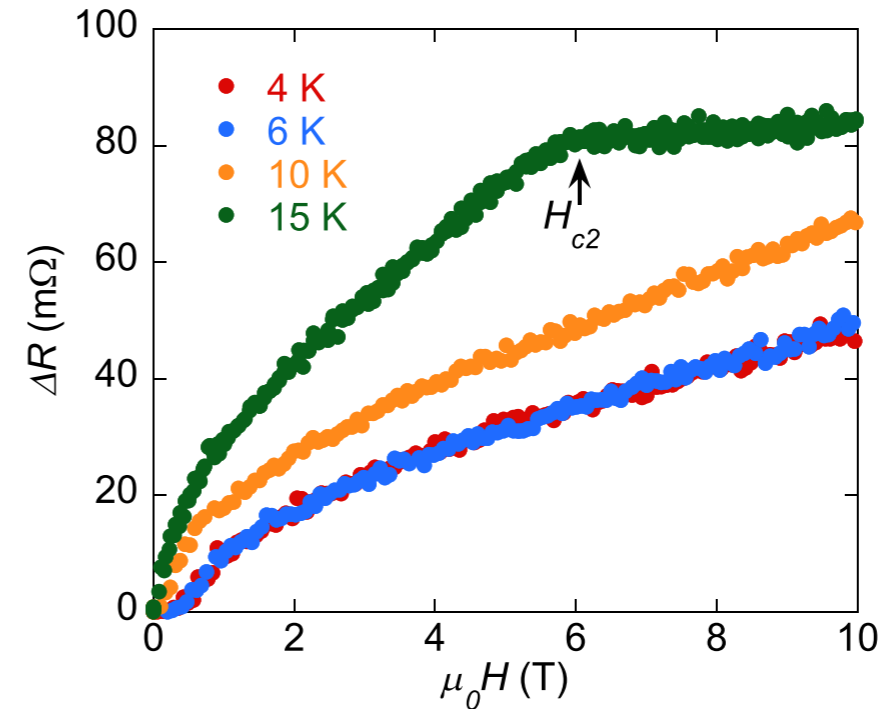


(a)

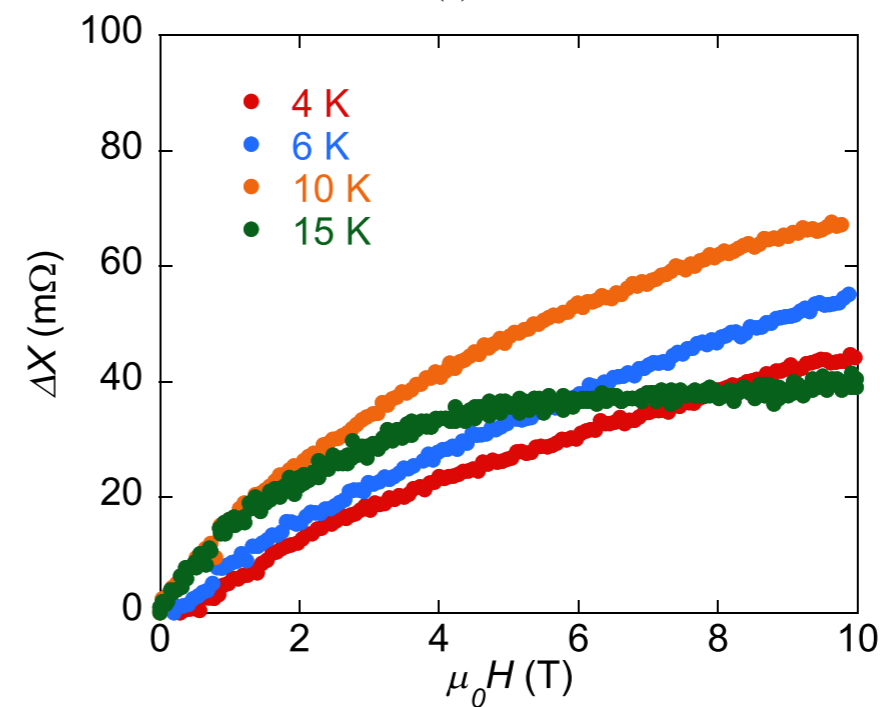


(b)

ZFC field sweeps

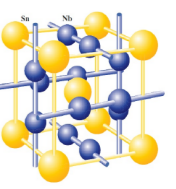


(a)

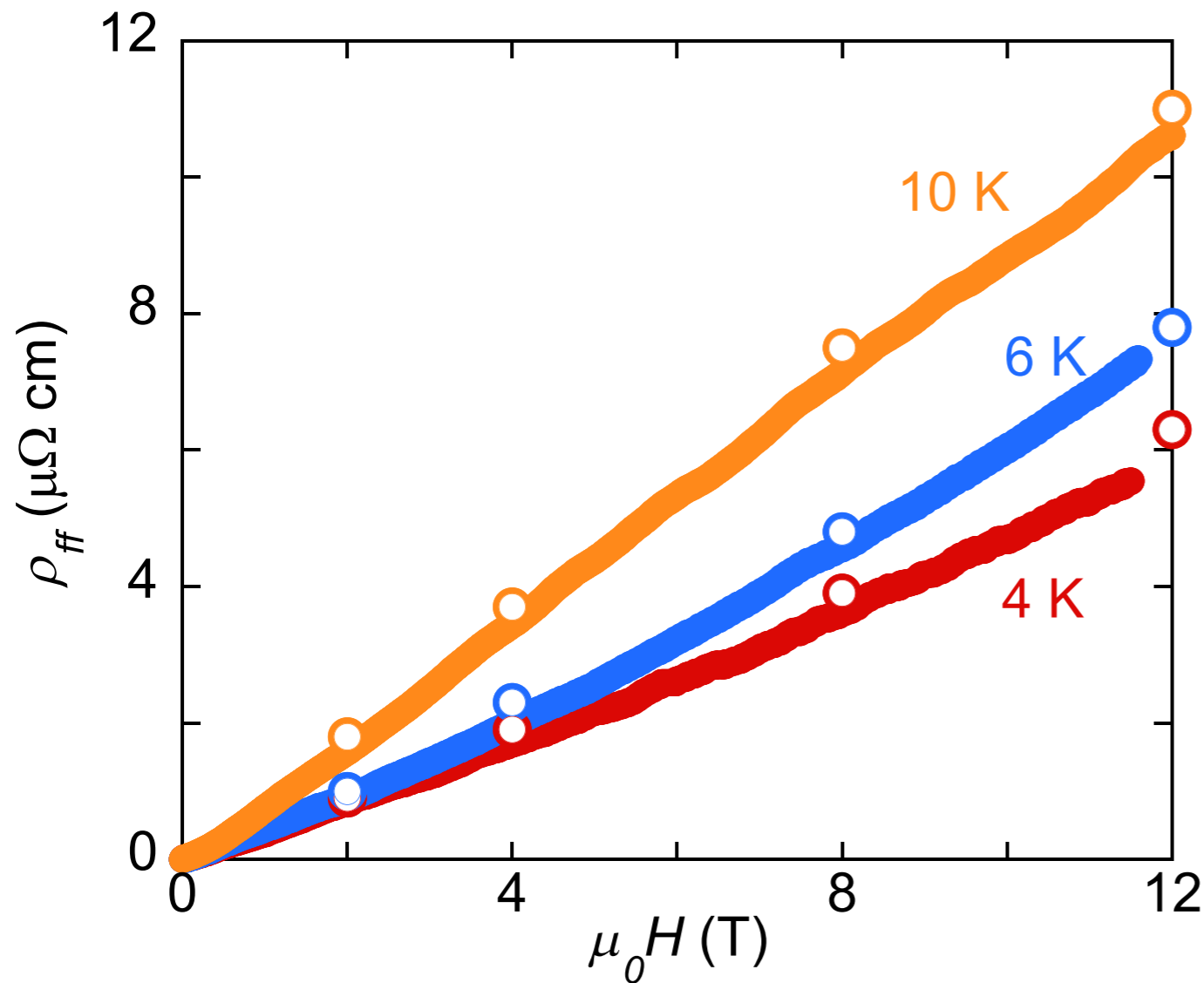


(b)

A. Alimenti et al., Supercond. Sci. Technol. 34 (2021) 014003
IEEE Trans. Appl. Supercond. 29 (2019) 3500104

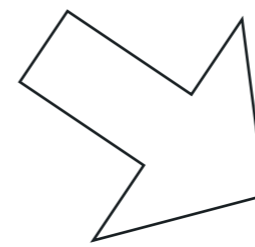


$$Z_S = R_s + iX_s \Rightarrow \rho_1 + i\rho_2 \Rightarrow \rho_{ff}$$

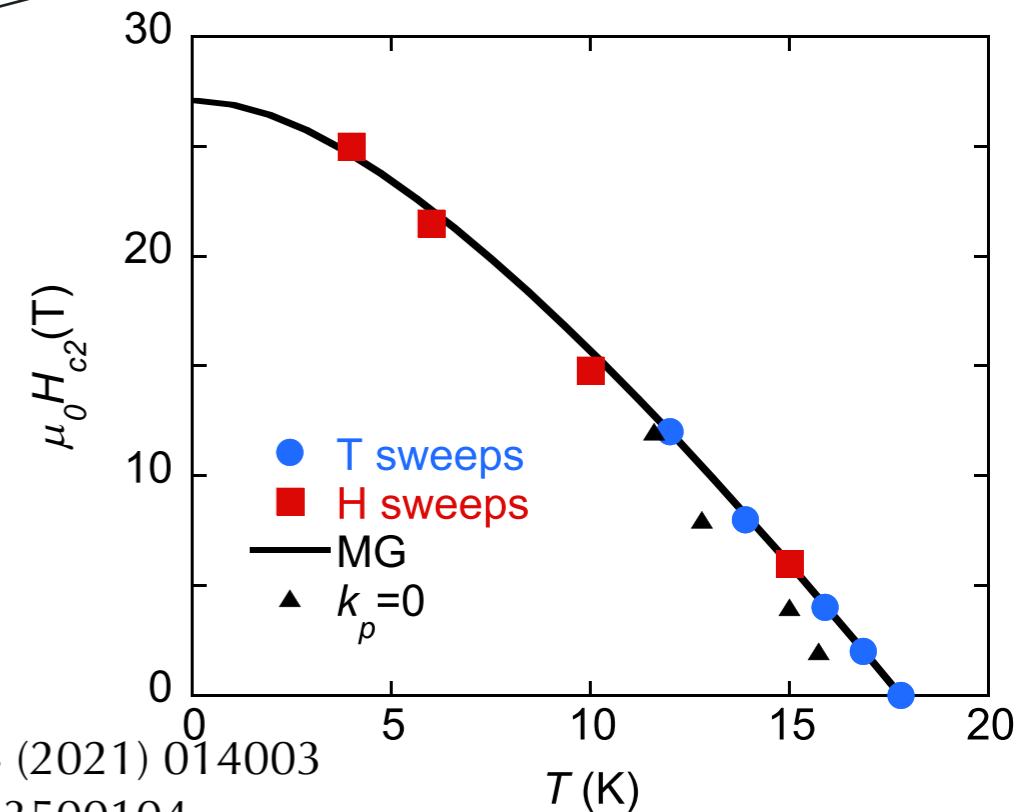


conventional
Bardeen-Stephen

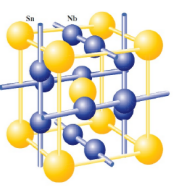
$$\rho_{ff} = \rho_n \frac{H}{H_{c2}}$$



Upper critical field



A. Alimenti et al., Supercond. Sci. Technol. 34 (2021) 014003
IEEE Trans. Appl. Supercond. 29 (2019) 3500104



$$Z_S = R_s + iX_s \Rightarrow \rho_1 + i\rho_2 \Rightarrow k_p$$

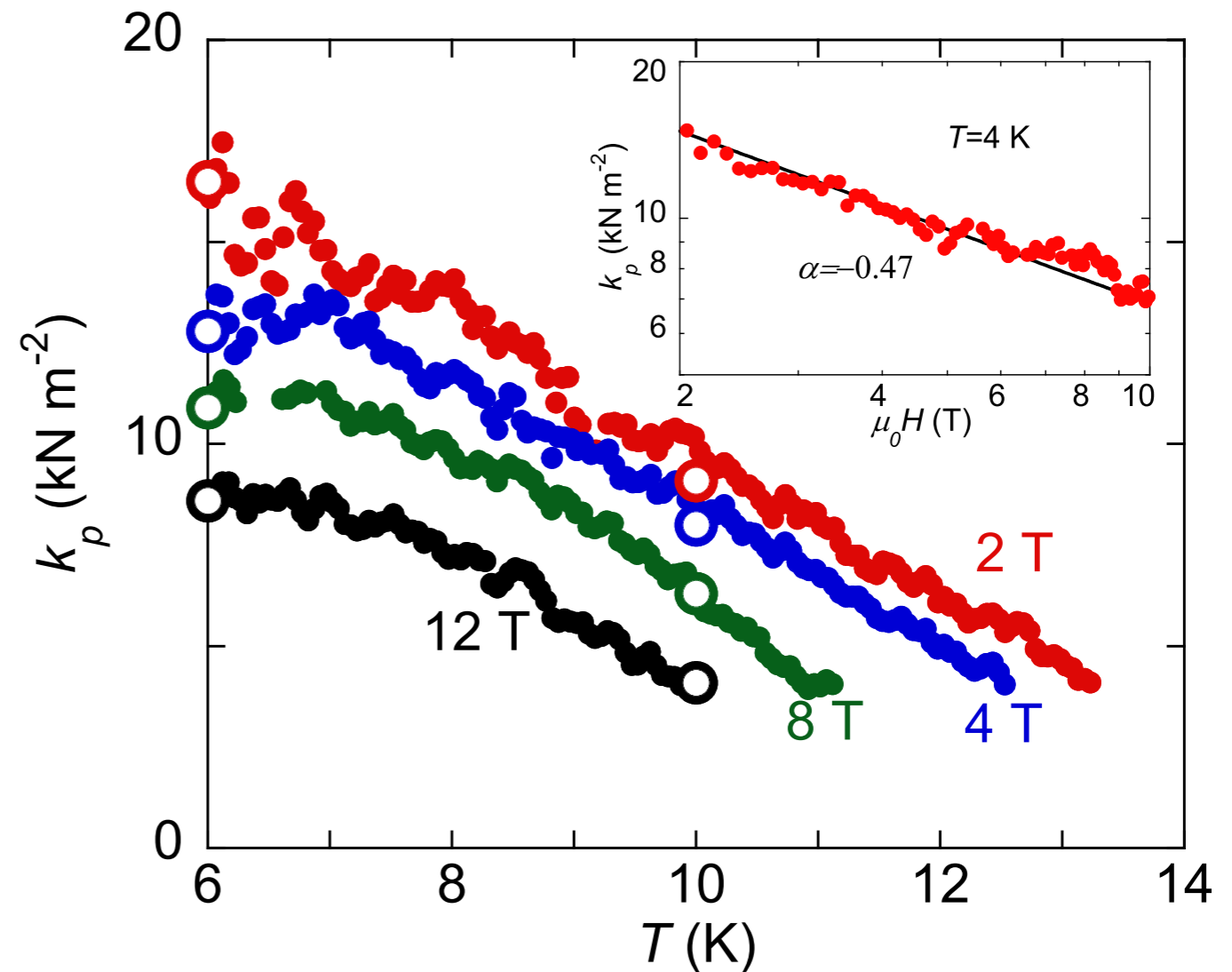
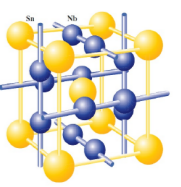
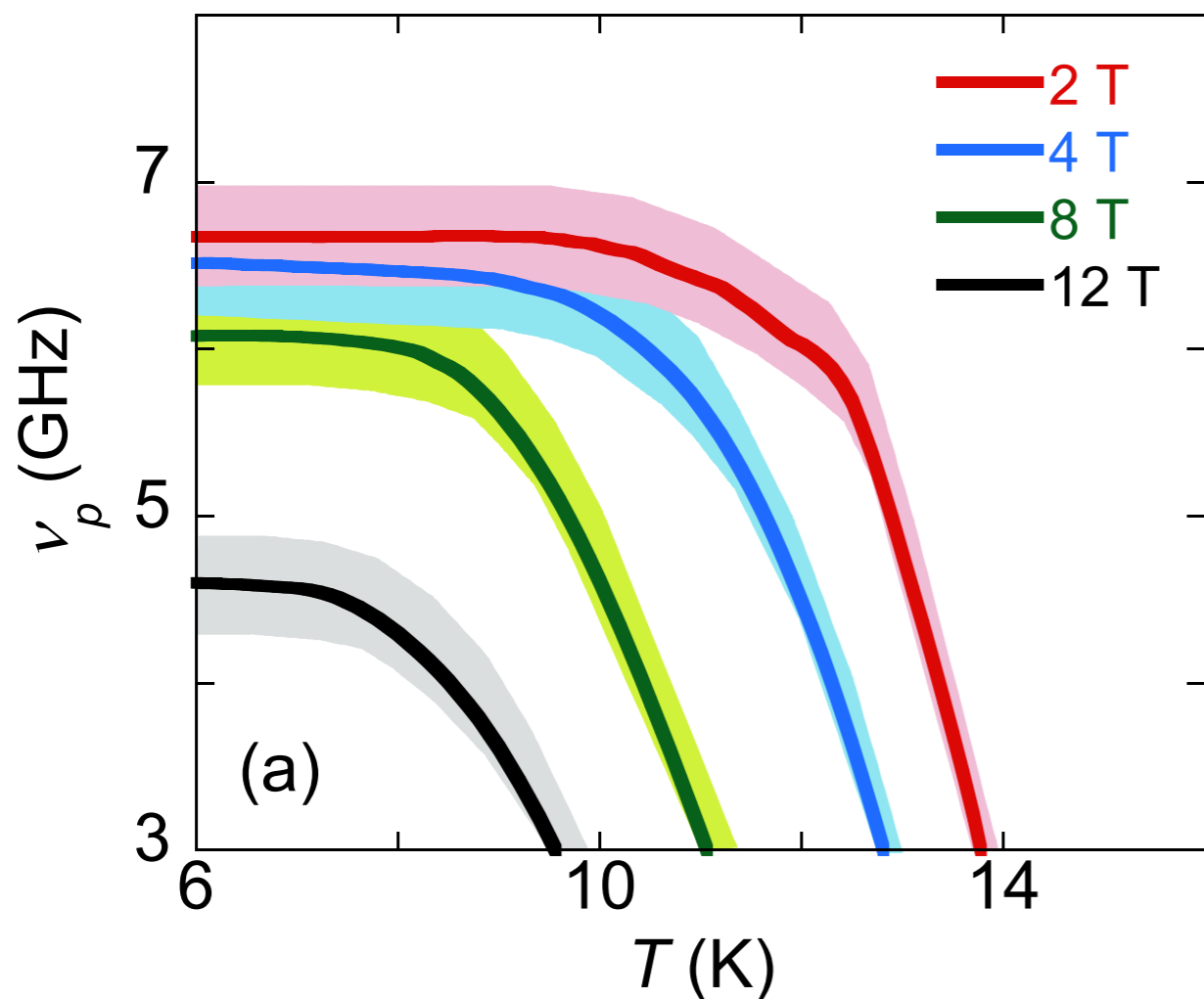


Figure 7. Measured pinning constant $k_p(H, T)$ in FC condition at 2 T, 4 T, 8 T, 12 T. The sparse empty cycles come from the ZFC measurements. In the inset $k_p(H)$ at 4 K is shown in a log-log plot to highlight the power dependence $k_p \propto H^\alpha$, with $\alpha = -0.47$, typical of the collective pinning regime.

A. Alimenti et al., Supercond. Sci. Technol. 34 (2021) 014003
IEEE Trans. Appl. Supercond. 29 (2019) 3500104



$$Z_S = R_s + iX_s \Rightarrow \rho_1 + i\rho_2 \Rightarrow \nu_p$$



Almost constant at low T
High values (with respect to Nb):
potential use for rf applications

Compare to:

Nb *thin films*:

- $t = 160 \text{ nm}$ $\nu_p \approx 1 \text{ GHz}$ ($T = 0.5 T_c$)

D Janjusevic et al, Phys Rev B, **74** 104501 (2006)

YBCO + BaZrO₃ *thin films*:

- $t = 100 \text{ nm}$ $\nu_p \approx 50 \text{ GHz}$ ($T = 0.8 T_c$)

N. Pompeo et al, APL 91, 182507 2007

YBCO – *coated conductor*:

- $t = 175 \text{ nm}$ $\nu_p \approx 70 \text{ GHz}$ ($T = 77 \text{ K}$)

K Torokhtii et al, IEEE-TAS, **26** 8001605 (2016)

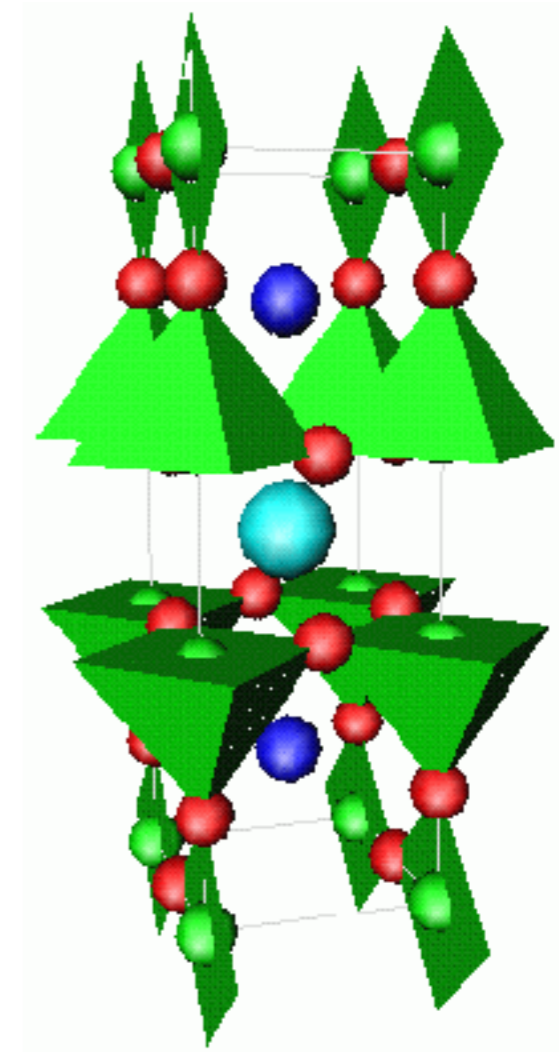
N. Pompeo et al, IEEE-TAS (2019)

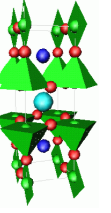
DOI: 10.1109/TASC.2019.2897956

A. Alimenti et al., Supercond. Sci. Technol. 34 (2021) 014003
IEEE Trans. Appl. Supercond. 29 (2019) 3500104

$\text{YBa}_2\text{Cu}_3\text{O}_{7-x}/\text{BaZrO}_3$

Vortex motion up to high fields





G. Celentano

Thin films YBCO - 5 mol.% BZO sample on SrTiO₃ single crystal substrate

grown by chemical deposition

BaZrO₃ nanoparticles (Ø~10 nm)

SEM

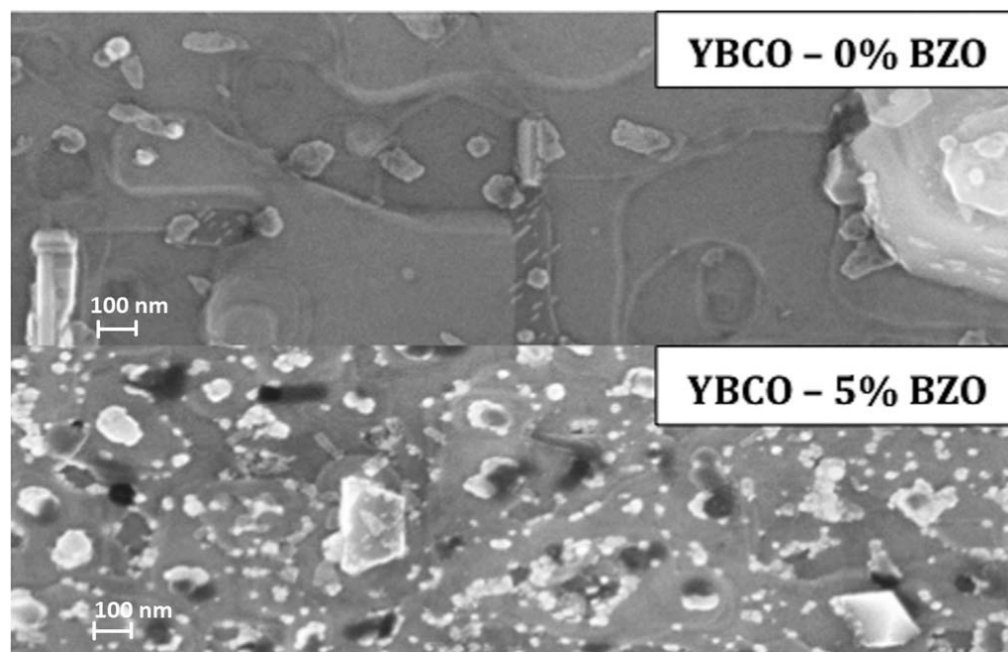
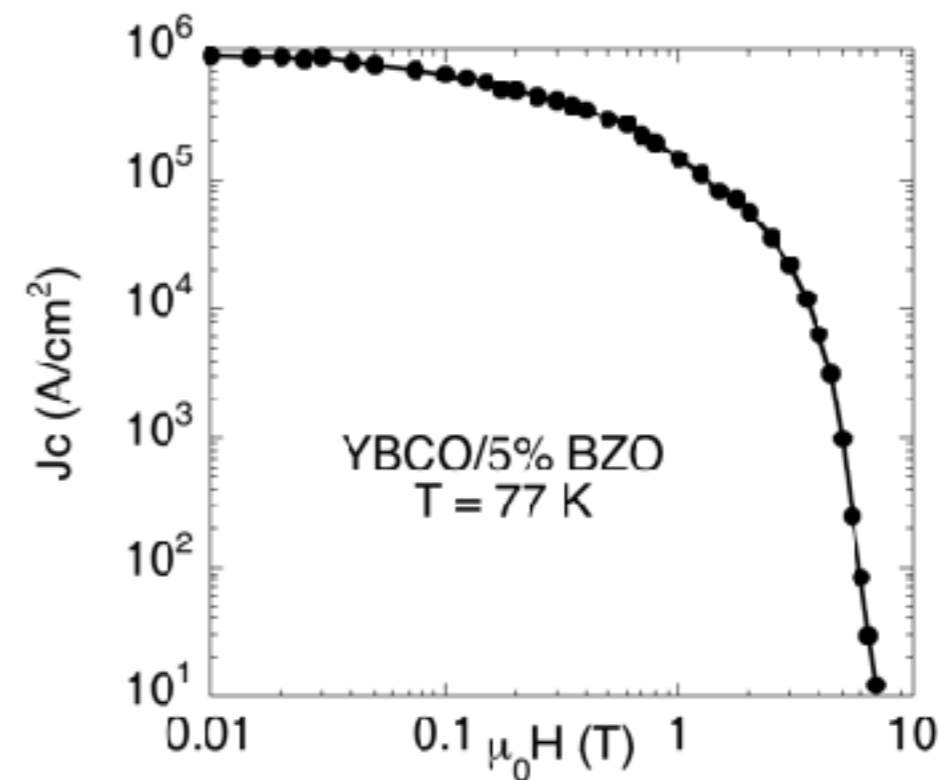


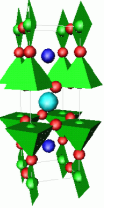
Fig. 1. SEM images of pure YBCO (upper panel) and YBCO/BZO (lower panel) films. The magnification is 100 k × for both images.

Critical current density

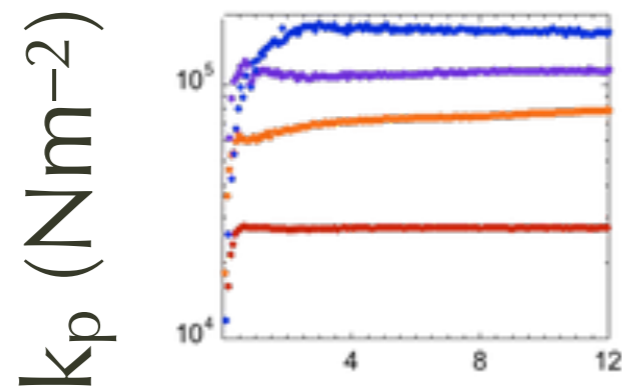
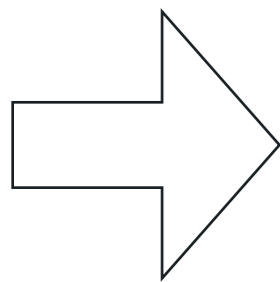
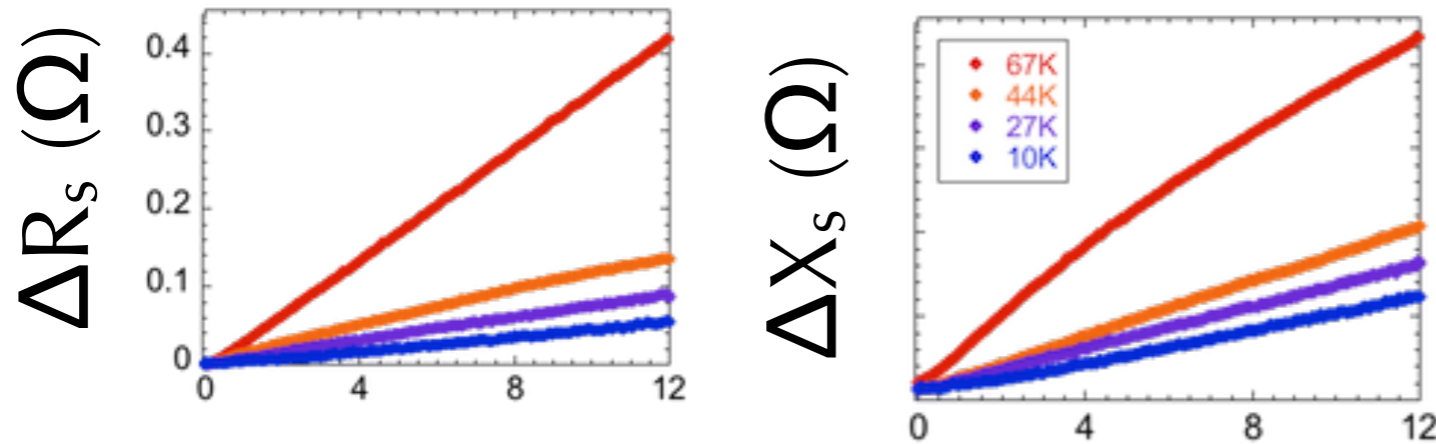


Sharp drop above 1 T
Typical of nanostructured YBCO

A. Frolova et al, IEEE TAS 26, 8001205
(2016)

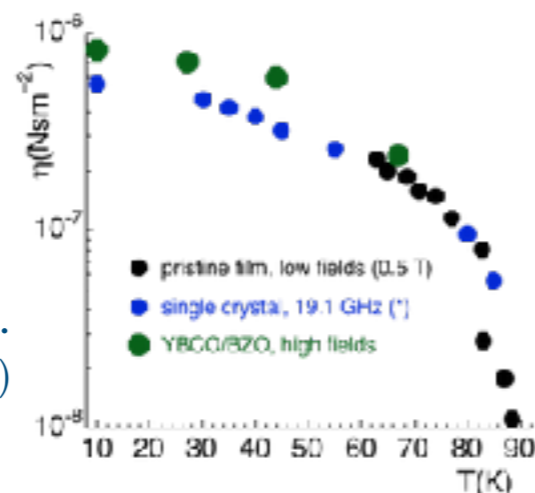


field-sweeps: data analysis



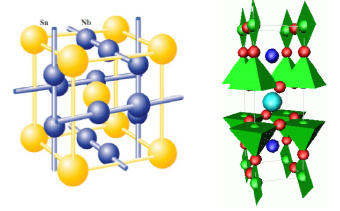
k_p is field-independent up to 12 T:
BZO nanoparticles produce very *steep* potential wells
(height is less efficient: J_c drops)

vs. T

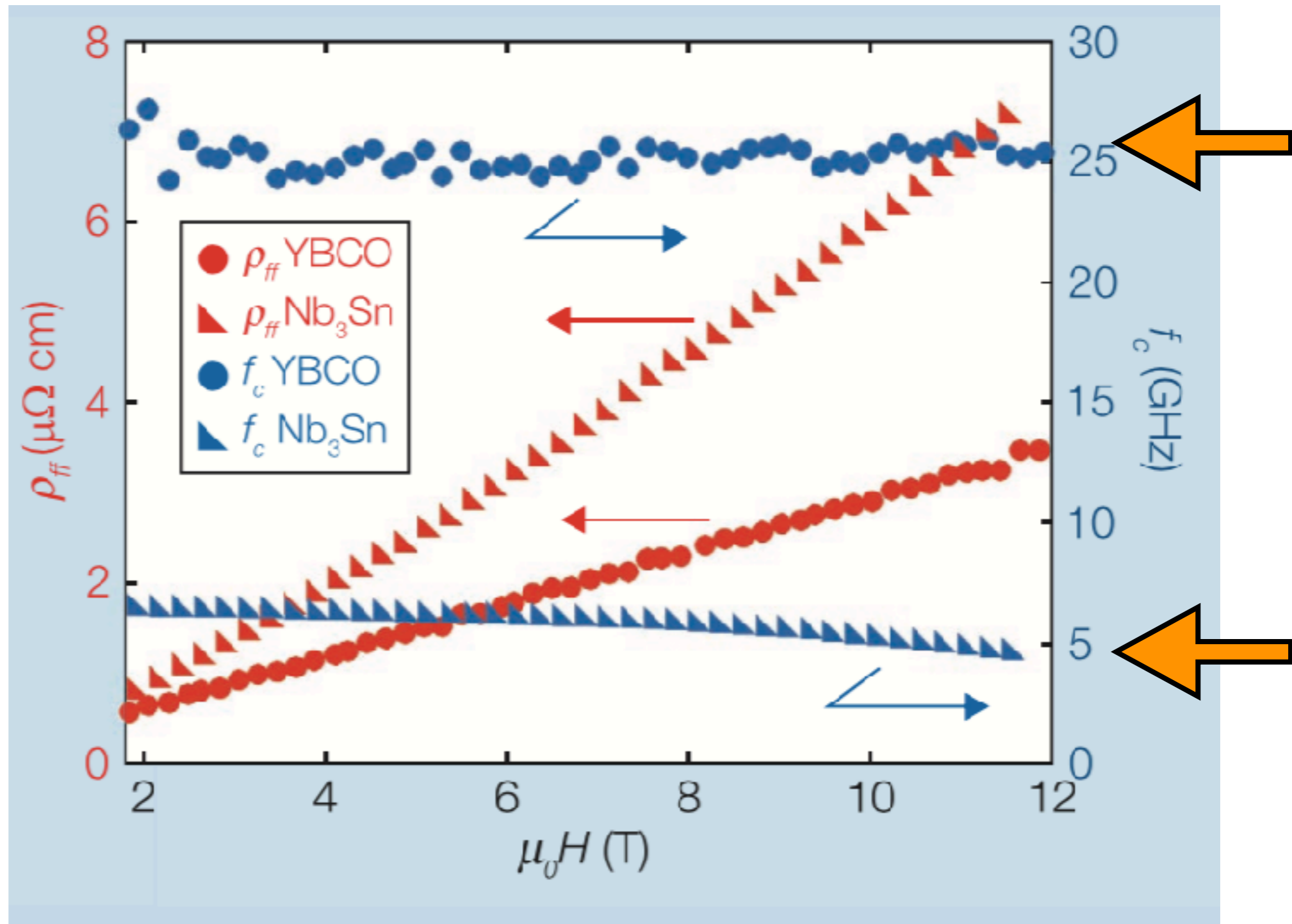


Vortex viscosity compares well with data
on single crystals and pristine films:
BZO nanoparticles do not affect the microscopic
(quasiparticle) state

(*) Tsuchiya et al.
PRB 63, 184517 (2001)



$T/T_c \approx 1/3$
 YBCO: 27 K
 Nb₃Sn: 6 K



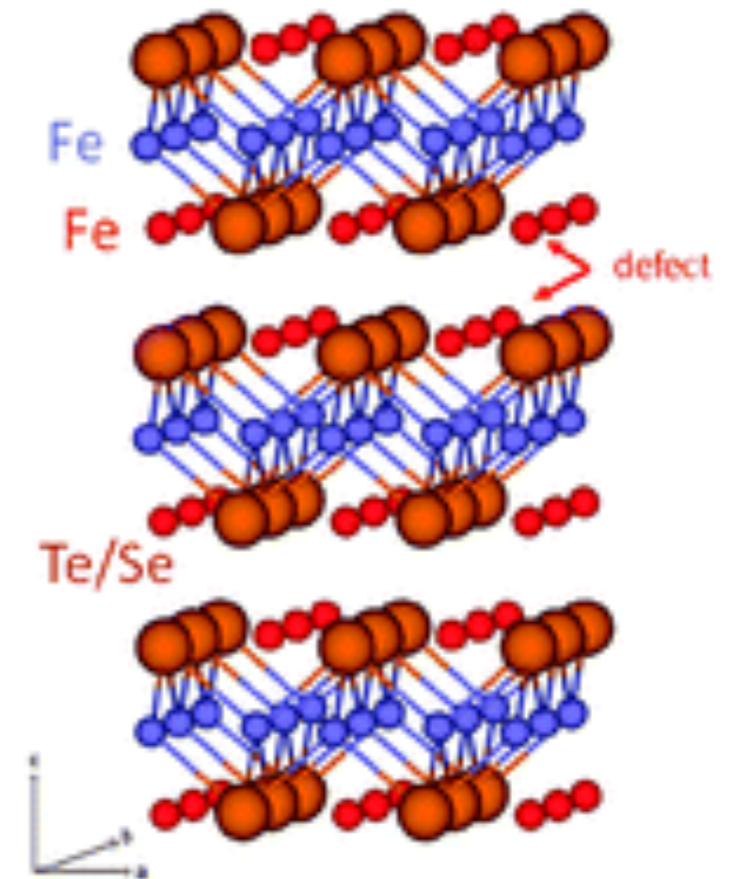
A. Alimenti et al. IEEE Instrum. and Meas. Mag. p.12 (dec. 2021)

N. Pompeo et al., in preparation

Fe(Se,Te)

a new player in the arena?

samples: thin films

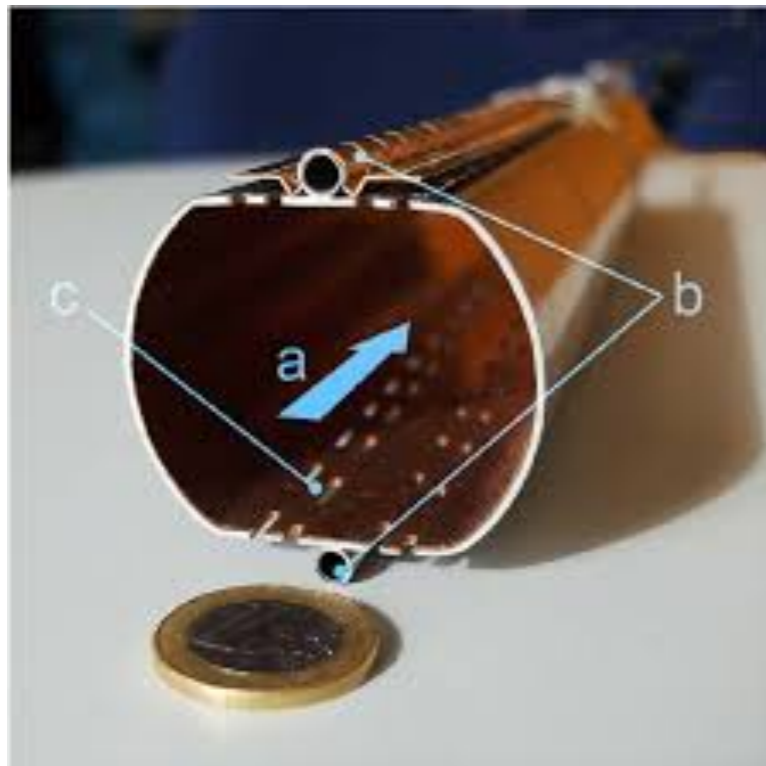


Samples:
V. Braccini

Perspectives

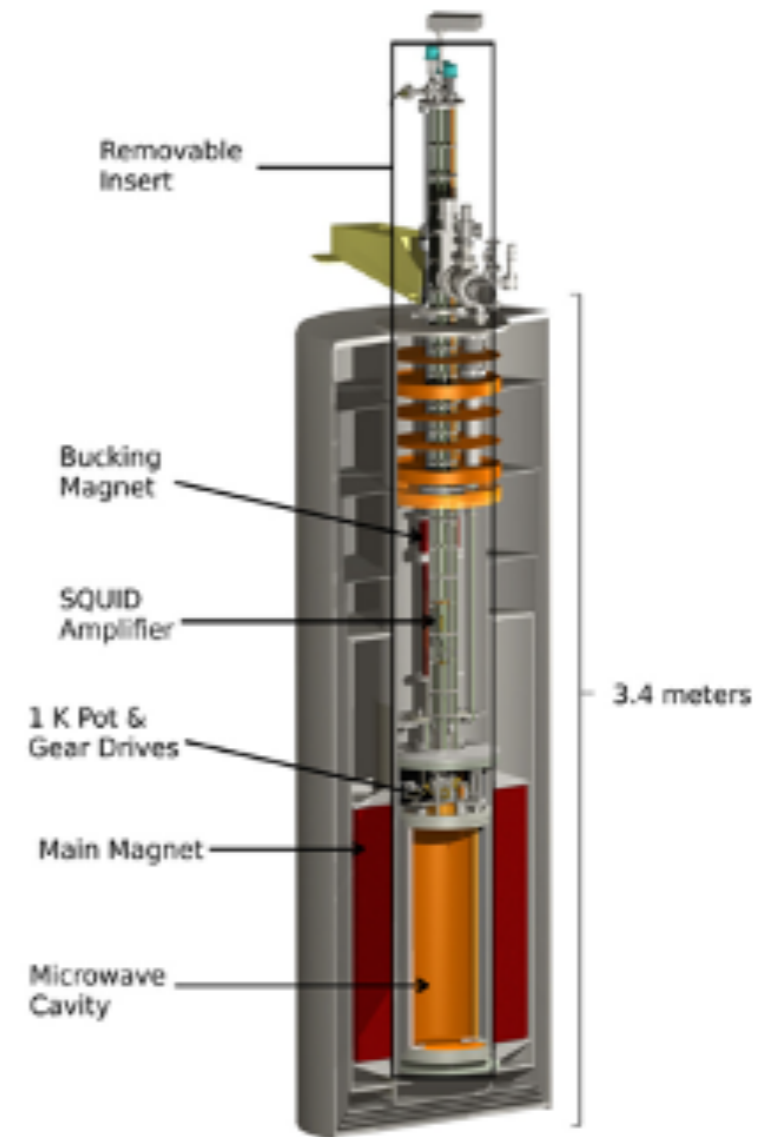
CERN-FCC

Low R_s for wake fields from travelling charged particles. "Beam-screen"



Operates in high-field: 16 T!

Cavity experiments for axion detections



Operates in "moderate" field: 2-8 T

→ high depinning frequency ν_p

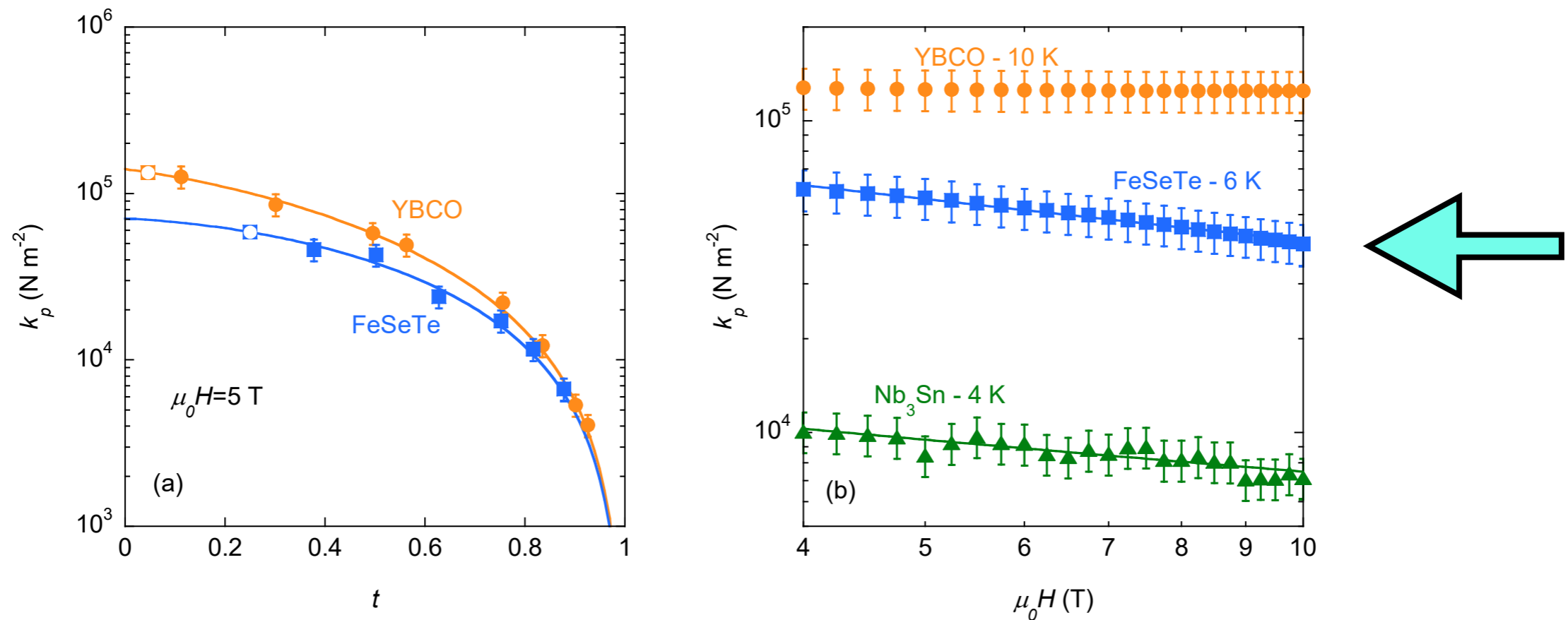
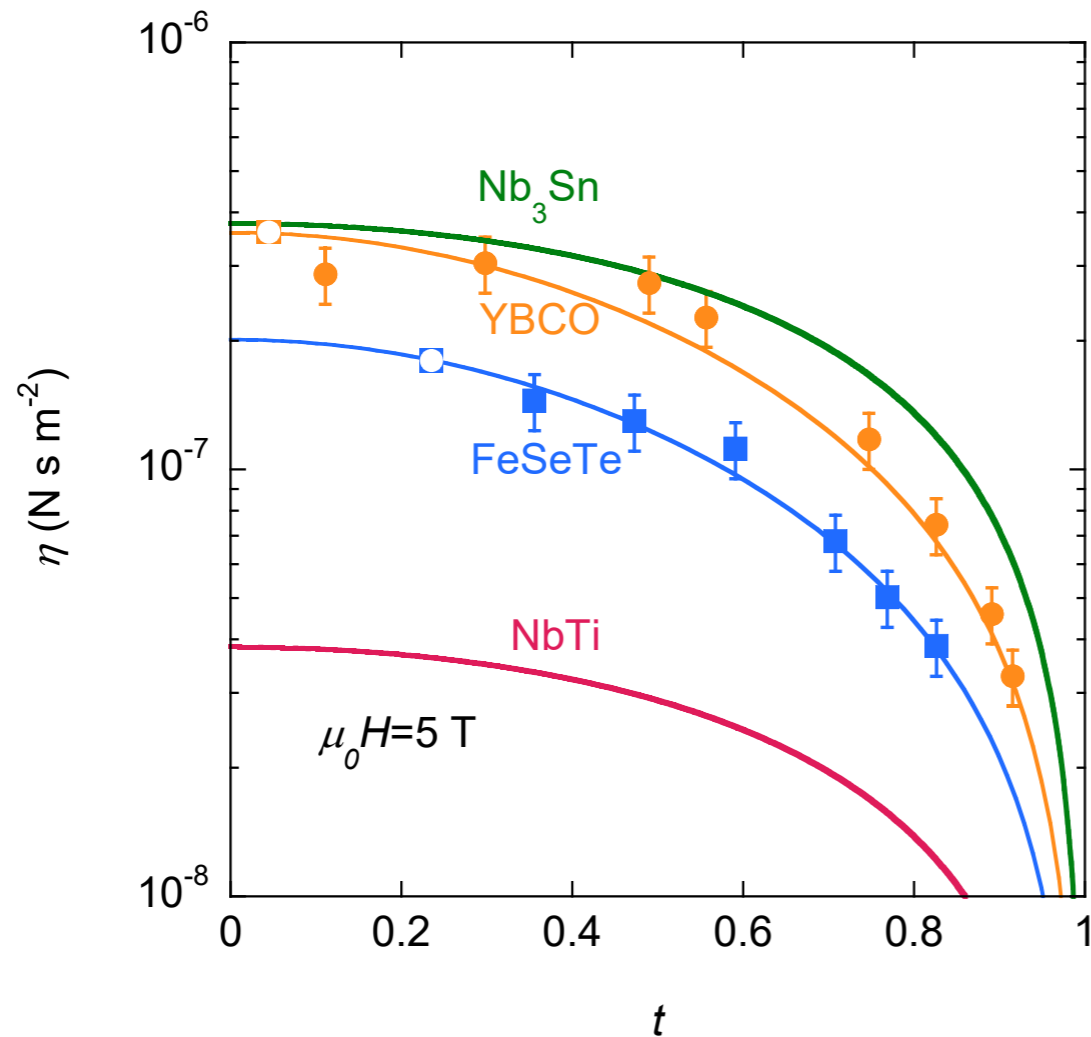


Figure 2. (a) Pinning constant k_p , measured on YBCO (orange circles) and FeSeTe (blue squares) in zero field cooling conditions (ZFC) at 5 T as a function of the reduced temperature $t = T/T_c$. The fit of the data is performed with the model developed in [46]: $k_p(t) = k_p(0)(1 - t)^{4/3}(1 + t)^2 e^{-T/T_0}$. The best fit parameters are: $k_p(0) = 1.40 \times 10^5 \text{ N m}^{-2}$ and $T_0 = 56 \text{ K}$ for YBCO, $k_p(0) = 7.07 \times 10^4 \text{ N m}^{-2}$ and $T_0 = T_c$ for FeSeTe. The empty square symbols represent the points extrapolated from the fit at 4 K (which will be used in the next elaborations). (b) Field dependence of k_p in Nb₃Sn (green triangle), YBCO, and FeSeTe samples measured at the lower temperatures. $k_p(H)$ in Nb₃Sn and FeSeTe is shown to be $\propto H^{-0.5}$.



FeSeTe: not too low viscosity

$$\rho_{ff} = \Phi_0 \frac{\mu_0 H}{\eta}$$

Figure 3. Viscous drag coefficient η measured on YBCO (orange circles) and FeSeTe (blue squares) at 5 T in ZFC as a function of the reduced temperature t . The measured data are approximated with $\eta(t) = \eta(0)(1 - t^2)/(1 + t^2)$ [46], with $\eta(0) = 3.59 \times 10^{-7} \text{ N s m}^{-2}$ for YBCO and $\eta(0) = 2.02 \times 10^{-7} \text{ N s m}^{-2}$ for the FeSeTe sample. The empty square symbols represent the points extrapolated from the fit at 4 K (which will be used in the next elaborations). The green and red curves represent $\eta(t)$ obtained by considering $\eta(t) = \Phi_0 B_{c2}(t)/\rho_n$ (from $\rho_{ff} = \rho_n B/B_{c2}$ and $\eta = \Phi_0 B/\rho_{ff}$) for Nb₃Sn and NbTi, respectively.

parametric plot

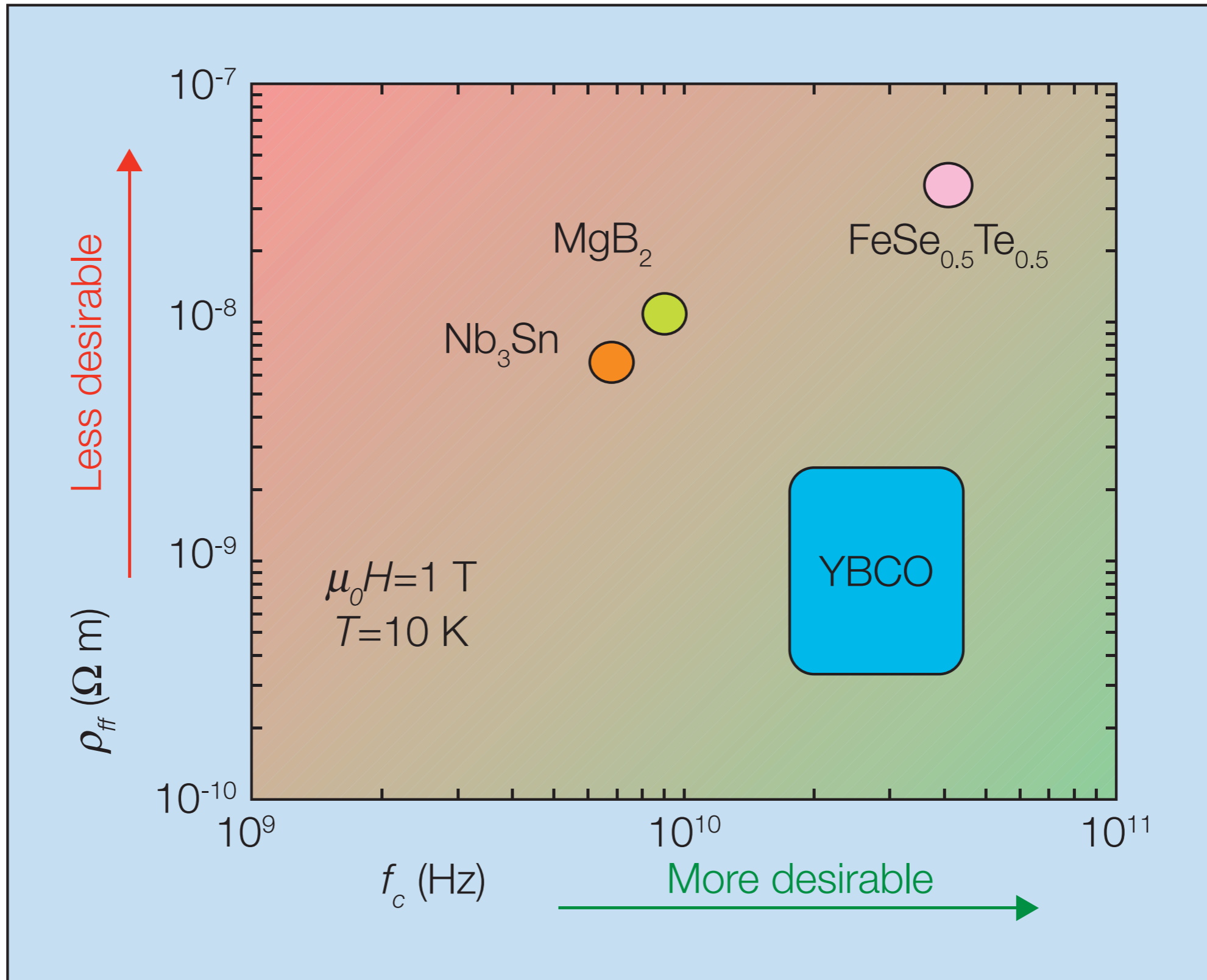


Table 2. Measured parameters of the SCs of interest used for the evaluation of R_s .

	NbTi	Nb ₃ Sn	YBCO	FeSeTe
k_p (N m ⁻²) at 4 K, 5 T	–	$(9.9 \pm 1.5) \times 10^3$	$(1.3 \pm 0.2) \times 10^5$	$(5.9 \pm 0.9) \times 10^4$
pinning regime	single	collective	single	collective
η (N s m ⁻²) at 4 K, 5 T	–	–	$(3.6 \pm 0.5) \times 10^{-7}$	$(1.8 \pm 0.3) \times 10^{-7}$
ρ_n ($\mu\Omega$ cm)	70 ± 2	14.8 ± 0.2	–	–
$B_{c2}(0)$ (T)	13.0 ± 0.5	27.0 ± 0.5	–	–
ν_p (GHz) at 4 K, 5 T	44 ± 7	4.2 ± 0.6	59 ± 9	52 ± 8
meas. geometry	$H \parallel J$	$H \perp J$	$H \parallel c$ axis, $H \perp J$	
γ	1	1	5.3 ± 0.7	1.8 ± 0.2
$\lambda(0)$ (nm)	270 ± 20 [17,47]	160 ± 20 [48]	150 ± 10 [49,50]	520 ± 50 [45]

optimization of Fe(Se,Te) possible (nanoengineering)!
 Maybe better or on par with YBCO,
 but good perspectives **for growing as a coating** (at least, FeSe)
 (cavity people would be very thankful...)

- **Microwaves: a versatile and powerful tool for vortex physics**
- New rf requirements for operation in a dc magnetic field
- Nb₃Sn and YBCO/BZO are mature major players in rf high-field experiments
- FeSeTe: at intrinsic properties level, competitive for rf with Nb₃Sn and, in perspective, with YBCO (but still non mature from material science point of view)

Thank you for your attention

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